

(19)



Europäisches Patentamt
European Patent Office
Office européen des brevets

(11) Publication number:

0 161 394
A1

(12)

EUROPEAN PATENT APPLICATION

(21) Application number: 85101590.9

(51) Int. Cl.⁴: **C 22 C 19/07, H 01 F 1/14**

(22) Date of filing: 26.11.81

(43) Date of publication of application: 21.11.85
Bulletin 85/47

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(84) Designated Contracting States: **DE FR GB IT NL**

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(60) Publication number of the earlier application in accordance with Art. 76 EPC: **0080521**

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(54) **Low magnetostriction amorphous metal alloys.**

(57) A magnetic alloy having a composition selected from the group consisting of $\text{Co}_{73}\text{Fe}_{4.5}\text{Mn}_{2.5}\text{B}_{20}$, $\text{Co}_{73}\text{Fe}_{2.5}\text{Mn}_{4.5}\text{B}_{20}$, $\text{Co}_{73}\text{Fe}_2\text{Mn}_5\text{B}_{20}$, $\text{Co}_{68}\text{Mn}_9\text{Ni}_5\text{B}_{20}$ and $\text{Co}_{68.4}\text{Mn}_{8.3}\text{Ni}_{3.3}\text{B}_{20}$.

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0161394

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Konto-Nr. 276 807 00 (BLZ 510 800 60)

European Patent Office
Erhardtstr. 27
8000 München 2

Datum Feb. 11, 1985

We/Wh

1 7000-1483

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10 Low Magnetostriction Amorphous Metal Alloys

Division from European Patent Application
No. 81 109 913.4

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1. Field of the Invention

This invention relates to amorphous metal alloys and, more particularly, to cobalt rich amorphous metal alloys that include certain transition metal and
20 metalloid elements.

1 2. Description of the Prior Art

There are three physical parameters which can inhibit the easy magnetization and demagnetization of magnetic materials: strong anisotropy, non-zero magnetostriction and, at high frequencies, low resistivity. Metallic glasses generally show resistivities greater than 100 micro ohm cm, whereas crystalline and polycrystalline magnetic metals generally show resistivities below 50 micro ohm cm. Also, because of their randomly disordered structures, metallic glasses are typically isotropic in their physical properties, including their magnetization. Because of these two characteristics, metallic glasses have an initial advantage over conventional magnetic metals. However, metallic glasses do not generally show zero magnetostriction. When zero magnetostriction glasses can be found they are generally good soft magnetic metals (R.C. O'Handley, B.A. Nesbitt, and L.I. Mendelsohn, IEEE Trans Mag-12, p. 942, 1976, U.S. Patents Nos. 4,038,073 and 4,150,981), because they satisfy the three approved criteria. For this reason, interest in zero magnetostriction glasses has been intense as indicated by the many publications on low magnetostriction metallic glasses (A.W. Simpson and W.G. Clements, IEEE Trans Mag-11, p. 1338, 1975; N. Tsuya, K.I. Arai, Y. Shiraga and T. Masumoto, Phys. Lett. A51, p. 121, 1975; H.A. Brooks, Jour. Appl. Phys. 47, p. 334, 1975; T. Egami, P.J. Flanders and C.D. Graham, Jr., Appl. Phys. Lett. 26, p. 128, 1975 and AIP Conf. Proc. No. 24, p. 697, 1975; R.C. Sherwood, E.M. Gyorgy, H.S. Chen, S.D. Ferris, G. Norman and H.J. Leamy, AIP Conf. Proc. No. 24, p. 745, 1975; H. Fujimori, K.I. Arai, H. Shiraga, M. Yamada, T. Masumoto and N. Tsuya, Japan, Jour. Appl. Phys. 15, p. 705, 1976; L. Kraus and J. Schneider, phys. stat. sol. a39, p. K161, 1977; R.C. O'Handley in Amorphous Magnetism, edited by R. Levy and R. Hasegawa (Plenum Press, New York 1977), p. 379; R.C. O'Handley, Solid State Communications 21, p. 1119, 1977;

1 R.C. O'Handley, Solid State Communications 22, p. 458,
1977; R.C. O'Handley, Phys. Rev. 18, p. 930, 1978; H.S.
Chen, E.M. Gyorgy, H.J. Leamy and R.C. Sherwood, U.S.
Patent No. 4,056,411, Nov. 1, 1977).

5 The existence of a zero in the magnetostric-
tion of Co-Mn-B glasses has been observed by H. Hilt-
zinger of Vacuumschmelze A.G., Hanau, Germany.

Reference to Co-rich glasses containing 6 atom
percent of Cr is made by N. Heiman, R.D. Hempstead and
10 N. Kazama in Journal of Applied Physics, Vol. 49, p.
5663, 1978. Their interest was in improving the
corrosion resistance of Co-B thin films. No reference
to magnetostriction is made in that article.

Saturation moments and Curie temperatures of
15 $\text{Co}_{80-x}\text{T}_x\text{P}_{10}\text{B}_{10}$ glasses (T = Mn, Cr, or V) were recently
reported by T. Mizoguchi in the Supplement to the Scien-
tific Reports of RITU (Research Institutes of Tonoku
University), A June 1978, p. 117. No reference to
their magnetostrictive properties was reported.

20 In Journal of Applied Physics, Vol. 50,
p. 7597, 1979, S. Ohnuma and T. Masumoto outline
their studies of magnetization and magnetostriction
in Co-Fe-B-Si glasses with light transition metal (Mn,
Cr, V, W, Ta, Mo and Nb) substitutions. They show that
25 the coercivity decreases and the effective permeability
increases in the composition range near zero magneto-
striction.

New applications requiring improved soft zero-
magnetic materials that are easily fabricated and have
30 excellent stability have necessitated efforts to develop
further specific compositions.

SUMMARY OF THE INVENTION

The present invention provides low magneto-
striction and zero magnetostriction glassy alloys that
35 are easy to fabricate and thermally stable. The alloys
have a composition selected from the group consisting of
 $\text{Co}_{73}\text{Fe}_{4.5}\text{Mn}_{2.5}\text{B}_{20}$, $\text{Co}_{73}\text{Fe}_{2.5}\text{Mn}_{4.5}\text{B}_{20}$, $\text{Co}_{73}\text{Fe}_2\text{Mn}_5\text{B}_{20}$, $\text{Co}_{66}\text{Mn}_9\text{Ni}_5\text{B}_{20}$ and $\text{Co}_{68.4}\text{Mn}_{8.3}\text{Ni}_{3.3}\text{B}_{20}$.

1 The purity of the above composition is that found
in normal commercial practice.

BRIEF DESCRIPTION OF THE DRAWINGS

5 The invention will be more fully understood and
further advantages will become apparent when reference is
made to the following detailed description of the preferred
embodiments of the invention and the accompanying drawing,
which shows the cobalt-rich corners of triangular diagrams
for compositions defined by the formula $(\text{Co}_{1-x-y}\text{Fe}_x\text{T}_y)_{80}\text{B}_{20}$,
10 where T is Mn.

 The amorphous alloys of the invention can be
formed by cooling a melt of the composition at a rate
of at least about 10^5 °C/sec. A variety of techniques
are available, as is now well-known in the art, for
15 fabricating splat-quenched foils and rapid-quenched
continuous ribbons, wire, sheet, etc. Typically, a
particular composition is selected, powders of the
requisite elements (or of materials that decompose
to form the elements, such as nickel-borides, etc.)
20 in the desired proportions are melted and homogenized,
and the molten alloy is rapidly quenched either on a
chill surface, such as a rotating cooled cylinder, or
in a suitable fluid medium, such as a chilled brine
solution. The amorphous alloys may be formed in air.
25 However, superior mechanical properties are achieved
by forming these amorphous alloys in a partial vacuum
with absolute pressure less than about 5.5 cm of Hg,
and preferably about 100 μm to 1 cm of Hg, as disclosed
in U.S. Patent No. 4,154,283 to Ray et al.

30 The amorphous metal alloys are at least 50
percent amorphous, and preferably at least 80 percent
amorphous, as measured by X-ray diffraction. However, a
substantial degree of amorphousness approaching 100 per-
cent amorphous is obtained by forming these amorphous
35 metal alloys in a partial vacuum. Ductility is thereby
improved, and such alloys possessing a substantial
degree of amorphousness are accordingly preferred.

 Ribbons of these alloys find use in soft

1 magnetic applications and in applications requiring low
magnetostriction, high thermal stability (e.g., stable
up to about 100°C) and excellent fabricability.

The following example is presented to provide
5 a more complete understanding of the invention. The
specific techniques, conditions, materials, proportions
and reported data set forth to illustrate the principles
and practice of the invention are exemplary and should
not be construed as limiting the scope of the invention.

10 EXAMPLE

An alloy melt of known composition was rapidly
quenched to form non-crystalline ribbons, presumably of
the same composition as the melt. The ribbons, typi-
cally 40 micrometers (μm) by 2 mm in cross section, were
15 cut into squares for vibration-sample magnetometer
measurements of specific magnetization $\sigma(4.2\text{K}, 9 \text{ KOe})$
and $\sigma(T, 9 \text{ KOe})$ with $295 \text{ K} < T < T_x$, the crystallization
temperature. Curie temperatures were obtained from the
inflection points in the $\sigma(T, 9 \text{ KOe})$ curves.

20 The magnetostriction measurements were made in
fields up to 4 KOe with metal foil strain gauges (as
reported in more detail by R.C. O'Handley in Solid State
Communications, Vol. 22, p. 485, 1977). The accuracy of
these measurements is considered to be within 10 percent
25 of full strain and their strain sensitivity is on the
order of 10^{-7} .

Co-rich glass compositions with positive and
negative magnetostriction can be added linearly to give
zero magnetostriction. For example, λ_s for $\text{Co}_{70}\text{Fe}_{10}\text{B}_{20}$
30 and $\text{Co}_{80}\text{B}_{20}$ glasses are $+4$ and -4×10^{-6} , respectively.
A 50-50 percent mixture of these glasses gives
 $\text{Co}_{75}\text{Fe}_{5}\text{B}_{20}$ which does in fact show $\lambda_s = 0$ (O'Handley et
al., IEEE Trans Mag-12, p. 942, 1976). Similarly, for
 $\text{Co}_{40}\text{Ni}_{40}\text{B}_{20}$, $\lambda_s = -7 \times 10^{-6}$ while for
35 $\text{Fe}_{80}\text{B}_{20}$, $\lambda_s = 32 \times 10^{-6}$. A linear mixture having $\lambda = 0$
would be $0.18 \times (\text{Fe}_{80}\text{B}_{20}) + 0.82 \times (\text{Co}_{40}\text{Ni}_{40}\text{B}_{20}) =$
 $\text{Co}_{33}\text{Ni}_{33}\text{Fe}_{14}\text{B}_{20}$ which is very close to the observed
 $\lambda_s = 0$ composition, $\text{Co}_{33.5}\text{Ni}_{33.5}\text{Fe}_{13}\text{B}_{20}$.

1 The rule of linear combination of opposing magnetostrictions
(LCOM) has been applied to develop additional zero magneto-
striction glasses. Table I lists several such glasses and
the drawing shows where they fall in the Co-rich corner
5 of a triangular composition diagram. The lines connecting
these newly developed $\lambda_s = 0$ compositions closely follow the
observations of Ohnuma and Masumoto (cited above) for
(Co Fe X)₇₈B₁₄Si₈ glasses (with X = Mn, Cr, V) despite the
different metalloids used in the two cases.

10

TABLE I

Some Near-zero Magnetostriction
Cobalt-rich Glasses Developed by the LCOM Method

15	Co ₇₃ Fe _{4.5} Mn _{2.5} B ₂₀	Co ₇₃ Fe ₂ Mn ₅ B ₂₀
	Co ₇₃ Fe _{2.5} Mn _{4.5} B ₂₀	

The magnetostriction of Co-rich glasses is
small because of the near-cancellation of two indepen-
dent mechanisms for the magnetostriction, a positive
20 two-ion interaction and a negative single-TM-ion term
(O'Handley, Phys. Rev. B 18, p. 930, 1978). As a re-
sult, the TM makeup for $\lambda_s = 0$ is nearly independent of
TM/M ratio.

An improvement on this approximation can be realized by
25 taking into account the fact that the strength of the
negative single-ion term varies linearly with the
concentration of magnetic ions.

The
two-ion term should vary as the number of TM pairs at
short range. However, observed trends in Co_{100-x}B_x
30 glasses (K. Narita, J. Yamasaki, and H. Fukunaga, Jour.
Appl. Phys. Vol. 50, p. 7591, 1979 and J. Aboaf and E.
Klokholm, ICM Munich Sept. 1979 to appear in Jour.

Magnetism and Magnetic Materials), are best described by
assuming the number of nearest neighbor TM pairs to be
35 independent of x. This implies that the nearest-
neighbor coordination of cobalt atoms by cobalt atoms

1 does not vary strongly with x . Thus the compositional
dependence of magnetostriction in Co-rich glasses is
well described at room temperature by:

$$\lambda_s \approx + 6.8 \times 10^{-6} - 10.2 \times 10^{-6} \times (100-x)/80 \text{ where the}$$

5 first term is the observed two-ion component
of magnetostriction (independent of composition x) and
the second is the single-ion component of magnetostric-
tion (which varies linearly with the TM concentration).

Thus the magnetostriction becomes less negative as
10 metalloid content increases, the change in λ being
 $+0.13 \times 10^{-6}$ per atom percent more metalloid.

Alternatively, the zero magnetostriction composition is
shifted to glasses richer in iron as $100-x$ increases,
the shift being approximately $+0.23$ percent Fe per
15 1 percent decrease in x .

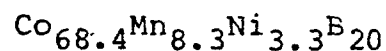
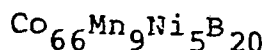
As a result, the Co-Fe-T ratios ($T = \text{Mn}$)
for $\lambda_s = 0$ in the drawing hold approximately for other
TM/M ratios in the glass-forming range $12 < x < 28$ atom
percent. A first order correction shifts the $\lambda_s = 0$
20 lines toward Fe by approximately 1 percent for every
4 percent decrease in x .

Metalloid type has little effect on the magni-
tude or sign of magnetostriction in Co-rich glasses
(O'Handley in Amorphous Magnetism eds. R. Levy and R.
25 Hasegawa, Plenum Press 1977, p. 379).

Table II sets forth some
typical near-zero magnetostriction compositions.

TABLE II

New Co-Ni Base Glassy Alloys or
30 Near-zero Magnetostriction Developed by LCOM Method.



1 Claim

A magnetic alloy having a composition selected from the group consisting of $\text{Co}_{73}\text{Fe}_{4.5}\text{Mn}_{2.5}\text{B}_{20}$, $\text{Co}_{73}\text{Fe}_{2.5}\text{Mn}_{4.5}\text{B}_{20}$, $\text{Co}_{73}\text{Fe}_2\text{Mn}_5\text{B}_{20}$, $\text{Co}_{66}\text{Mn}_9\text{Ni}_5\text{B}_{20}$ and $\text{Co}_{68.4}\text{Mn}_{8.3}\text{Ni}_{3.3}\text{B}_{20}$.

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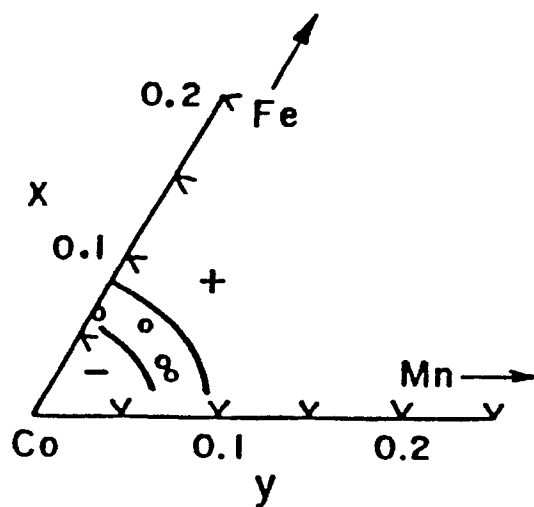
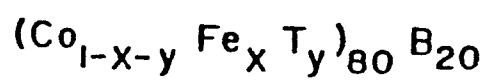
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DOCUMENTS CONSIDERED TO BE RELEVANT			
Category	Citation of document with indication, where appropriate, of relevant passages	Relevant to claim	CLASSIFICATION OF THE APPLICATION (Int. Cl.4)
D,X	<u>US - A - 4 056 411 (CHEN)</u> * Abstract; fig. I * --	1	C 22 C 19/07 H 01 F 1/14
A	<u>DE - A1 - 2 553 003 (ALLIED CHEMICAL CORP.)</u> * Page 2, lines 10-21 * --	1	
A	<u>EP - A1 - 0 021 101 (VACUUM-SCHMELZE)</u> * Abstract * ----	1	
The present search report has been drawn up for all claims			TECHNICAL FIELDS SEARCHED (Int. Cl.4) C 22 C H 01 F
Place of search VIENNA		Date of completion of the search 23-07-1985	Examiner ONDER
CATEGORY OF CITED DOCUMENTS X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background O : non-written disclosure P : intermediate document T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons & : member of the same patent family, corresponding document			