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- 54 Permanent magnet alloy.
- (a) A permanent magnet of R₂Co₁₇ type crystal structure consisting of, in percent by weight, at least one rare earth element within the range of 24 to 28, cobalt within the range of 48 to 53, copper within the range of 2 to 4.9, iron within the range of 18 to 30 and zirconium within the range of 1.7 to 3.0. By substituting zirconium for a portion of copper an optimum combination of coercive force and residual magnetization (saturation induction) may be achieved.

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PERMANENT MAGNET ALLOY

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The present invention relates to permanent magnets of R_2Co_{17} type crystal structure.

It is known in the production of rare earth-cobalt permanent magnets (R-Co) that iron may be used to replace a significant portion of the cobalt when zirconium is added to the composition. It is also known that additions of copper may be made to compositions of this type. However, with a copper addition the residual magnetization (saturation induction) is decreased. Likewise as iron is increased there is a corresponding reduction in coercive force.

It is accordingly a primary object of the present invention to provide a permanent magnet alloy containing at least one rare earth element, preferably samarium, cobalt, iron and copper wherein an optimum combination of coercive force and residual magnetization is achieved.

A more specific object of the invention is to provide a permanent magnet alloy of this type wherein copper is reduced and replaced by a zirconium addition, whereby an optimum combination of coercive force and residual magnetization may be achieved.

The present invention provides a permanent magnet of R₂Co₁₇ type crystal structure consisting of, in percent by weight, at least one rare earth element within the range of 24 to 28, cobalt within the range of 48 to 53, copper within the range of 2 to 4.9 to less than 5, iron within the range of 18 to 30 and zirconium within the range of 1.7 to 3.0.

Broadly in the practice of the invention the permanent magnet is of an alloy of the general formula R₂Co₁₇ wherein R is at least one rare earth component, preferably samarium, and the Co component is cobalt. The alloy, in weight percent, consists of at least one rare earth element, preferably samarium, within the range of 24 to 28, cobalt within the range of 48 to 53, copper within the range of 2 to 4.9, iron within the range of 18

to 30 and zirconium within the range 1.7 to 3.0. By maintaining copper at an amount less than 5% and adding zirconium within the above stated range, the adverse affect of copper with regard to residual magnetization is eliminated and thus an optimum combination of coercive force and residual magnetization is achieved.

As specific examples of the practice of the invention the following alloy compositions were employed:

7.0		Weight	, Perc	ent		
10	Alloy	Sm	<u>Co</u>	Cu	<u>Fe</u>	Zr
	A	25.6	49.9	3.8	18.6	1.7
	В	25.1	48.7	3.9	19.7	2.6
	Commercial Alloy	26.6	50.4	5.9	14.7	2.4

15 The alloys were produced by induction melting and casting, whereupon they were then crushed and ball milled to a particle size within the range of 5 to 10 microns. The powder was then oriented in a magnetic field and samples thereof were both pressed by a pulsating magnetic 20 field in combination with hot isostatic pressing and also by die pressing in a transverse magnetic field. Thereafter, the magnets were heat treated at 1200°C for 1 hour, cooled for 2 hours to 1150°C and held at this temperature for 5 hours, quenched, and then heated to 25 850°C and aged for 17 hours, cooled for 13 hours to 400°C, held at 400°C for 1 hour to 10 hours, and then quenched.

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Hysteresis loops were measured on these magnets and the results are set forth in TABLE I.

	TAI	BLE I	
05	EFFECT OF MAGNETIZE	ING FIELD ON TH	E REMANENCE*
	AND INTRINSIC CO	DERCIVE FORCE O	N ALLOY B
	Magnetizing		
	Field Strength	В	H _{ci}
	0e	<u> </u>	<u>0e</u>
10	3,000	4,000	5,400
	6,500	8,500	8,200
	10,600	9,700	9,300
	15,000	10,400	10,500
	~ 60,000	10,800	10,700
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^{*}Max value for saturation $B_s = 11,300G$ measured at 16 kOe.

For the above alloy so processed TABLE II shows a comparison between ball milled powder and jet milled powder on the magnet properties of transversed die pressed blocks.

TABLE II

COMPARATIVE EVALUATION OF BALL MILLED
AND JET MILLED POWDER ON ALLOY B AND COMMERCIAL ALLOY*

Br Hci

		B _r G_	^H ci <u>Oe</u>	Alloy
	Ball Milled Powder	9,900	8,600	В
30	Jet Milled Powder	10,600	10,100	В
30	Ball Milled Powder	9,600	9,000	Commercial
	Jet Milled Powder	10,250	9,300	Commercial

^{*}Commercial alloy with 26.6 Sm, 50.4 Co,

^{14.7} Fe, 5.9 Cu, 2.4 Zr

TABLE III shows that cold isostatic pressing produces higher remanence than the transverse die pressed blocks.

TABLE III

COMPARISON OF TRANSVERSE DIE BLOCKS AND
ISOSTATICALLY PRESSED SAMPLES

		$\mathtt{B}_{\mathtt{r}}$	^H ci	
		G	0e	Alloy
	Transverse Die Pressing	10,600	10,100	В
10	Isostatic Pressing	10,800+	10,700	В
	Transverse Pressing	10,250	9,300	Commercial
	Isostatic Pressing	10,550	7,900	Commercial

TABLE IV shows the effect of heat treatment on the magnetic properties of the tested magnets.

		TAB	LE IV				
		$\mathtt{B}_{\mathtt{r}}$	Hc	$^{ m H}$ ci	BH max		
	ALLOY	G	0e	0e	MG0e		
20	A	10,000	7,300	10,200	22.4		
	A	10,950	7,900	11,450	27.5		
	В	10,950	8,350	17,950	24.0		

An alloy of the composition 26.0 samarium, 49.0

cobalt, 3.9 copper, 19.2 iron, 2.5 zirconium, closely similar in composition to Alloy B, was jet milled and die pressed with the applied field in the same direction as the pressing direction. These magnets were solution treated over a temperature range of 1080 to 1180°C for five hours and aged at different temperatures as indicated in TABLES V to VII.

TABLE V

MAG	(TEV	C PROP	EF	RTI	ES O	T TI	HE A	LLOY	
AGED	ΑT	850°C	_	17	HRS	13	HRS	COOL	_
			١						_

	AGED A	400°C -	2 HRS	>	
	Solution				
05	Treat	•			
	Temperature	Br	HC	H _{ci}	BHmax
	°c	G	0 <u>e</u>	0e	MGOe
	1180	9,500	7,570	17,320	18.5
	1160	9,750	7,080	14,450	19.3
10	1120	9,670	5,800	11,450	15.3
		TABLE	VI		
		ric properti	30		
	AGED A	г 820 [°] С - 17	· · · · · · · · · · · · · · · · · · ·	>	
15		400 ^O C -	2 HRS		
	Solution				
	Treat				
	Temperature	^B r	Hc	Hci	BH
	°C	G	<u>0e</u>	<u>0e</u>	<u>MGOe</u>
20	1180	10,170	7,850	14,150	21
	1160	9,575	6,630	12,500	18
	1140	9,500	6,600	11,600	16.4
	1120	9,800	5,800	7,500	17.5
٥٢	•	m v D v E	17 T T		
25	MACNIE	TABLE TIC PROPERTII		r r o v	
		r 780°C - 17			
	11045 11.	_	2 HRS	>	
•	Solution				
30	Treat				
	Temperature	$^{\mathtt{B}}\mathtt{r}$	Нс	Hci	BH
	°c	<u>G</u>	0e	0e	MGOe_
	1180	10,000	4,100	5,000	16.2
	1160	9,650	3,630	4,250	13.2
35	1140	9,350	3,100	3,800	10.2

8,500

2,350

2,650

This same alloy composition was jet milled and die pressed with the applied field perpendicular to the pressing direction. These magnets were solution heat treated at 1180 or 1150°C and aged at 850°C. The magnetic properties obtained are shown in TABLE VIII.

TABLE VIII

MAGNETIC PROPERTIES OF TRANSVERSE PRESSED MAGNETS
SOLUTION TREATED AT TWO DIFFERENT TEMPERATURES
Solution

Tre	at
-----	----

Temperature OC	B _r	H C Oe	H _{ci} 	BH max MGOe
1180	10,600	8,100	14,200	24.3
1150	10,550	7,300	12,320	23.0

As may be seen from these specific examples, the desired combination of coercive force and residual magnetization may be obtained by continuous cooling after the aging treatment.

CLAIMS:

- 1. A permanent magnet of R₂Co₁₇ type crystal structure characterised in consisting of, in percent by weight, at least one rare earth element within the range of 24 to 28, cobalt within the range of 48 to 53, copper within the range of 2 to 4.9 to less than 5, iron within the range of 18 to 30 and zirconium within the range of 1.7 to 3.0.
- 2. A permanent magnet according to claim 1, characterised in that R is samarium.
- 3. A permanent magnet according to claims 1 and 2, characterised in that said magnet is aged and thereafter continuously cooled prior to quenching.

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EUROPEAN SEARCH REPORT

Y	GB - A - 2 089 3' * Page 2, lin line 14; fine 1	ne 9 - page 6, ig. 2,4,6,7 *	Relevant to claim	H 01 F 1/04
Y	* Page 2, line 14; for the second sec	ne 9 - page 6, ig. 2,4,6,7 * 140 (TOKUNAGA) Line 39 - column		
	line 14; fi US - A - 4 284 4 * Column 4, 1	ig. 2,4,6,7 * 140 (TOKUNAGA) Line 39 - column	1-3	
	* Column 4,	Line 39 - column	1-3	
	* Column 4,	Line 39 - column	1-3	
			ļ	
7.7	110 4 4 000 4		1 0	
Y	US - A - 4 322 2		1-3	
	* Column 5, l line 22; fi	ine 19 - column 6, lg. 1.2 *	,	
	,			
A	CH - A5 - 607 25	3 (KAWASAKI)	1,2	
	* Claim I; co			
	-			TECHNICAL FIELDS SEARCHED (Int. CI. 3)
				C 22 C 19/00
				C 22 C 38/00
				H 01 F 1/00
				H 01 F 41/00
	•			
I	The present search report has b	een drawn up for all claims		
	Place of search	Date of completion of the search		Examiner
	VIENNA	08-05-1984		PIRKER
X : parti Y : parti docu	CATEGORY OF CITED DOCL icularly relevant if taken alone icularly relevant if combined warment of the same category nological background written disclosure	after the	r principle undentent document filing date nt cited in the a nt cited for othe	rlying the invention , but published on, or oplication r reasons