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(54) **Metal anodes for electrolytic acid solutions containing fluorides or fluoroanionic complexes.**

(57) The present invention relates to metal anodes for oxygen evolution from solutions containing fluorides or anionic fluorocomplexes such as tetrafluoroborates and hexafluorosilicates, said anodes having a metal substrate or matrix selected in the group comprising nickel-copper alloys with a copper content in the range of 2.5 and 30% by weight, tungsten or tantalum, niobium or titanium, combinations thereof or alloys of the same with palladium, nickel or yttrium. Said anodes further comprise electrocatalytic compounds for oxygen evolution dispersed in the metal matrix. In the case of nickel- copper alloys, useful electrocatalytic compounds are cerium or tin dioxides, with suitable additives, while for tungsten, cobalt added with nickel, iron, copper or palladium may be used. The same electrocatalytic compounds may be advantageously applied to said metal substrate or matrix in the form of a coating using the conventional technique of thermal decomposition of paints containing suitable precursors or by thermal deposition such as plasma-spray.

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Electrolytes containing anionic fluorocomplexes are commonly used in conventional technologies for the electrolytic recovery of metals, such as lead, tin, chromium. In the specific case of lead recovery from batteries scraps, the scraps are leached with acid solutions containing tetrafluoroborates  $\text{BF}_4^-$  and hexafluorosilicates  $\text{SiF}_6^-$ . The electrolysis of these solutions produces lead as a solid deposit: therefore the electrolytic cells are diaphragmless and have a very simple design. However, this advantage has been so far counterbalanced by the scarce resistance of the substrates to the aggressive action of anionic fluorocomplexes on the anodes whereat oxygen is evolved. Further a parasitic reaction may take place with formation of lead dioxide which subtracts lead to the galvanic deposition of the metal, thus reducing the overall efficiency of the system.

Upon carefully considering the prior art teachings found for example in U.S. 3,985,630, 4,135,997, 4,230,545, 4,272,340, 4,460,442, 4,834,851 and in Italian patent application no. 67723A/82, it may be concluded that :

- anodes made of carbon or graphite, as such or coated by lead dioxide, are known in the art but offer a rather limited active lifetime, in the range of a hundred hours due to the oxidizing action of oxygen evolution. Obviously this brings forth higher maintenance costs for substituting the anodes and additional costs connected to the consequent production losses;
- anodes made of titanium, coated by lead dioxide or platinum or oxides of the platinum group metals, still undergo corrosion, though to a far less extent with respect to carbon or graphite, in any case insufficient for counterbalancing the higher construction costs;
- anodes made of tantalum coated by platinum metal or metal oxides offer a much longer lifetime than titanium but the production costs are extremely high;
- the parasitic reaction of lead dioxide deposition onto any type of anode may be prevented adding a suitable inhibitor to the leaching solution, for example phosphoric acid, antimony acid or arsenic acid. However, the quantities required may spoil the compactness of the lead metal deposit. This problem is overcome by resorting to an anode having a coating made of metals or oxides of the platinum group metals and at least one element comprised in the group of arsenic, antimony, bismuth, tin. In this case a remarkably lower quantity of inhibitor to prevent the anodic deposition of lead dioxide is required and the deterioration of the produced lead deposit is eliminated.

It is therefore evident that the prior art does not provide for an anode offering both a long lifetime (higher than 1000 hours) and a limited cost, which are both necessary features for a wide industrial application.

The present invention permits to overcome the disadvantages of the prior art by providing for an anode characterized by a reduced cost, high resistance to the aggressive conditions of oxygen evolution in solutions containing anionic fluorocomplexes and even free fluorides, good catalytic properties for oxygen evolution, that is lower electrolysis potential with consequently reduced energy consumptions.

The anode of the present invention comprises a matrix made of one or more metals or metal alloys capable of passivating by forming a protective layer of oxides or oxyfluorides and one or more compounds of suitable elements capable of favouring oxygen evolution, said elements being embedded into the matrix or alternatively applied to the same in the form of an external coating. Said anode is suitable for use in electrometallurgical processes for the deposition of lead, tin, chromium, from solutions containing fluorocomplex anions such as tetrafluoroborates and hexafluorosilicates or free fluorides.

The present invention comprises also the electrolytic process for recovering metals in cells equipped with anodes and cathodes and fed with acid solutions containing metal ions and anionic fluorocomplexes such as tetrafluoroborates and hexafluorosilicates, wherein said anodes are of above mentioned type.

The following description will take into consideration the particular case of electrolytic recovery of lead, for simplicity sake. In this process the leaching solution to be electrolyzed has the following composition:

- tetrafluoroboric acid,  $\text{HBF}_4$ , or hexafluorosilic acid,  $\text{H}_2\text{SiF}_6$ : 40-240 g/l;
- dissolved lead :40-80 g/l;
- temperature: 15-35 °C;
- current density (anodic and cathodic): 150-2000 A/m<sup>2</sup>.

Electrolysis occurs between the anode and the cathode, with the following reactions:

- cathode:  $\text{Pb}^{++}(\text{complex}) + 2\text{e}^- \rightarrow \text{Pb}(\text{compact metal})$
- anode :  $\text{H}_2\text{O} - 2\text{e}^- \rightarrow 2\text{H}^+ + 1/2\text{O}_2$  (main reaction)  $\text{Pb}^{++}(\text{complex}) + 2\text{H}_2\text{O} - 2\text{e}^- \rightarrow \text{PbO}_2 + 4\text{H}^+$  - (parasitic reaction)

Suitable elements for the anode are : titanium, niobium, tantalum, tungsten or alloys thereof such as :

- titanium-palladium (Pd 0.2%),
- titanium-nickel (Ni 0.5-1.5%);
- titanium-yttrium

- titanium-tantalum (Ta 0.5-5.0%)
- titanium-niobium (Nb 0.5-5.0%)
- titanium-tungsten (W 0.5-5.0%)
- copper-tantalum (niobium);
- 5 - titanium-tantalum (niobium)

Further, it has been surprisingly found that alloys of nickel-copper, obtained either by sinterization of the powders of the elements or by melting and casting in suitable moulds readily passivate when put in contact with the aforementioned solutions, that is they become coated by a protective layer of oxides or oxyfluorides or insoluble fluorides when the copper content is in the range of 2.5 to 30% and more preferably between 5 and 20%.

The poor conductivity of the protective film formed on the above metals gives rise to a high potential and consequently to high energy consumptions in the process of lead recovery.

It has been found that when using tungsten and nickel-copper alloys, if suitable elements are dispersed into the metal matrix, the oxygen evolution potential is remarkably reduced, bringing the energy consumption to quite acceptable levels for industrial applications for the production of lead.

Suitable compounds for anodes based on nickel-copper are cerium oxide,  $\text{CeO}_2$ , added with  $\text{Nb}_2\text{O}_5$  (1-5%),  $\text{NiO}$  (0.5-2%),  $\text{Pr}_6\text{O}_{11}$  (0.5-2%),  $\text{CuO}$  (0.5-2%) and tin dioxide,  $\text{SnO}_2$ , added with  $\text{Sb}_2\text{O}_3$  (0.5-4%) and  $\text{CuO}$  (0.5-2%); while for anodes based on tungsten, addition of cobalt (5-35%) optionally mixed with minor amounts of iron and nickel (1-2%), copper, palladium and cerium result more positive.

The same results are alternatively obtained by applying to the metal matrix a coating exhibiting electrocatalytic properties for oxygen evolution, chemical stability and possibly limited porosity to ensure an adequate protection to the metal matrix.

In the case of tungsten and nickel-copper alloys suitable coatings are obtained by cerium and tin oxides as above described for the dispersion in the metal matrix. As for the other alloys, testing has shown that a suitable coating must comprise a matrix made of tungsten or other metal of the VIB group (70-99%), cobalt (1-30%) as the electrocatalyst for oxygen evolution to inhibit possible parasitic reactions and further comprising suitable additives selected from the group comprising nickel, palladium, cerium and copper, or optionally a combination of the same, (0.5-2%).

The following examples describe various embodiments of the present invention without limiting the invention to the same.

#### EXAMPLE 1

Eight rods having a diameter of 20 mm, 100 mm long, made of nickel-copper alloys, having different compositions, have been prepared by monostatic lateral pressing (about  $250 \text{ kg/cm}^2$ ) starting from the powders of the elements (1-10 microns) and subjected to subsequent thermal treatment in inert environment at  $950\text{-}1150^\circ\text{C}$  for 6-12 hours (preferably between  $980$  and  $1080^\circ\text{C}$  for 8-10 hours) followed by a second oxidizing treatment in air at  $900\text{-}1300^\circ\text{C}$  for 100-600 hours (preferably  $970\text{-}1000^\circ\text{C}$ , 300-400 hours for copper contents higher than 10-15%).

At the same time three reference samples have been prepared as follows:

- two rods having a diameter of 20 mm, 100 mm long, based on commercial Monel<sup>(R)</sup>, one of the 400 type and the other of the K500 type oxidized at the conditions used for the samples obtained by sinterization
- one sheet of  $10 \times 100 \times 1 \text{ mm}$  made of commercial graphite coated by a deposit of  $\beta\text{-PbO}_2$  obtained by galvanic deposition from nitrate bath.

The sintered rods and the reference samples have been tested as anodes in the electrolysis from a fluoroboric solution, which is the typical electrolyte used for metal lead recovery from batteries scraps.

The operating conditions and the results are reported in the following Table.

TABLE 1.1

OPERATING CONDITIONS	
<ul style="list-style-type: none"> <li>- HBF<sub>4</sub>, tetrafluoroboric acid</li> <li>- Temperature</li> <li>- Cathode</li> <li>- Procedure</li> </ul>	<p>80 g/l Ambient Lead</p> <p>Determination of the corrosion potential (PC) by electrochemical potentiostatic procedure and analysis of the solution and cathodic deposit; comparison with the oxygen evolution potential (PO) detected on a graphite electrode coated by beta-PbO<sub>2</sub>. The value Delta V = PC - PO defines the stability or instability degree of the various materials.</p>

TABLE 1.2

No.	SAMPLES	RESULTS				
		Anodic Potential V(NHE)		Delta V		
		Corrosion	O <sub>2</sub> Evolution	PC-PO		
		PC Volts	PO Volts	(Volts)		
		400	1000	400	1000	
		A/m <sup>2</sup>	A/m <sup>2</sup>	A/m <sup>2</sup>	A/m <sup>2</sup>	
1	Beta-Pb <sub>2</sub> on graphite		2.07 2.24			
2	Monel 400 type	+ 0.38		-1.69	1.86	corroded
3	Monel K500 type	+ 0.39		-1.68	-1.86	"
4	Ni 99-Cu 1	- 0.1		-2.17	-2.34	"
5	Ni 98-Cu 2	+ 0.36		-1.71	-1.88	"
6	Ni 97.5-Cu 2.5	+ 1.30		-0.77	-0.94	"
7	Ni 95-Cu 5	> 2.3		>0.23	>0.06	passivated
8	Ni 90-Cu 10	> 2.3		>0.23	>0.06	"
9	Ni 80 Cu 20	> 2.3		>0.23	>0.06	"
10	Ni 70 Cu 30	+ 0.99		-1.08	-1.25	corroded
11	Ni 65-Cu 35	+ 0.43		-1.65	-1.81	"

The above results lead to the following considerations:

- oxygen evolution on beta-PbO<sub>2</sub> occurs at potentials (PO) comprised between 2.07 and 2.24 Volts at current densities between 400-1000 A/m<sup>2</sup>. It is evident that any material having a Corrosion Potential (PC) lower than these values is characterized by instability (tendency to dissolve). The various potentials refer to a reference normal hydrogen electrode (NHE);
- the materials with a copper content between 5 and 20% are stable under oxygen evolution.

Similar materials obtained not by sinterization but by moulding with casting wax showed the same behaviour.

## EXAMPLE 2

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12 rods having a diameter of 20 mm, 100 mm long, made of sintered nickel-copper alloys have been prepared as described in Example 1, the only difference being the addition of preformed powders (pigments) based on tin oxide and cerium oxide. The electrolysis conditions and the results expressed in terms of anodic potentials, V(NHE) for oxygen evolution at 1000 A/m<sup>2</sup> after 300 h, cathodic faradic efficiency % calculated on lead and stability/ un- stability of the material under corrosion, are reported in Tables 2.1. and 2.2

TABLE 2.1

- HBF <sub>4</sub> , tetrafluoroboric acid	150 g/l
- lead ion	60 g/l
- H <sub>3</sub> PO <sub>4</sub> , phosphoric acid	2 g/l
- temperature	Ambient
- cathode	Lead
- anodic current density	1000 A/m <sup>2</sup>

TABLE 2.2

5	SAMPLES						RESULTS		REMARKS		
	Composition						O <sub>2</sub> evolution		Faradic		
	%						PO V(NHE)		efficiency		
	No.	Matrix	Additives				initial	300h	%		
10	<u>Ni-Cu</u>		<u>SnO<sub>2</sub>-Sb<sub>2</sub>O<sub>3</sub></u>								
15	<u>90-10</u>										
	1	95	5	=			6.8	=	100	corroded	
20	2	95	4.90	0.10			2.5	2.6	100	not corroded	
	3	90	9.80	0.20			2.45	2.8	100	"	
25	<u>Ni-Cu</u>		<u>SnO<sub>2</sub> - Sb<sub>2</sub>O<sub>3</sub></u>								
	<u>80-20</u>										
	4	95	5	=			6.8	=	100	corroded	
30	5	95	4.9	0.1			2.5	2.38	100	not corroded	
	6	90	9.8	0.2			2.45	2.38	100	"	
35	<u>Ni-Cu</u>		<u>CeO<sub>2</sub>-Ta<sub>2</sub>O<sub>5</sub>-NiO-Pr<sub>6</sub>O<sub>11</sub></u>								
	<u>90-10</u>										
	7	95	5	==	==	==	8.5	==	100	corroded	
40	8	95	4.9	0.1	==	==	2.8	2.65	100	not corroded	
	9	95	4.8	0.1	0.1	==	2.9	2.6	100	"	
	10	95	4.8	0.1	0.05	0.05	2.8	2.55	100	"	
45	11	90	9.6	0.2	0.1	0.1	2.7	2.3	100	"	
	<u>80 20</u>										
50	12	90	9.6	0.2	0.1	0.1	2.8	2.40	100		

The results obtained on Ni-Cu alloys bring to the following conclusions:

55 Tin dioxide:

- corrosion on SnO<sub>2</sub> without additives
- no visible corrosion under operation with O<sub>2</sub> evolution on SnO<sub>2</sub> added with Sb<sub>2</sub>O<sub>3</sub> after 300 hours of

operation

#### Cerium dioxide

- 5     - anodic corrosion on  $\text{CeO}_2$  without additives  
       - no visible corrosion under operation with oxygen evolution after 300 hours of operation with  $\text{CeO}_2$  containing additives  
       - increasing electrocatalytic activity according to the following order:



Similar results may be obtained with Ni-Cu structures coated by an electrocatalytic coating, having the same composition as the particles used for the dispersion embedded in the matrix, said coating being applied by thermal decomposition of a paint containing suitable precursors. It is also to be pointed out that  
 15 the addition of only 2 g/l of phosphoric acid ensures 100% cathodic Faradic efficiency: this means that no lead dioxide is formed at the anode.

#### EXAMPLE 3

20     Four rods, with a diameter of 20 mm, 100 mm long, made of nickel-copper alloy, have been obtained by casting the component metals together with powders based on tin oxide and/or cerium oxide (diameter 40-60 microns). Said samples have been tested as anodes for the electrolysis of fluoroboric solutions according to the conditions and procedures described in Example 2. The results are reported in Table 3.1.

25     **TABLE 3.1**

SAMPLE No.	Composition %					O <sub>2</sub> Evolution		Faradic
	Matrix	Additives				PO Volts (NHE)	Init.	Efficiency (%)
<hr/>								
	Ni-Cu		SnO <sub>2</sub> Sb <sub>2</sub> O <sub>3</sub> CuO					
	90-10							
1	95	4.8	0.15	0.05		2.5	2.8	100
3	95	4.8	0.15	0.05		2.5	2.38	100
	Ni-Cu		CeO <sub>2</sub> Ta <sub>2</sub> O <sub>5</sub> NiO Pr <sub>6</sub> O <sub>11</sub>					
	80-20							
2	95	4.8	0.1	0.05	0.05	2.9	2.7	100
4	95	4.8	0.1	0.05	0.05	2.8	2.35	100

The samples reported in Table 3.1 showed also that metal structures made of  $\text{Cu}_{20(10)}\text{-Ni}_{80(90)}$  after addition of  $\text{SnO}_2$  or  $\text{CeO}_2$  containing additives do not undergo any visible corrosion when used as anodes for oxygen evolution.

#### EXAMPLE 4

15 commercial tungsten rods with different contents of cobalt, nickel and iron have been used as



anodes for oxygen evolution in the electrolysis of fluoroboric solutions as illustrated in Example 2. The results are reported Table 4.1

TABLE 4.1

SAMPLES					RESULTS		REMARKS	
<u>Composition</u> (%)					O <sub>2</sub> Evolution		Faradic	
					PO Volts(NHE)		Efficiency	
No.	W	Co	Ni	Fe	Initial	300 h	%	
1	100				>6.00		passivated	
2	90	10			2.6	2.4	100	slight Co
								corroded
3	80	20			2.3	2.3	100	heavy Co leaching
4	70	30			2.2	2.2	100	heavy Co leaching
5	65	35			2.2	2.2	100	corroded
6	95	==	5	==	3.2	=	=	close to
								passivation
7	90	==	10	==	2.6	3.5	=	"
8	95	==	==	5	3.8	=	=	"
9	90	==	==	10	2.3	4.1	=	"
10	90	8	1	1	2.2	2.3	100	not corroded
11	80	15	2.5	2.5	2.1	2.2	100	"
12	63	35	1	1	2.1	2.1	100	"
13	60	38	1	1	2.1	2.1	=	"
14	58	40	1	1	2.0	4.0	=	corroded
15	58	38	2	2	2.0	5.0	=	passivated

These results lead to the following conclusions:

- tungsten is stable when used as anode in fluoroboric solutions (passivation)
- elements like Co, Ni, Fe in minor amounts perform an electrocatalytic activity for oxygen evolution
- the following series show an electrocatalytic activity increasing as per the following order : Fe < Ni < Co < Co + Ni + Fe
- a critical concentration threshold for each additive or combination of the same has been found beyond which passivation or corrosion phenomena occur.

Similar results may be obtained by applying to the tungsten structure an electrocatalytic coating as

described in Example 2.

#### EXAMPLE 5

- 5 6 rods having a diameter of 20 mm, 100 mm long, labelled as follows:  
 sample 1 as in Example 2, no. 6  
 sample 2 as in Example 2, no. 12  
 sample 3 as in Example 3, no. 3  
 sample 4 as in Example 3, no. 4  
 10 sample 5 as in Example 4, no. 4  
 sample 6 as in Example 4, no. 11  
 have been used as anodes for electrolysis of fluorosilic solutions containing lead ions and phosphoric acid.  
 The electrolysis conditions are reported in Table 5.1.

15 TABLE 5.1

- $\text{H}_2\text{SiF}_6$ , fluorosilicic acid	100 g/l
- $\text{H}_3\text{PO}_4$ , phosphoric acid	6 g/l
- lead ions	60 g/l
- temperature	ambient
- anodic current density	1000 A/m <sup>2</sup>
- cathode	lead

25 The results are reported in Table 5.2.

TABLE 5.2

SAMPLES No.	RESULTS			REMARKS
	O <sub>2</sub> Evolution PO		Faradic	
	Volts (NHE)		Efficiency	
	Initial	300 h	%	
1	2.45	2.38	100	not corroded
2	2.8	2.45	100	"
3	2.5	2.38	100	"
4	2.8	2.35	100	"
5	2.2	2.22	100	"
6	2.1	2.2	100	

#### EXAMPLE 6

55 Seven anodes having a passivable metal matrix and a coating based on tungsten and cobalt were prepared; further four anodes were also tested as shown herebelow. The anodes, in the form of sheets, 100 x 10 x 1 mm, of commercial pure titanium, were sandblasted and samples 1 to 3 were further subjected to

chemical pickling in boiling 20% HCl. All the samples were then coated by different kinds of coatings and tested at the same conditions illustrated in Example 2. The description of the anodes and the results of the tests are reported in Tables 6.1 and 6.2.

TABLE 6.1

SAMPLE No.	COATING Composition %	Thickness (micron)	Load g/m <sup>2</sup>	Application Procedure
1	RuO <sub>2</sub> TiO <sub>2</sub> (50) (50)	11.2	20 (Ru)	painting + thermal decomposition
2	IrO <sub>2</sub> Ta <sub>2</sub> O <sub>5</sub> (50) (50)	10.5	20 (Ir)	"
3	Pt Sb (>98) (<2)	2	21.5 (Pt)	galvanic deposition
4	beta PbO <sub>2</sub>	800	//	"
5	W	155	//	plasma jet
6	Co	130	//	"
7	Co	140	//	thermo spray
8	W + Co (97.5)(2.5)	145	//	plasma jet
9	W + Co (90) (10)	135	//	"
10	W + Co (80) (20)	130	//	"
11	W + Co (70) (30)	130	//	"

TABLE 6.2

SAMPLE No.	RESULTS			REMARKS
	O <sub>2</sub> Evolution	Faradic		
	PO Volts (NHE)	Efficiency		
	Initial	300 h	%	
1	1.75	//	//	corroded after 125 h
2	1.80	//	//	corroded after 140 h
3	1.76	1.67	80	corroded
4	1.93	1.63	70	corroded
5	>3.0	//	//	passivated
6	1.9	1.59	60	Co leaching, corroded
7	1.93	1.68	50	complete Co leaching, corroded
8	2.09	2.35	100	slight Co leaching, incipient passivation
9	2.05	2.08	100	" "
10	2.00	1.75	60	heavy Co leaching, corroded
11	2.00	1.63	30	" "

Conventional coatings on titanium, such as noble metal oxides (e.g. RuO<sub>2</sub> and IrO<sub>2</sub>) stabilized by valve metals, noble metals ((e.g. Pt) and lead dioxide (beta PbO<sub>2</sub>) are mechanically (PbO<sub>2</sub>) and/or chemically (Pt, IrO<sub>2</sub>, RuO<sub>2</sub>) unstable also after a few dozens of hours with the consequent corrosion of the substrate areas remained uncoated. The coatings based on tungsten passivated after a few minutes. The coatings based on cobalt corroded after a few hours while coatings based on tungsten-cobalt with cobalt contents around 10% show neither corrosion nor passivation. Lower cobalt contents do not prevent the passivating action of tungsten from prevailing with time while with higher cobalt contents dissolution is observed which causes mechanical instability of the remaining coating.

## EXAMPLE 7

15 sheets, 10 x 10 x 1 mm, of commercial pure titanium, after sandblasting with corindone (pressure: 7 atm: distance of spraying pistol from substrate: 30-35 cm; abrasive grain : irregular shape, sharp edged, average diameter about 300 microns) were coated by plasma jet or thermospray technique with tungsten and cobalt coatings containing nickel, palladium and copper as doping elements. The samples thus obtained were used as anodes in the electrolysis of lead fluoroborate solutions at the same conditions as

illustrated in Example 2. The characteristics of the anodes are reported in Table 7.1 and the relevant results in Table 7.2.

TABLE 7.1

SAMPLE		COATING		
No.	Matrix	Composition	Thickness	Application
		%	microns	Procedure
1	Ti	W + Co (90) (10)	140	plasma spray
2	"	W + Co + Ni (89)(10.5) (0.5)	145	plasma spray
3	"	W + Co + Ni (89)(10) (1.0)	145	plasma spray
4	"	W + Co + Ni (89)(9.5) (1.5)	135	plasma spray
5	"	W + Co + Ni (88)(10) (2.0)	130	plasma spray
6	"	W + Co + Pd (89)(10.5)(0.5)	100	plasma spray
7	"	W + Co + Pd (89)(10) (1.0)	110	plasma spray
8	"	W + Co + Cu (88)(11.5) (0.5)	105	plasma spray
9	"	W + Co + Cu (89)(10.5) (0.5)	115	plasma spray
10	"	W + Co + Cu (89)(10) (1.0)	125	plasma spray
11	"	W + Co + Cu (90)(8.5) (1.5)	125	plasma spray
12	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	130	plasma spray
13	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	130	thermo spray
14	"	W + Co + Ni + Cu (89)(10)(0.5)(0.5)	145	plasma spray
15	"	W + Co + Ni + Cu (89)(10)(0.5)(0.5)	120	thermo spray
16	"	W + CeO <sub>2</sub> (97.5) (2.5)	100	plasma spray
17	"	W + CeO <sub>2</sub> + Co (92.5) (2.5) (5)	110	plasma spray
18	"	W + CeO <sub>2</sub> + Co (87.5) (2.5) (10)	100	plasma spray

TABLE 7.2

5	SAMPLE	RESULTS			REMARKS
	O <sub>2</sub> Evolution		Faradic		
	PO Volts (NHE)		Efficiency		
	No.	Initial	300 h	%	
10	1	2.04	2.17	100	slight Co leaching
	2	2.05	2.19	100	" "
15	3	2.04	2.08	100	no corrosion
	4	2.04	2.09	100	" "
	5	2.02	2.18	100	slight Co leaching
20	6	2.09	2.06	100	no corrosion
	7	2.07	2.10	100	Pd traces in solution
25	8	2.09	2.22	100	no corrosion
	9	2.07	2.18	100	" "
	10	2.07	2.14	100	no corrosion
30	11	2.06	2.23	75	Cu traces in solution
	12	2.06	2.09	100	no corrosion
35	13	2.07	2.07	100	" "
	14	2.05	2.10	100	" "
	15	2.08	2.08	100	" "
40	16	//	//	100	passivated
	17	2.11	2.15	100	no corrosion
45	18	2.05	2.10	100	no corrosion

The results permit to state that minimum quantities of nickel, palladium, copper (1-1.5%) in a possible combination improve the chemical and electrochemical stability of the coatings. For each additive an optimum concentration has been determined in the range of 1-1.5% corresponding to the best performances. The presence of nickel, copper and palladium in the above concentrations avoids or in any case remarkably reduces the anodic leaching of cobalt. The combined presence of the above elements, for example Ni + Pd or Ni + Cu, to an amount of 1-1.5% stabilizes the operating potential. This effect is particularly enhanced when the coating is applied by thermospray.

## EXAMPLE 8

17 sheets made of commercial titanium and titanium alloys (100 x 10 x 1 mm) were prepared according

to the procedures described in Example 7 and coated by plasma or thermospray technologies with deposits based on W + Co, W + Co + Ni, W + Co + Ni + Pd, W + Co + Ni + Cu. The samples were tested as anodes in the electrolysis conditions described in Example 2 but with a double anodic current density (2000 A/m<sup>2</sup>). The characteristics of the samples are reported in Table 8.1 while the results are reported in Table 8.2.

TABLE 8.1

SAMPLE		COATING		
No.	Matrix	Composition	Thickness	Application
		%	microns	Procedure
1	Ti	W + Co (90)(10)	120	plasma
2	"	W + Co + Ni (89)(10)(1)	140	plasma
3	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	135	plasma
4	"	W + Co + Ni + Cu (89)(10)(0.5)(0.5)	135	plasma
5	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	120	thermo-spray
6	TiPd	W + Co (90)(10)	110	plasma
7	"	W + Co + Ni (89)(10)(1)	105	plasma
8	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	110	plasma
9	"	W + Co + Cu + Pd (89)(10)(0.5)(0.5)	115	thermo-spray
10	TiNi	W + Co (90)(10)	120	plasma
11	"	W + Co + Ni (89)(10)(1)	105	plasma
12	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	115	plasma
13	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	115	thermo-spray
14	Ti-Y	W + Co (90)(10)	120	plasma
15	"	W + Co + Ni (89)(10)(1)	125	plasma
16	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	130	plasma
17	"	W + Co + Ni + Pd (89)(10)(0.5)(0.5)	130	thermo-spray

TABLE 8.2

	SAMPLES		RESULTS		REMARKS
	No.	O <sub>2</sub> Evolution	Faradic		
		PO Volts (NHE)	Efficiency		
		Initial	300 h	%	
5	1	2.14	1.93	100	Co leaching, corroded
10	2	2.18	2.05	100	slight Co leaching, corroded
	3	2.19	2.05	100	corroded
15	4	2.19	2.08	100	corroded
	5	2.15	2.10	100	corroded
20	6	2.18	2.13	100	not corroded
	7	2.18	2.13	100	"
25	8	2.18	2.13	100	"
	9	2.18	2.15	100	"
30	10	2.19	2.15	100	"
	11	2.17	2.15	100	"
35	12	2.17	2.15	100	"
	13	2.17	2.16	100	"
40	14	2.20	2.01	100	Co leaching, corroded
	15	2.22	2.02	100	"
45	16	2.18	2.08	100	"
	17	2.19	2.01	100	"

The results obtained at 2000 A/m<sup>2</sup> lead to the following considerations:

- titanium structures, accidentally contacting the electrolyte due to chemical or mechanical removal of the coating, undergo a remarkable corrosion; this negative behaviour is less important with ternary or quaternary deposits, for these latter especially when obtained by thermo-spray, being more compact;
- titanium-yttrium (Y 0.35%) samples show a similar behaviour compared with samples of commercial titanium, with the same coating;
- titanium-palladium (Pd 0.20%) and titanium nickel (Ni 1.5%) samples show a higher stability. Corrosion is lower as it can be seen from the anodic potential values which are stable with time: in fact an increasing potential is a symptom of passivation of the coating, while a decreasing potential shows corrosion of the substrate.



## EXAMPLE 9

5 sheets (100 x 10 x 1 mm) made of titanium, tantalum, niobium, tungsten and of a nickel (90%)-copper (10%) alloy, after a surface treatment as described in Example 7, have been coated by a coating of W (89) + Co (10) + Hi (0.5) + Pd (0.5) applied by plasma jet.

The samples have been tested as anodes in the electrolysis of lead fluoroborates solutions at the same conditions as illustrated in Example 8. The results are reported in Table 9. The cathodic deposition efficiency of lead was 100%.

TABLE 9

No.	Matrix Composition %	Coating Thickness micron	O <sub>2</sub> Evolution PO Volts(NHE) Initial 500 h		REMARKS
1	Ti	140	2.20	2.06	corroded
2	Ta	135	2.17	2.17	no corrosion
3	Nb	145	2.21	2.17	slightly corroded
4	W	125	2.18	2.18	no corrosion
5	Ni(90) + Cu(10)	130	2.20	2.20	no corrosion

The results lead to the following considerations:

- when the substrate is made of tantalum, tungsten or Ni(90) - Cu(10) alloy, a good stability and constant anodic potentials of the coatings applied to the same are experienced;
- the substrate made of titanium is unstable and the anodic potential of the coating rapidly decreases with time;
- an intermediate situation is experienced with the substrate made of niobium with anodic potentials slightly decreasing with time.

## Claims

1. Anode for oxygen evolution from solutions containing fluorides or anionic fluorocomplexes such as tetrafluoroborates and hexafluorosilicates, comprising a passivable metal matrix and an electrocatalytic compound for oxygen evolution, containing at least one additive, characterized in that said electrocatalytic compound is selected in the group comprising cobalt, cerium dioxide or tin oxide.
2. The anode of claim 1 characterized in that said electrocatalytic compound comprises cobalt and at least one additive selected in the group of nickel, copper, iron, palladium, cerium.
3. The anode of claim 1 characterized in that the electrocatalytic compound comprises cerium dioxide and at least one additive selected in the group of niobium oxide, nickel oxide, praseodymium oxide, copper oxide.
4. The anode of claim 1 characterized in that the electrocatalytic compound comprises tin oxide and at least one additive selected in the group of antimony oxide or copper oxide.
5. The anode of claim 1 characterized in that said metal matrix at least on the more external side is made of tungsten, tantalum, niobium, titanium, alloys thereof or alloys of the same with palladium, nickel, yttrium and nickel-copper alloys with a nickel content comprised between 5 and 20%.
6. The anode of claim 1 characterized in that said electrocatalytic compound is present in said metal matrix as an alloy or as a dispersion.
7. The anode of claim 1 characterized in that said electrocatalytic compound is in the form of a coating applied to said metal matrix.
8. The anode of claim 6 characterized in that said metal matrix comprises tungsten and said elec-

trocatalytic compound comprises cobalt and at least one additive selected in the group comprising nickel, copper, iron, palladium, cerium.

- 5 9. The anode of claim 5 characterized in that the concentration of tungsten in said metal matrix is comprised between 70 and 99% by weight.
10. The anode of claim 8 characterized in that the concentration of cobalt is comprised between 1 and 30% by weight.
- 10 11. The anode of claim 8 characterized in that the concentration of said additives is comprised between 0.5 and 2% by weight.
12. The anode of claim 6 characterized in that said metal matrix comprises an alloy of nickel-copper with a copper content in the range of 5 to 20% by weight and said electrocatalytic compound comprises  
15 cerium dioxide doped by at least one oxide selected among niobium oxide, nickel oxide, praesodimium oxide, copper oxide as such or in combination thereof.
13. The anode of claim 7 characterized in that said metal matrix comprises tungsten or nickel-copper alloys with a copper content comprised between 5 and 20% by weight and said coating comprises cerium  
20 dioxide as said electrocatalytic compound containing at least one additive selected in the group niobium oxide, nickel oxide, praesodimium oxide, copper oxide or combinations of the same.
14. The anode of claim 7 characterized in that said metal matrix comprises tungsten or nickel-copper alloys with a copper content comprised between 5 and 20% by weight and said coating comprises as said  
25 electrocatalytic compound tin dioxide containing at least one additive selected among antimony oxide or copper oxide or a combination of the same.
15. The anode of claim 7 characterized in that said matrix comprises nickel-copper alloys with a copper content comprised between 5 and 20% by weight and said coating comprises tungsten and cobalt as  
30 said electrocatalytic compound containing at least one additive selected in the group of nickel, copper, iron, palladium, cerium.
16. Process for the electrolytic recovery of metals carried out in cells equipped with anodes and cathodes and fed with solutions containing metal ions and fluorides or anionic fluorocomplexes characterized in  
35 that said anodes are those described in claims 1-15.
17. The process of claim 16 characterized in that said metal is lead. 18. The process of claim 17 characterized in that said solutions are added with phosphoric acid up to 2 grams per liter.

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## EUROPEAN SEARCH REPORT

Application Number

EP 92 10 3474

DOCUMENTS CONSIDERED TO BE RELEVANT			
Category	Citation of document with indication, where appropriate, of relevant passages	Relevant to claim	CLASSIFICATION OF THE APPLICATION (Int. Cl.5)
D,A	EP-A-0 268 102 (S.E.R.E. S.R.I.) 25 May 1988 * page 9; example 5; tables 2.1 * * page 11; claims 1,5.7 * ---	1,16-18	C25C7/02 C25C1/18
A	EP-A-0 099 866 (ENERGY CONVERSION DEVICES, INC) 1 February 1984 * page 20, line 3 - line 5 * * page 26; table 1 * * page 32, line 13 - line 19 * * page 34; claims 1-5 * ---	1,2,5,6,7	
A	US-A-4 243 503 (DONALD F. LIEB) 6 January 1981 * column 6; example 1 * -----	1,4,5,7	
			TECHNICAL FIELDS SEARCHED (Int. Cl.5)
			C25C C25B
The present search report has been drawn up for all claims			
Place of search THE HAGUE		Date of completion of the search 03 JUNE 1992	Examiner GROSEILLER P.A.
<b>CATEGORY OF CITED DOCUMENTS</b>			
X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background O : non-written disclosure P : intermediate document		T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons ----- & : member of the same patent family, corresponding document	