

(54) Coated article.

(57) An article for use in high temperature corrosive environments is protected by successive coatings. The coatings comprise

i) a first coating of one or more metals or alloys,

ii) an optional second coating of a mixture of one or more ceramics and one or more metals or alloys,

iii) a third coating of one or more ceramics, and

iv) a fourth coating of one or more precious metals or alloys thereof, the fourth coating being substantially non-porous.

This invention relates to an article for use at high temperatures and in corrosive environments and in particular to a protected article. This invention also relates to a method of coating metallic substrates.

The glass industry has a particular need for materials which have both good stability at high temperatures and good corrosion resistance for the manufacture of apparatus that contacts molten glass. Examples of such apparatus are crucibles, stirrers, crown spinners, thermocouple sheaths, etc. The difficulties encountered with apparatus which either contacts molten glass or the atmosphere above the molten glass are the high temperatures, of the order of 1000°-1600°C, the viscosity and abrasiveness of the molten glass and the reactivity of the glass itself. Molten glass will attack, for example crucibles or containers in which it is held, and this attack is most pronounced at the glass line where the glass and air interface and the surface of the glass contacts the

- 10 crucible or container wall. With time the glass attacks the wall of the apparatus, eg crucible or container and forms a hollow at the glass line as shown in Figure 1 of the accompanying drawing which depicts a portion of a wall 1 of a crucible containing molten glass 2 with air 3 above the glass. At the glass 2-air 3 interface denoted by 4 the wall of the crucible has eroded 5 due to the corrosive effect of the glass. Furthermore as a result of the reactivity of the glass small fragments of the apparatus in contact with glass may break off and contaminate
- 15 the glass. This is a particular disadvantage when the glass is being used for high quality products such as television tubes which should have no visible defects. The chemical attack by the molten glass may also result in metals or compounds being leached into the glass which results in undesirable discolouration.

Apparatus for use in the glass industry is often fabricated from transition or refractory metals or alloys such as nickel alloys or metals or alloys of molybdenum, tantalum, etc. in order to provide apparatus with a reasonable time period in service. Platinum group metals or alloys have also been used as the material for such apparatus since these metals and alloys have good corrosion resistance to molten glass and good thermal stability. However, the one major disadvantage with using platinum group metals is the high cost of the material itself. To overcome the disadvantage of having apparatus fabricated solely from platinum group metals and alloys apparatus is made with a metallic substrate that is coated or clad therewith, in particular with platinum.

- In order to provide effective protection the coating or cladding must be of sufficient thickness to protect the substrate for a service life of eg, 3-6 months. These coatings or claddings provide some protection and enable the service life to be increased. The coating or cladding of platinum however does not prevent the major problem of diffusion of oxidants through to the substrate which results in corrosion of the substrate.
- Gas turbines blades operate at high temperatures between 600°C and 1100°C and in corrosive environments. A coating of ceramic applied to the blades provides some protection. However, although the ceramic coating does protect the blades from the high temperatures the ceramic coating suffers considerable corrosion. Another disadvantage is the tendency for spalling to occur with the ceramic coating lifting from the blades and becoming detached.

An object of the present invention is to provide an article with improved corrosion resistance and heat stability which overcomes at least some of the disadvantages of the prior art, especially the problem of diffusion of oxidants through to the substrate.

Accordingly the invention provides an article for use at high temperatures and in corrosive environments comprising a metallic substrate on which is deposited successively,

i) a first coating of one or more metals or alloys,

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- ii) an optional second coating of a mixture of one or more ceramics and one or more metals or alloys,
 - iii) a third coating of one or more ceramics, and

iv) a fourth coating of one or more precious metals or alloys thereof, the fourth coating being substantially non-porous.

The invention further provides a method of making an article comprising applying, successively, to a metallic substrate a first coating of one or more metals or alloys, an optional second coating of a mixture of one or more ceramics and one or more metals or alloys, a third coating of one or more ceramics and a fourth coating of one or more precious metals or alloys thereof, and making the fourth coating substantially non-porous.

References following to a metal or metals include an alloy or alloys of the metal or metals, unless otherwise indicated.

- ⁵⁰ Preferably the metallic substrate is a metal that has some corrosion resistance and/or thermal stability at high temperatures such as refractory metals, eg nickel, molybdenum, tantalum, niobium or tungsten or an alloy of any of these. Other suitable alloys are high melting point intermetallics such as nickel aluminides, cobalt aluminides, tantalum aluminides, platinum group metal intermetallics etc. The metallic substrate may have a surface oxide layer.
- ⁵⁵ The metal of the first coating is preferably compatible with the ceramic of the second coating such that once the substrate is coated the ceramic is not degraded by the metal reacting with the ceramic. The metal is preferably selected from metals that have one or more of the following characteristics; i) good corrosion resistance and good thermal stability, ii) good ductility and iii) thermal coefficient of expansion intermediate the thermal

coefficient of expansion of the metallic substrate and the ceramic of the second coating. Preferably the metal is such that the coating is formed with release of energy, preferably by formation of an intermetallic compound. Suitable metals are nickel-aluminium, nickel-chromium, palladium-aluminium, hafnium-aluminium, titanium, tantalum and palladium. Most preferred is an alloy of nickel 80wt% aluminium 20wt%.

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The first coating may be deposited onto the substrate by known methods of application such as thermal spraying and in particular flame or vacuum- or air-plasma spraying.

The second coating comprises a mixture of one or more ceramics and one or more metals. Preferably the ceramic consists essentially of at least one of alumina, magnesia, hafnia, thoria and zirconia. Most preferably the ceramic consists essentially of alumina. Optionally the ceramic may include a minor amount of one or more

of yttria, silica, titania and calcia. Most preferably the ceramic is of high purity of the order of 99% purity. The 10 metal is preferably the same metal as that of the first coating. Preferably the second coating has a composition in which the proportion of metal to ceramic decreases in gradations from that at the interface adjacent the first coating to that at the interface adjacent the third coating. The second coating may be formed by thermal spraying, for example by vacuum- or air-plasma spraying and in particular the coating may be formed by the deposition of one or more layers. A plurality of layers can be employed to form a step-wise gradation of the metal

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to ceramic ratio as mentioned above.

The third coating of ceramic preferably consists essentially of alumina. Suitable other materials for the third coating may be magnesia, hafnia, thoria or zirconia or a mixture thereof. Optionally the ceramic may include a minor amount of one or more of yttria, silica, titania and calcia. This coating may be deposited by thermal spraying and in particular by flame or vacuum- or air-plasma spraying one or more layers.

The fourth coating is substantially non-porous and is chosen from one or more of the precious metals. Preferably the coating is of platinum. An alloy of platinum may also be used, eg platinum with 5% gold, or zirconia grain stabilised (ZGS) platinum. Other platinum group metals may also be employed. The coating may be deposited for example by flame or vacuum- or air-plasma spraying, as a powder, or by plating. Preferably the coating

- is applied by thermal spraying and in particular by flame or vacuum- or air-plasma spraying. The coating can 25 be made non-porous by after treatment. The after treatment may comprise shot peening, melt glazing, hot isostatic pressing (hip) or cold isostatic pressing (cip). The thickness of the fourth coating is typically, between 75 and 200µm; however, there is no upper limit as to how thick the fourth coating may be. Preferably, the thickness of the fourth coating is at least 90µm.
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Usually the article consists of the metallic substrate and the four specified coatings. However, in some cases, only the first, third and fourth coatings need be employed.

In a particular embodiment, the article has an additional fifth coating between the third and fourth coatings, the fifth coating being a mixture of one or more ceramics and one or more precious metals or alloys thereof. Preferably the ceramic consists essentially of alumina. Other suitable ceramics may be magnesia, hafnia, thoria

- or zirconia or a mixture thereof. Optionally the ceramic may include a minor amount of one or more of yttria, .35 titania, silica, and calcia. Most preferably the ceramic is of high purity, of the order of 99% purity. Preferably the fifth coating has a composition in which the proportion of metal to ceramic increases in gradations from that at the interface with the third coating to that at the interface with the fourth coating. The fifth coating may be formed by thermal spraying, for example by flame or vacuum- or air-plasma spraying and in particular the coat-
- 40 ing may be formed by the deposition of one or more layers. A plurality of layers can be employed to form a step-wise gradation of the metal to ceramic ratio as mentioned above.

Figure 2 is a cross-section of an article according to the present invention.

An embodiment of the present invention is shown in Figure 2. A metallic substrate 10 has a first coating 11 of a metal or alloy, a second coating 12 of a mixture of one or more metals and alloys at least one ceramic, 45 a third coating 13 of at least one ceramic and a fourth non-porous coating 14 of precious metal or alloy. In the second coating 12 the proportion of metal or alloy to ceramic decreases from this interface with the first coating 11 to the interface with the third coating 13 as schematically depicted with '-' for metal or alloy of the first and second coatings and '+' for the ceramic of the second and third coatings.

The invention will now be described by way of example only.

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COMPARATIVE EXAMPLE A

Nicrobell A is the trade name of a Cr/Al/Ti/Y2O3/Fe alloy, produced by Nicrobell Pty Ltd. Nicrobell X is the trade name of a Cr/Si/Mg/Ni alloy produced by Nicrobell Pty Ltd.

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A number of test samples of refractory alloys in the form of tubes were coated with a platinum coating of various thicknesses. The platinum was deposited by means of a fused salt plating bath. One of the samples, Nicrobell A with a platinum coating of 60 µm had been subjected to a preoxidation treatment so that the Nicrobell A substrate had a thin oxide layer on to which the platinum was applied. The samples were tested by placing

the sample in contact with molten bottle glass at a temperature of 1200°C for a period of time. Each sample was placed with one end in an alumina crucible and bottle glass powder added such that on heating the crucible to a temperature of 1200°C a portion of the sample is in contact with molten glass. At the end of the test the samples were visually examined to see the corrosive effect on the sample by the glass. The results for a 24 hour test are given in Table 1 below. The alloys Nicrobell X and Nicrobell A consist of chromium 20%, aluminium 4.5%, titanium 0.5%, Y_2O_3 0.5%, remainder iron, and chromium 14.0%, silicon 1.4%, magnesium 0.2% and

Table 1

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15	Substrate	Pt Coating Thickness μm	
	Nicrobell A	46	Ni diffused through Pt coating.
	Nicrobell A	60	Pt coating lifted off by diffusion.
20	Nicrobell X	72	Fe diffused through Pt coating.
			Al reaction with Pt.

remainder nickel, the percentages being by weight.

The results of the tests given in Table 1 above show that a single coating of platinum between 46μm and 72μm onto a metallic substrate does not prevent corrosion by molten glass.

Examples 1-12

30 A number of tubes were fabricated from a commercial nickel-based alloy, nickel 75wt%, chromium 25wt%. A first coating of nickel was applied thereto followed by a second coating of nickel and alumina formed by three layers having the Ni:Al₂O₃ proportions 1:1, 1:3 and 1:9 respectively. The third coating of alumina was applied as one or more layers. The first, second and third coatings were applied using vacuum plasma spraying. Examples 1 to 4 were provided with a conductive surface by sputter deposition of 0.2μm films of platinum in prepa-

ration for a fourth coating of platinum deposited by electro-deposition from a bath of composition disclosed in Example of EP 0358375 A with 5-5.5g of platinum per litre. Examples 1, 2 and 3 had an additional top coating of 0.2µm of gold deposited by sputtering. The fourth coating of Examples 5 to 8 was applied by a fused salt process. Examples 9-12 had the fourth coating deposited by flame spraying. The Examples were tested for 300 hours by contacting with molten glass as described in Comparative Example A and the results are given in Tables 2, 3 and 4. AGL, GL and BGL denote 'Above Glass Line', 'Glass Line' and 'Below Glass Line'.

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10			L.	substrate oured. strate not tached.	olackened, n coating. iened on	diffusion ut lifted. intact – king.
15			Coating Condition After Test	Coating peeled back, substrate still intact, Pt coloured. Coating eroded, substrate not corroded. Coating generally detached. Glass colour - green.	Coating blistered, blackened, still intact. Some erosion through coating. Substrate protected. Still intact, loosened on cooling. Glass colour - green.	(Ni/Cr Pt) intact b but layer protected. it some crac. our - green
20			ing Conditi	Coating pe still inta Coating e corroded. Coating ge Glass colo	Coating blis still intact. Some erosion Substrate pro Still intact cooling. Glass colour -	Darkened through J Attacked substrate Intact bu Glass col
25				AGL GL BGL	AGL GL BGL	AGL GL BGL
) Hours	ting ss +Au 0.2µm	Yes	Yes	Yes
30	TABLE 2	IST - 300	4th Coating Pt thickness µm +Au (18.0	18.1	17.5
35	TABI	DURATION OF TEST - 300 Hours	3rd Coatin g No of Al ₂ 0 ₃ Layers	5	-	4
		Ц	8 1:9		н	-
40			2nd Coating Ni:Al2 ⁰³ 1:1 1:3 ³ 1		1	7
			2nc Nj 1:1		-	T
45			G)	25%	\$ <u>5</u>	۲ ۲
50			Substrate	Ni75%/Cr25%	Ni75%/Cr25%	Ni 75%/Cr25%
			Example			
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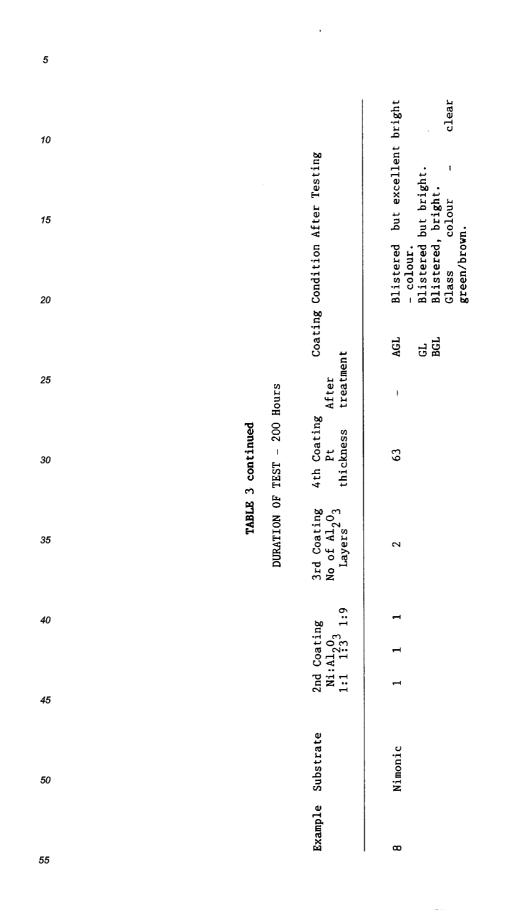
10					ion t coating ntact.
15				Coating Condition After Test	Darkened (Ni/Cr diffusion through Pt), intact but lifted. Substrate protected - coating blistered severely. Pt protected sample. Intact. Glass colour - green.
20				ng Condit	Darkened through lifted. Substra blistere Pt prote Glass co
25					AGL GL BGL
		ued	300 Hours	oating t ness +Au 0.2µm	No
30	:	TABLE 2 continued	e TEST -	ıg 4th Coatin ⁾ 3 Pt µш +Au	11.0
35		TABLE	DURATION OF TEST - 300 Hours	3rd Coating 4th Coating No of Al ₂ 0 ₃ Pt Layers thickness µm +Au C	~
40				2nd Coating Ni:Al ₂ 03 1:1 1:3 ³ 1:9	
45				2nd Ni: 1:1	
50				Substrate	Ni 75%/Cr25%
00				Example Su	A N
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10			ing	but black	nd intact, akdovn. but very	erent. /brown. ght Pt aring. - general /brown.
15			Coating Condition After Testing t	Generally intact crystalline surface. Slight blistering. General protection.	Glass colour - green. Generally very good and intact, but black crystalline. Slight blistering/breakdown. Some Ni diffusion but very	<pre>protective, very adherent. Glass colour - green/brown. Blistered mostly bright Pt coloured. Intact, slight blistering. Slight spallation - gen protection. Glass colour - green/brown.</pre>
20			Coating Co t	AGL Gel CL SL BGL Ger	Gla AGL Ger but GL Sli BGL Som	Pro Gla Gla Gla Bli GL Int BGL Sli Fro Gla
25		lours	After treatmen			Shot A peened G B
30	TABLE 3	EST - 200 Hours	4th Coating Pt thickness	51	54	59
35	TAI	DURATION OF TEST	3rd Coating No of Al2 ⁰ 3 Layers	5	г	4
40			ing 3 1:9		7	г
			2nd Coating Ni:Al ₂ 0 ₃ 1:1 1:3 ³ 1:		1	F
45			21 1:	2	~	-
50			Example Substrate	Ni 75%/Cr25%	Ni 75%/Cr25%	Ni 75%/Cr25%
			Example	Ŋ	Q	~
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10				ocked off.	en.	ly moved. rown.
15			ıfter Testing	ce test. but OK. out could be Ly intact. pale green.	louration. mid/dark green.	Louration. : significant BGL. but can be rei dark green/b
20			Coating Condition After Testing	AGL Mainly as before test. GL Slight erosion, but OK. BGL Glass adheres but could be ocked off. Coating generally intact. Glass colour - pale green.	Significant colouration. Slight erosion. Glass adheres. Glass colour - mid/dark	Significant colouration. Visible but not significantly different from BGL. Glass adheres but can be removed. Glass colour - dark green/brown.
25		S	Coati	AGL M GL S BGL G C C	AGL S GL S BGL G G	AGL S GL V d BGL G G G
30	TABLE 4	TEST - 402 Hours	4th Coating Pt thickness µm	112	88	115
35	TA	DURATION OF TEST	3rd Coating No of Al ₂ 0 ₃ Layers	8	1	4
40			ng 31:9	1	H	-
			2nd Coating Ni.Al2 ⁰³ 1:9 1:1 1:3 ⁰³ 1:9	1	1	
45				'Cr25% 1	/Cr25% 1	Ni75%/Cr25% 1
50			Substrate	Ni75%/Cr25%	Ni75%/Cr25%	Ni 75%/
			Example	б	10	11
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10 15 20			Coating Condition After Testing	Blackened colouration and blistering. Slight blistering but generally as BGL. Glass adheres, coating fragments on trying to remove glass. Glass colour - dark green/brown.
25		ILS		AGL GL GL BGL BGL GL G GL G GL G GL G GL G GL G GL G
30	TABLE 4 continued	DURATION OF TEST - 402 Hours	4th Coating Pt thickness µm	120
35	TABLE 4	DURATION OF	3rd Coating No of Al ₂ 0 ₃ Layers	2
40			2nd Coating Ni:Al2 ⁰³ 1:9 1:1 1:3 ³ 1:9	1 1
45				Ni75%/Cr25%
50			e Substrate	Nİ 75%
			Example	12
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Conclusion

Examples 1-12 demonstrate that the four-layer coating system is capable of protecting base metal substrate. However, very thin fourth coatings were largely ineffective below the glass line.

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Comparative Examples B and C

Example 13

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- Example 13 comprised a closed end tube of Nicrobell A with a first coating of Ni 96wt% Al 4wt%, a second coating of nickel and alumina formed by three layers having Ni:Al₂O₃ proportions of 1:1, 1:3 and 1:9, a third coating of alumina and a fourth coating of platinum of thickness 6.1µm. Comparative Example B was a tube of Nicrobell A with a first coating of Ni 96wt% Al 4wt%, a second coating of alumina graduated from low to high density and a third coating of platinum of thickness 4.8µm. For Comparative Example C a Nicrobell A closed
- end tube was coated with a first coating of alumina graduated from low to high density and a second coating of platinum, thickness 5.5µm. The coatings of Ni 96wt% AI 4wt%, alumina and nickel and alumina were applied by vacuum plasma spraying. After the surface was rendered conductive by sputter-deposition of a sub-micron film of platinum, the fourth coatings of platinum were deposited by electro-deposition from a bath of composition as described in Example 1 of EP 0358375 A with 5-5.5g of platinum per litre. These examples were tested for
- 300 hours by contacting the tubes with molten glass as described above in Comparative Example A and the results are given in Table 5 below. The Nicrobell A alloy has the composition chromium 14wt%, silicon 1.4wt%, magnesium 0.25wt% and remainder nickel.

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10 15			Coating Condition After Testing	Coating flaked off on cooling. Substantial glass line erosion of ceramic coating. 20% of surface covered by an adherent alumina layer.	Coating flaked off on cooling. Substantial glass line erosion of ceramic coating.	Coating flaked off on cooling. Substantial glass line erosion of ceramic coating. 90% of surface covered by an adherent alumina layer.
20			Coating C	AGL Coating cooling. GL Substant erosion BGL 20% of s adherent	AGL Coating cooling. GL Substant erosion	AGL Coat cool GL Subs eros BGL 90% adhe
25		- 300 Hours	4th Coating	1	ı	Pt 6.1µm
30	TABLE 5	DURATION OF TEST -	3rd Coating	Pt 4.8µm	I	A1203
35		DURAT	2nd Coating	Al2 ⁰ 3 graduated low to high density	d Pt 5.5µm	3 layers of Ni & Al ₂ O ₃ proportions 1:1 1:3 1:9
40			1st Coating 2	Ni96% A14%	Al2 ⁰³ graduated low to high density	Ni 96% Al 4%
<i>4</i> 5 50			Substrate	Nicrobell A	Nicrobell A	Nicrobell A
			Example	<u>م</u>	U	13

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The results given in Table 5 above indicate that to provide articles which can be used at high temperatures and in corrosive environments four different specific coatings are required. While in each case the Pt layer was too thin to be effective, and flaked off, the third layer was significantly more intact in the four coating Example as compared with the three coating Comparative Example.

Examples 14-21

Examples 14 to 21 employed closed end tubes of Nicrobell A or Nicrobell C as substrates. Nicrobell C is the trade name of an alloy of composition chromium 23.9wt%, silicon 1.75wt%, magnesium 0.15wt%, niobium 0.49wt% and remainder nickel, produced by Nicrobell Pty Ltd. The composition of the first coating for these examples was nickel 80wt%, aluminium 20wt%. The second coating of Ni 80wt% Al 20wt% and alumina was deposited as three layers with various proportions of Ni 80wt%, Al 20wt%: alumina. Details are given in Table 6 below. Between 4 and 30 layers of alumina were deposited to form the third coating. The first, second and third coatings were deposited by vacuum plasma spraying. The fourth layer of platinum was formed by flame-spraying followed by an after treatment of shot-peening to ensure the fourth coating was substantially non-por-

ous.

The examples were tested by contacting them with molten glass for 300 hours as described above. The results are given in Table 6 below.

		4th Coating Coating Condition After Testing Pt thickness	AGL Good. GL Glass adheres - no other attack. BGL Slight diffusion generally 0K. Glass colour - pale green.	AGL Good. GL Glass adheres - no other attack. BGL Patchy diffusion generally OK. Glass colour - pale green.	Not Tested.
	Hours	4th Coating Pt thickness	Ē	86	120
TABLE 6	DURATION OF TEST - 300 Hours	3rd Coating No of Al2 ⁰ 3 Layers µm	4	12	30
	DURAT	2nd Coating Ni:Al2 ⁰ 3 1:1 1;3 ³ 1:9		1	1
		2nd Coating Ni:Al ₂ 03 1:1 1;3 1	-	1	
		Substrate	Nicrobell A	Nicrobell A	Nicrobell A
		Example	14	15	16

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			1	υ υ		
10			ter	oliation, it otherwi (Patchy) good.	otherwise	, but but in (Patchy) ery good. pale green
15			Coating Condition After Testing	Very slight exfoliation, generally good. Glass aderes, but otherwise OK. Slight diffusion (Patchy) generally very good.	Good. Glass adheres, otherwise OK. Some diffusion.	Some exfoliation, but generally good. Glass adheres, but otherwise OK. Slight diffusion (Patchy) but generally very good. Glass colour - pale green.
20			tting Col Testing	Very gene Glas OK. Slig gene		
			Соа	AGL GL BGL	AGL GL BGL	AGL GL BGL
25		Hours	4th Coating Pt thickness μm	123	91	117
30	TABLE 6 continued	DURATION OF TEST - 300 Hours	3rd Coating No of Al ₂ O ₃ Layers	12	4	12
35	TAF	LATION				
		DUF	ing 3 1:9	1	Υ	ო
40			2nd Coating Ni:Al ₂ 0 ₃ 1:1 1:3 ³ 1	e	I	ì
			2nd Ni:	I	I	I
45						
40			ate	ell A	ell A	ell A
			Substrate	Nicrobell	Nicrobell	Nicrobell
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			Example	17	18	19
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10 15			Coating Condition After Testing	Completely intact, no noticeable change. Boundary line visible, but looks OK. Few areas of diffusion (green) but generally very good. Glass colour - pale green.	Slight diffusion but OK except where cracked open during cooling. No significant attack. Glass adheres - some areas of diffusion, especially bottom. Glass colour - pale green.
20			Co	AGL GL BGL	AGL GL BGL
25	-	- 300 Hours	4th Coating Pt thickness µm	102	104
30	TABLE 6 continued	DURATION OF TEST - 300	3rd Coating No of Al ₂ 0 ₃ Layers	4	12
35	H	RATI	_		
40		DU	2nd Coating Ni:Al ₂ 03 1:1 1:3 ³ 1:9	1 1 1	ຕ ເ ເ
45			Substrate	Nicrobell C	Nicrobell C
50			Example	20	21
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Example 16 was not tested as cracks appeared prior to platinum coating. This was probably due to the thickness of the third coating which was formed from thirty layers of alumina. The results obtained for Examples 14, 15 and 20 with a second coating having the amount of metal decreasing from the interface with the first coating to the interface with the third coating are better than for Examples 17, 18, 19 and 21 where the proportion of metal to ceramic was the same through the thickness of the second coating.

Example 22-25

Examples 22-25 employed substrates of Nicrobell C. The first coat for each was nickle 80wt%, aluminium
20wt%. Examples 22-24 employed second coats of Ni 80wt% Al 20wt% and alumina deposited in the ratios
1:1, 1:3 and 1:9, and third coats, of 8 layers of either alumina or zirconia. Example 25 omitted coats 2 and 3. The first, second, and third coats were deposited by vacuum plasma spraying. The fourth layer of Pt or Pt/5wt% Au was flame sprayed followed by shot peening to ensure that the fourth layer be substantially non-porous. The examples were tested in molten glass at 1200°C for 400 hours, the results are given below.

DURATION OF TEST -400 Hours

Coatings

10	ID	Substra	te lst coat		3rd No. of Al ₂ 0 ₃	4th PGM thicknes	SS
15	22	Nicrobe	ell C Ni20	%Al 1/1/1	8	Pt	81µm
	23	Nicrobe	ell C Ni20	%Al 1/1/1	8	Pt	176µm
	24	Nicrobe	ell C Ni20	%Al 1/1/1	8	Pt/Au	76µm
20	25	Nicrobe	ell C Ni20	%Al -	-	Pt	99µm
	Resu	<u>ults</u> – Coa	ating condi	tion after t	est		
25	22.	AGL	good, inta	ct. Pt too	thin to w	ithstand	cooling
		GL	no differe	nt to BGL			
30		BGL	glass adhe	res- on cool	ing in si	tu Pt pa	rtially cracks
			Glass colo	ur - green,	substrate	e protect	ed.
35	23.	AGL	very good,	completely	intact		
		GL	glass adhe	eres			
		BGL	glass adhe	eres- on cool	ing howev	ver coati	ng stays intact
40			Glass cold	our – pale gr	een, subs	strate pr	cotected.
45	24.	AGL	cracked, d	lue to lack o	of ductil:	ity on co	ooling
		GL	coating co	omes off on o	cooling		
		BGL	minimal ad	herence of g	glass, cra	acks on o	cooling
50			Glass cold	our - green,	substrate	e protect	ted.

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25. AGL mauve, brown due to Ni/Cr diffusion through Pt GL Pt in colour, glass coloured BGL coating OK Glass colour - dark green, substrate not proected.

10 The results obtained for Examples 22, 23 and 24 show that the coatings have protected the substrates from molten glass attack, the thicker the platinum group metals (PGM) layer (Example 23) the better the degree of protection. A thicker layer of PGM gives more ductility. Adequate PGM coating ductility is essential to promote protection.

15 Example 26-28

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Examples 26-28 are components to be used in temperature measurement. They were manufactured using Nicrobell C substrates, for trials in an industrial glass bottle manufacturing plant. The coating system applied was identical to that for Example 23. The example components were placed in a forehearth containing flowing molten flint glass at a temperature which ranged from 1130-1160°C for varying times. The details are given below.

25	ID	Substrate		2nd 1st:3rd 1:1, 1:3, 1:9		4th PGM thickness		Typical temperature °C
30	26	Nicrobell	C Ni2	0%Al 1/1/1	8	Pt 169µm	24 hrs	1137°C
	27	Nicrobell	C Ni2	0%Al 1/1/1	8	Pt 171μm	100 hrs	1160°C
35	28	Nicrobell	C Ni2	0%Al 1/1/1	8	Pt 174µm	1000 hrs	1160°C

Results- Coating condition after test

After testing all Examples (26-28) were completely intact, no corrosion had occurred nor cracking on cooling. The example components had been removed from the molten glass before cooling unlike the laboratory tests where the examples cooled in situ. The glass below the glass line had not adhered to the samples evenly. Some areas had been left completely free of glass. On cooling, particularly Example 28, the attatched glass cracked and fell away from the metal surface. The coating was concluded to have 'nonstick' characteristics. This is an added advantage reducing any cross-contamination and aiding recycling and examination of the components.

Example 29-32

Examples 29-32 employed substrates of Nicrobell C. The first coat was nickle 80wt%, aluminium 20wt%, the second coats were Ni 80wt%, Al 20wt% and alumina deposited in the ratios 1:1, 1:3 and 1:9. The third coats were 8 layers of alumina. The first, second, and third coats were deposited by vacuum plasma spraying. The fourth layers, of Pt or Pt/5wt% Au were flame sprayed, with this coating in some cases being completed by deposition of ZGS Pt or PT 5wt%Au. This was followed by shot peening to ensure that the outer metal coating be substantially non-porous.

55 The examples were tested in molten glass in the laboratory at 1200°C for 400 hours, the results are given below.

DURATION OF TEST -400 Hours

5	I.D) Substrate		lst coat	Coatings 2nd 1st:3rd 1:1, 1:3, 1:9	3rd No. of layers Al ₂ 03				
10	29	Nicrobell C		Ni20%Al	1/1/1	8	Pt 85µm	+ ZGS Pt	69µm	
	30	Nicrob	ell C	Ni20%Al	1/1/1	8	Pt 82µm	+ Pt/5%A	100µm	
15	31	Nicrob	ell C	Ni20%Al	1/1/1	8	Pt/5%Au	121µm		
	32	Nicrob	ell C	Ni20%Al	1/1/1	8	Pt/5%Au	158µm		
	Res	Results- Coating condition after test								
20	29.	AGL	good,	intact co	pating					
		GL	no cor	rosion						
25	BGL slight deformation in coating due to cooling in situ								itu, no	
			diffus							
30			Glass	colour -	pale green,	substr	ate pro	tected.		
	30.	AGL	good,	intact co	bating					
35		GL	no cor	rosion						
	BGL good intact coating, however a slight grey surface									
I	I		appear	ance						
40			Glass	colour -	pale green,	substr	ate prot	ected.		
45	31.	AGL	single	cooling	crack					
		GL	no obvious effect							
	I	BGL	a grey	surface	appearance.					
50			Glass	colour -	green, subst	tantial	ly prote	ected.		

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- 32. AGL good coating
 - GL no obvious change

BGL a white grey surface appearance but intact Glass colour - pale green, substrate protected.

10 The results for Examples 30-34 indicate that a molten glass resistant coating can produced from ZGS Pt. To achieve protection with Pt 5wt%Au a considerably thicker metal coating is required than with Pt alone.

Example 33-38

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- 15 The substrate employed in Examples 33-38 were selected from Nicrobell C and MA956 alloy closed end tubes. MA956 is the trade name of an alloy of composition Fe 74wt%, Cr 20wt%, Al 4.5wt%, Ti 0.5wt% and Y₂O₃ 0.5wt% produced by Inco Alloys Ltd. The first coat was nickel 80wt%, aluminium 20wt% or nickel 95.5wt%, aluminium 4.5wt%, deposited by vacuum plasma spraying (VPS) and air plasma spraying (APS) respectively. For the APS samples in this instance the second coating was absent however for Example 37 the coating sys-
- 20 tem was as in examples 27-29. The third layer was of alumina of thicknesses 120-245 μm. The fourth layer of Pt was deposited and processed as per Examples 27-29. The examples were tested in molten glass for 200 hrs at 1200°C, details presented below.

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DURATION OF TEST -200 Hours

5		Coatings								
10	ID		Spraying Sechnique	lst coat	2nd Ni:Al ₂ 03 1:1, 1:3, 1:9	3rd No. of Al ₂ 0 ₃	4th PGM thickness			
	33	Nicrobell C	APS Ni	4.5%Al	-	135µm	Pt 148µm			
15	34	Nicrobell C	APS Ni	4.5%Al	-	245µm	Pt 159µm			
10	35	MA956	VPS Ni	20%Al	1/1/1	160µm	Pt 138µm			
	36	MA956	VPS Ni	20%Al	-	160µm	Pt 160µm			
20	, 37	MA956	APS Ni	4.5%Al	-	220µm	Pt 164µm			
	38	MA956	APS Ni	4.5%Al	_	120µm	Pt 166µm			
25		The thi	ckness of	the ce	ramic layer	s that ha	we been quoted			
			vious exa	-			n approximate			
30) conv	eration facto	r is that o	one lay	ver is equa	l to 20μm				
	Resu	<u>lts</u> – Coating	condition	after	test					
35	22									
	33.				- no crack	S				
40	,									
		BGL coating OK Glass colour - green, substrate substantially protected.								
		GIASS	colour – gi	ceen, s	substrate s	ubstantia	lly protected.			
45	34.	AGL Very s	light Ni/C							
	;				ISION					
50										
		BGL good coating Glass colour - pale green, substrate protected.								
		GIASS	corour – pa	ire gre	en, substra	ate prote	cted.			

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	35.	AGL	no observable effect
		GL	no observable effect
5		BGL	no observable effect
			Glass colour - pale green, substrate protected.
10	36.	AGL	no observable effect
		GL	no observable effect
15		BGL	no observable effect
			Glass colour - pale green, substrate protected.
20	37.	AGI.	localized failure with breakaway oxidation
	57.	GL	coating OK, no diffusion
25		BGL	coating OK, no diffusion
			Glass colour – pale green, substrate essentially
30			protected.
30			
	38.	AGL	coating disrupted in localised zones
35		GL	coating ok
		BGL	coating ok
			Glass colour – pale green, substrate essentially
40			protected.

For the short duration no noticeable difference occured between the VPS examples without or with the barrier layer. The APS examples indicate that as long as sufficient thickness of barrier layer is laid down, to offset of increased porosity, protection is given. Examples 35-36 indicate, iron based alloys, such as the corrosion

45 of increased porosity, protection is given. Examples 35-36 indicate, iron based alloys, such as the corrosion resistant alloy MA956, can be given additonal protection for service in molten glass enviroment by application of a coating system according to this invention.

50 Claims

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- 1. An article for use in high temperature corrosive environments comprising a metallic substrate on which is deposited successively,
 - i) a first coating of one or more metals or alloys,
 - ii) an optional second coating of a mixture of one or more ceramics and one or more metals or alloys,iii) a third coating of one or more ceramics, and

iv) a fourth coating of one or more precious metals or alloys thereof, the fourth coating being substantially non-porous.

- 2. An article according to claim 1 wherein the metallic substrate has deposited all four said coatings.
- 3. An article according to claim 1 or 2 wherein the first coating is of a metal or alloy selected from nickelaluminium, nickel-chromium, palladium-aluminium, hafnium-aluminium, titanium, tantalum and palladium.
- **4.** An article according to claim 3 wherein the first coating is of an alloy of nickel 80wt% aluminium 20wt%.
- 5. An article according to any preceding claim wherein the ceramic of the second and third coatings is alumina.
- 6. An article according to claim 5 wherein the ceramic is of purity 99%.
- 7. An article according to any proceeding claim wherein the second coating has a composition in which the proportion of metal to ceramic decreases from the interface adjacent the first coating to the interface adjacent the second coating.
- 8. An article according to any preceding claim wherein the fourth coating is of platinum or an alloy of platinum.
- 9. An article according to any preceding claim wherein the fourth coating is of platinum.
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- **10.** An article according to any preceding claim in contact with molten glass or the atmosphere above molten glass.
- 11. A method of making an article claimed in any of the preceding claims comprising applying, successively, to a metallic substrate a first coating of one or more metals or alloys, a second coating of a mixture of one or more ceramics and one or more metals or alloys, a third coating of one or more ceramics and a fourth coating of one or more precious metals or alloys thereof, and making the fourth coating substantially non-porous.
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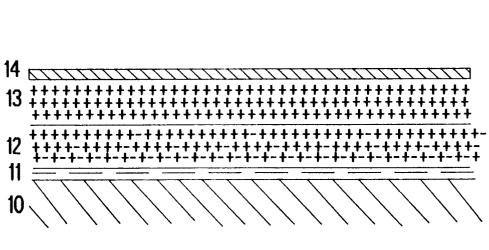


FIGURE 1

