



(12) **NEW EUROPEAN PATENT SPECIFICATION**

(45) Date of publication and mention
of the opposition decision:
15.03.2006 Bulletin 2006/11

(51) Int Cl.:
B22D 23/00 ^(2006.01) **C22B 9/16** ^(2006.01)
B22D 17/00 ^(2006.01) **B22D 17/10** ^(2006.01)
B22F 1/00 ^(2006.01)

(45) Mention of the grant of the patent:
12.04.2000 Bulletin 2000/15

(86) International application number:
PCT/AU1993/000454

(21) Application number: **93918800.9**

(87) International publication number:
WO 1994/006586 (31.03.1994 Gazette 1994/08)

(22) Date of filing: **06.09.1993**

(54) **PARTICULATE FEEDSTOCK FOR METAL INJECTION MOLDING**

PULVERMISCHUNG ZUM SPRITZGIESSEN VON METALL

MATERIAU PARTICULAIRE D'ALIMENTATION POUR MOULAGE PAR INJECTION

(84) Designated Contracting States:
DE FR GB IT SE

(30) Priority: **11.09.1992 AU PL463692**
29.06.1993 AU PL968093

(43) Date of publication of application:
16.08.1995 Bulletin 1995/33

(60) Divisional application:
99201696.4 / 0 960 673

(73) Proprietor: **Thixomat, Inc.**
Ann Arbor, MI 48104 (US)

(72) Inventors:
• **KJAR, Anthony R.**
Blackburn, VIC 3130 (AU)
• **IACocca, Ronald G.**
State College, PA 16801 (US)
• **GERMAN, Randall M.**
State College, PA 16803 (US)
• **MIHELICH, John Louis**
Prospect, KY 40059 (US)

(74) Representative: **Hallybone, Huw George et al**
Carpmaels and Ransford,
43 Bloomsbury Square
London WC1A 2RA (GB)

(56) References cited:
DE-A- 1 758 656 **DE-A- 3 639 737**
GB-A- 2 182 063 **US-A- 3 881 913**
US-A- 4 063 942 **US-A- 4 460 407**
US-A- 4 694 881 **US-A- 4 694 882**
US-A- 5 040 589

- **Look-up Table Mesh/Microns**
- **Design Engineering Feb. 1989, page 33**
- **Hoechst Data Sheet "Karbid und Metallurgische Produkte", Sept. 1978**
- **Brochure of Reade Manufacturing Company**
- **Brochure of Reynolds Metals Company**
- **Analysis of Eckart Granules of 12.12.2000**
- **PFEIFER H.: "Grundlagen der Fordertechnik" 1989**
- **MENGES G.: "Einführung in der Kunststoffverarbeitung" 1986**
- **"Granulierung der thermoplastischen Kunststoffen" VDI-Verlag GmbH 1974**
- **JOHANNABER F.: "Kunststoffmaschinenführer" 1992**
- **Brochure from Poudmet**
- **ERICKSON S.: Proceedings 44th Annual World Magnesium Conference, Tokyo, 1987**
- **"Thixomolding" Proceedings 47th Annual World Magnesium Conference, Cannes, France, 1990**
- **KURIHARA K. et al: Japanese language article, pages 255-260, April 1981 and English translation thereof**
- **GHOSH D. et al: Proceedings on the International Symposium on Advances in Production and Fabrication of Light Metals and Metal Matrix Composites, Alberta, August 23-27, 1992**
- **PASTERNAK D. et al: Proceedings of the Second International Conference of the Semi-Solid Processing of Alloys and Composites, Massachusetts Institute of Technology, Cambridge, Massachusetts, 1993**
- **PASTERNAK D. et al: Proceedings of the Second International Conference of Semi-Solid Processing of Alloys and Composites, Massachusetts Institute of Technology, June 10-12, 1992**

Description

[0001] The present invention relates to the injection moulding or casting of thixotropic alloys. As used herein, the terms "composite" or "alloy composite" include an alloy matrix having ceramic reinforcement, and includes metal matrix composites.

[0002] The semi-solid processing of alloys and composites is an area of technology in which much interest is presently being shown. Such processing generally requires the formation of a thixotropic alloy which is subsequently processed. Thixotropic alloys are produced when solid particles of a metal or alloy are homogeneously suspended in a liquid phase of molten metal. The semi-solid mass thus produced has thixotropic rheology.

[0003] Thixotropic alloys may be processed to produce metal articles by injection moulding.

[0004] A number of processes to produce thixotropic alloys have been proposed. United States Patent Nos. 4,694,881 and 4,694,882 both assigned to the Dow Chemical Corp., describe processes for producing thixotropic alloys which comprise feeding solid particles of a metal alloy from a hopper into an extruder, such as a screw extruder. In U.S. 4,694,881, the solid particles are heated in the extruder to a temperature above the liquidus temperature of the alloy. The molten mass thus obtained is subsequently cooled to a temperature between the solidus and liquidus temperatures and subjected to shearing to break the dendritic structure that would otherwise form. The resulting liquid-solid composition of a thixotropic alloy is injected into a mould to form a moulded product.

[0005] United States 4,694,882 describes a similar process, except that the feed alloy particles are heated to a temperature between the solidus and liquidus temperatures, without complete melting of the feed metal particles taking place.

[0006] Both of the above processes utilise feed particles or chips of a convenient size for handling. The patents especially describe the use of chips having an irregular shape. The size of the particles used is described as not being critical to the invention, although relatively small particle sizes are preferred because of heat transfer and handling requirements.

[0007] DE-A-1758656 describes particulate alloy compositions for use in impact extruding. The compositions comprise aluminium or magnesium alloys in particulate form wherein at least about 1/3 of the particles have a ratio of the longest to shortest dimension of at least about 2. Preferably, the particles are flat chips or platelets having a high aspect ratio.

[0008] Experiments carried out by the present applicant have shown that the particles used in the process described in U.S. Patents Nos. 4,694,881 and 4,694,882 are prone to block the hopper and seize the screw extruder. Further, the particles do not exhibit good packing characteristics which can cause difficulty in achieving sufficient heat transfer rates to cause the partial melting of the metal particles and also render control over the temperature more difficult.

D.Ghosh et al. in "Advances in Production and Fabrication of Light Metals and Metal Matrix Composites: Proceedings of the International Symposium on Advances in Production and Fabrication of Light Metals and Metal Matrix Composites, Edmonton, Alberta August 23-27, 1992" M.M. Avedesian et al. eds., Metallurgical Society (Canada), pages 399-411 describe thixotropic injection molding of machine produced magnesium alloy chips, nominally 2 mm in size (-7 to +18 mesh screen product).

L. Pasternate et al. in "Proceedings of the Second International Conference on the Semi-Solid Processing of Alloys and Composites: Massachusetts Institute of Technology, Cambridge, Massachusetts, June 10-12, 1992", Stuart B. Brown and Merton C. Flemings eds., Minerals Metals and Materials Society, pages 159-169, describe thixotropic injection molding of magnesium alloy, wherein the feed material used is a machined chip with a geometry approximately 1 mm square by 2-3 mm in length.

[0009] The present inventors have now developed particles of metal alloys and composites that are particularly suitable for use in producing thixotropic alloys and in the injection moulding of such alloys.

[0010] According to a first aspect, the present invention provides a method for producing a thixotropic alloy comprising: providing a particulate material comprising particles of a metal alloy or composite, wherein said particulate material has a tap density of at least 50% of the theoretical density, and wherein a portion of said particles is shaped such that each of said particles in said portion has a ratio of the length of its largest dimension to its effective diameter in the range of 1.2 to 4.0 and has a largest dimension in the range of 0.5 to 5 mm, and wherein said portion of said particles comprise at least 40% by weight of said particulate material; heating the particulate material and shearing the particulate material, thereby producing a substantially homogenous mixture of solid particles and liquid.

[0011] Preferably, the particles in said portion of the particulate material are shaped such that the ratio of the length of the largest dimension of a particle to the effective diameter of the particle is in the range of 1.2 to 3.0, more preferably 1.2 to 2.0. As used hereinafter, the ratio of the length of the largest dimension of a particle to the effective diameter of the particle will be denoted by the term "aspect ratio".

[0012] The effective diameter of a particle may be determined by determining the smallest circle that the particle will be able to pass through. The diameter of this circle is the effective diameter of the particle.

[0013] Preferably, the particles in said portion have a largest dimension in the range of 1 to 3 mm.

[0014] The particles are shaped such that the tap density of the mass of particles is at least 50% of the theoretical density of the alloy or composite.

[0015] The particles preferably have a substantially smooth surface texture.

[0016] The particles in said portion comprise at least 40% by weight of the mass of particles, preferably at least 60% by weight, more preferably at least 80% by weight, most preferably at least 95% by weight of the mass of particles.

[0017] In one embodiment, the particles preferably have an approximately ovoid shape. Such particles may also be described as having a shape similar to a rugby football or as being the shape formed by the solid of revolution of an ellipse or generally elliptical shape about a longitudinal axis.

[0018] In another embodiment, the particles may have a generally tear drop shaped profile or have a profile that may be described as a flattened tear drop. In this embodiment, in a longitudinal cross-section of a particle, a first end of the particle will have a generally hemispherical or hemi-ovoidal shaped portion. The generally hemispherical or hemi-ovoidal shaped portion may be flattened, usually at a leading edge thereof. This portion will taper to a second end of the particle, where the particle will terminate at a point or at a portion having a small radius of curvature. The overall shape of the particle may be considered to be formed generally as the solid of revolution of the planar shape of the cross-section profile. Although the particle should have a substantially smooth surface texture, it will be appreciated that the particles will have a small degree of surface roughness (as will the football shaped particles).

[0019] The thixotropic condition may be produced by any suitable process that involves heating and shearing the particles. However, it is particularly preferred that the thixotropic condition is produced by use of a screw extruder apparatus. In this case, the feed particles may be supplied to a screw extruder whereupon they enter a first heating zone and are heated to a temperature above the melting point of the alloy or composite. The molten material may then pass to a second zone where the molten metal is cooled to a temperature below the liquidus temperature and above the solidus temperature. Solidification of some of the material will occur to form a mixture of solid particles and liquid. The screw of the extruder is caused to rotate such that the mixture is sheared to prevent the formation of large crystal structures and a thixotropic material is formed.

[0020] Alternatively, the feed particles may be heated in a first zone of the screw extruder to a temperature above the solidus temperature of the material but below the liquidus temperature of the material. Shear is applied to the resulting mixture of liquid and solid particles by rotation of the screw of the extruder to produce the thixotropic material. It will be appreciated that the method of the present invention is not restricted to use of a screw extruder, but that any means that is capable of heating the feed particles to the required temperature and supplying a shearing force to the mixture of liquid metal and solid particles may be used. For example, the mixture may be subjected to the action of a rotating plate or it may be forced to travel through a tortuous path extruder in order to impart sufficient shearing force to the mixture to produce the thixotropic material. As a further alternative, electromagnetic stirring may be used to obtain the thixotropic material.

[0021] The feed particles may be supplied from a hopper by gravity feed or conveyor feed.

[0022] The thixotropic material formed by the method of the invention is especially suitable for use in the production of metal components by injection moulding.

[0023] Accordingly, the present invention also provides a method for producing an article, comprising: heating and shearing a particulate material according to the present invention thereby producing a substantially homogenous mixture of solid particles and liquid; injecting the mixture into a mould; allowing the mixture to at least partially solidify in the mould; and removing the article from the mould. The particles used in the method of the present invention may be of any required metal alloy or composite thereof. Some suitable materials include metal and intermetallic alloys based on lead, aluminium, zinc, magnesium, copper and iron. The preferred particles are alloys of aluminium.

[0024] The invention will now be further described with reference to the Figures in which:

Figure 1 shows a schematic profile view of "football" shaped particles for use in the method of the invention;

Figure 2 shows a scanning electron micrograph of the actual particles shown schematically in Figure 1;

Figure 3 shows a schematic cross-section view of another particle for use in the method of the invention;

Figure 4 shows a similar view to Figure 3 showing the calculation of aspect ratio for such particles;

Figures 5 and 6 show scanning electron micrographs of further particles for use in the method of the present invention;

Figure 7 shows a percentage frequency distribution of aspect ratio for granule type 1;

Figure 8 shows a percentage frequency distribution of the dimension "length" for granule type 1;

Figure 9 shows a percentage frequency distribution of the dimension "width" for granule type 1;

- Figure 10 shows a percentage frequency distribution of aspect ratio for granule type 2;
- Figure 11 shows a percentage frequency distribution of the dimension "length" for granule type 2;
- 5 Figure 12 shows a percentage frequency distribution of the dimension "width" for granule type 2;
- Figure 13 shows a scanning electron micrograph of particles according to the invention which have a more needle-like structure;
- 10 Figure 14 shows photomicrographs of a slurry produced in crucible tests at 575°C using granule type 1;
- Figure 15 shows photomicrographs of a slurry produced in crucible tests at 590°C using granule type 1;
- Figure 16 shows photomicrographs of a slurry produced in a crucible test at 575° using granule type 2; and
- 15 Figure 17 shows photomicrographs of a slurry produced in a crucible test at 590°C using granule type 2.

[0025] In a preferred embodiment, a substantial proportion of the particles of the particulate material used in the method of the present invention have an approximately ovoid particle shape with a ratio of the largest dimension to the effective diameter of between 1.2 and 3.0, more preferably 1.2 to 2.0. This ratio may be designated the aspect ratio of the particles. These particles can be further characterised as being in the shape of an elongated sphere or shaped like a rugby ball. A preferred shape of the particles is shown schematically in Figure 1. The aspect ratio for the particles is determined from the ratio of length to effective diameter for the particles. Thus, referring to Figure 1, the invention requires that:

25 $L/D = 1.2 \text{ to } 4.0$, preferably $1.2 - 3.0$, more preferably $1.2 - 2.0$

[0026] The dimension L lies within the range of 0.5 to 5mm.

[0027] Figure 2 shows a scanning electron micrograph of actual particles that are generally ovoid shape. The particles may also be described as of generally cylindrical shape and having rounded ends.

30 [0028] In a further embodiment, the particles have a generally tear drop shape that may be flattened at one end. With reference to Figure 3, which shows a cross-sectional view of a particle, particle 20 of generally flattened tear drop shape has a first end 21 that is in the form of a generally hemispherical or hemi-ovoidal shape. First end 21 may be flattened at leading edge 22. Particle 20 is shaped such that first end 21 tapers towards second end 23. Second end 23 terminates at a point or at a portion 24 having a small curvature of radius.

35 [0029] Figure 3 shows a cross-sectional view of particle 20. The overall shape of the particle may be considered to be in the form of a solid of revolution of the cross-section about longitudinal axis 25.

[0030] Referring to Figure 4, the aspect ratio of particle 20 falls within the range of 1.0 to 4.0, preferably 1.2 to 3.0, more preferably 1.2 to 2.0. As with the football shaped particles, the aspect ratio of particle 20 is given by the ratio UD . Here, dimension L may be considered to be the maximum height of the particle. Dimension D is the diameter of the smallest circle that the particle is able to pass through.

40 [0031] Scanning electron micrographs of further particles that fall within the scope of the present invention are shown in Figures 5 and 6.

[0032] The particulate matter used in the method of the present invention should include a substantial proportion of particles shaped according to the embodiments described above. In producing the particulate matter of the invention, it has been found that a substantial proportion of irregularly shaped particles are also formed and become included in the particulate matter. The presence of such irregularly shaped particles does not unduly affect the properties of the particulate matter unless the irregularly shaped particles are present in an unacceptably large amount.

45 [0033] When used in the methods of the present invention for producing a thixotropic material or a metallic article by the injection moulding of a metal alloy or composite, at least 40% by weight of the mass of feed particles are sized such that the overall length of the particles is in the range of 0.5 to 5 mm, preferably 1 to 3 mm. This allows convenient handling of the particles whilst also avoiding binding or clogging of the screw, in the case where a screw extruder is used.

50 [0034] The particulate material used in the method of the present invention has a combination of properties that is not found in any metallic particulates currently known to the applicants and these combination of properties make the particulates especially suitable for use as feedstock in thixomolding processes. The particulate material of the invention has a tap density that is at least 50% of the theoretical density. This ensures good particle to particle contact and allows adequate heat transfer rates to be achieved in the heating zone. This allows for relatively short heating times to be used to cause the initial melting or partial melting of the particles and it also allows for close control over temperature to be maintained to enable the thixotropic state to be maintained. The particulate material is relatively free flowing and will be unlikely to block a feed hopper. The mixing torque required to turn the screw when the particulate material fills a screw

extruder is not unacceptably high and the particles are sufficiently large to ensure that particles cannot slip between the walls of the extruder and the screw to cause binding of the screw.

[0035] The properties of a group of particulate materials were determined in order to compare them with the properties of the mass of particles used in the method of the present invention. The particles used for comparison purposes were made of aluminium and consisted of powder (100 μ m), needles, granules and irregular shaped machining chips. Although some of these particles showed properties in one category that were superior to the properties of the particles of the invention in that category, none of the comparative particles had a combination of properties that were as desirable or useful as the properties of the particulate matter of the invention.

[0036] The particulate material used in the method of the present invention may be mixed with particles of other shapes and sizes. However, this is generally not preferred due to possible problems associated with segregation and settling of the resultant mixture.

[0037] In order to quantify the performance of particulate matter used in the method of the invention, a series of comparative tests were run to compare the properties of the "football" particles with a series of commercially available particles. The particles used for comparison purposes were aluminium granules, aluminium needles, aluminium spherical powder (100 μ m average particle size) and aluminium machinery chips. These particles were tested for particle size, particle shape, apparent density, tap density, flow rate through a standard funnel, mixing torque and angle of repose. The data obtained is shown in Table 1.

[0038] Using three characterisation tests of flow time, tap density and mixing torque, the particles were ranked according to performance (a ranking of "1" signifies the best performance). The rankings are shown in Table 2.

TABLE 1

Particles	Particle Size		Particle Shape	Apparent Density		Tap Density		Mixing Torque (in - lbs)	Angle of Repose (°)
	Average Length (mm)	Average Width (mm)		g/cc	% of theoretical	g/cc	% of theoretical		
granules			Irregular	0.54	20.0	0.63	23.2	7.20	34
needles	4.29	0.62	needles	1.08	40.0	1.39	51.5	19.20	32
machining chips			irregular	0.20	7.4	0.23	8.4		40
machining chips (tumbled)			irregular	0.19	7.0	0.24	8.8		35
machining chips (milled-light)			irregular	0.19	7.0	0.22	8.1		35
machining chips (milled-heavy)			irregular	0.24	8.7	0.30	11.2		43
granules			irregular	0.54	20.1	0.60	22.0	22.80	30
spherical powder	0.10		spherical	1.39	51.5	1.61	60.0		24
particulate matter of the invention	1.63			1.49	55.3	1.56	57.9	15.20	22

TABLE 2

Ranking of particulates using key parameters			
Particulate	Rank		
	Flow Time	Tap Density	Mixing Torque
Needles	3	3	3
Granules	5	4	1
Granules	4	5	4
Spherical Powder	1	1	-
Particulate Matter of the Invention	2	2	2

[0039] At first glance, it appears that the spherical powder provides the best performance in two of the three categories. However, the powder seized between the screw and the wall of the torque measuring device and it is likely that this will also occur in thixomolding apparatus. Accordingly, the spherical powder is unsuitable as a feedstock for thixomolding.

[0040] Once the spherical powder has been eliminated as a potential feedstock, it is apparent that the particulate matter of the present invention is the most suitable for use as a feedstock for thixomolding processes.

[0041] In order to demonstrate the advantages of the present invention, a number of particles were prepared and compared with particles that are not encompassed by the present invention.

[0042] The particles that fall within the scope of the present invention have been denoted as "granule type #1" and "granule type #2". The summary of the granule dimensions is given in Table 3.

TABLE 3

Summary of Granule Dimensions					
Granule Type (sample number)	Length (mm)		Width (mm)		Aspect Ratio
	Average	Std. Dev.	Average	Std. Dev.	
Type #1 (158)	3.55	1.39	2.46	0.74	1.41
Type #2 (189)	3.99	1.35	2.90	0.78	1.36

[0043] Particle size analysis of granule type #1 and granule type #2 was carried out and the results of this particle size analysis, given as percentage frequency distribution of aspect ratio, percentage frequency distribution of the dimension "length" and percentage frequency distribution of the dimension "width" (diameter), for granule type #1 and granule type #2, are shown in Figures 5 to 10. The granules were produced from an A1 7% Si alloy.

[0044] Granule types #1 and #2 were found to be free flowing as no mixing torque could be measured. In addition, the granules transported easily along the barrel of the torque measuring device. The granules were found to have an apparent density of from 56-58% of the theoretical apparent density and a tap density of 69% of the theoretical tap density.

[0045] For comparative purposes, samples of particles comprising mainly needles were obtained. All of the needles caused seizing of the screw during moulding screw simulation. The apparent density of the needles ranged from 39 to 45% of the theoretical value and the top density ranged from 50 to 59% of the theoretical value. The needles were of a similar aluminium alloy as the granule types #1 and #2.

[0046] Several experiments with an A1 7% Si alloy were also carried out in which the granule types #1 and #2 and the needles were used to make a slurry of solid metal with liquid metal. These trials simulated the formation of a thixotropic alloy. The slurry was produced in a stirred silicon carbide crucible. The stirrer had two flights of blades. The procedure involved preheating a sufficient amount of particles to 400°C. The furnace temperature was set at 590°C, which is between the solidus and liquidus temperatures for the aluminium alloy used in the particles. The pre-heated particles were charged into the crucible such that the second flight of the stirrer made contact with the particles during stirring, although the particles did not cover the second flight of blades at this stage. The stirring speed was set at 100 rpm.

[0047] Aluminium alloys are expected to be a difficult feedstock for thixomolding processes because at about 400°C, aluminium-containing particles stick to each other. This particle adhesion would tend to produce blockages in the feed screw of a thixomolding apparatus.

[0048] The crucible tests to simulate the formation of a thixotropic alloy showed that granule types #1 and #2 both produced a slurry without any difficulties. Observations of the method were as follows:

- on initial and subsequent furnace charges, no evidence of granule adhesion (i.e., binding together was not apparent
- after stirring for approximately 30-40 minutes the onset of granule melting was obvious with the formation of large, solid lumps of material
- a decrease in the stirring efficiency was noticed as material continuously built-up around the crucible wall.
- to increase stirring efficiency, stirring was periodically stopped to allow material removal from the crucible wall. In addition, if material build-up was rapidly re-established, a granule addition was then carried out to facilitate build-up removal and good mixing
- granule additions were also necessary due to a reduction of material volume during melting.

[0049] With regard to the needles, some problems were encountered in producing a slurry using needles. These include:

- evidence of needles binding together due to the 400°C preheating stage. This observation was made during the initial and subsequent charges associated with the trial
- the binding together of the needles was accentuated when the needles came in contact with the hot walls of the crucible. On mixing, large lumps formed immediately causing the motor to labour. (Note: stirring was stopped for 15 minutes and the furnace temperature increased to allow material "softening".
- once the lumps had broken down, there were no problems with mixing the material, except for material build-up around the crucible wall.

[0050] In addition to the above difficulties, it is also noted that the needles would tend to seize the screw of the thixomolding apparatus during feeding.

[0051] A mass of more needle-like particles, a scanning electron micrograph of which is shown in Figure 13, were also subjected to a crucible test. These particles, which had an average length of 2.8 mm and an average width of 0.8 mm (aspect ratio of 3.4) fall within the scope of the present invention. Although the difficulties mentioned above in respect of needles were present to some degree, the particles of Figure 13 were able to form useful slurries and hence would be an acceptable feedstock for thixomolding. Seizing of the screw is likely to be less of a problem with the particles of Figure 13 than with long, thin needles having aspect ratios above 4.

[0052] The slurries obtained using granule types #1 and 412 were allowed to solidify and photomicrographs were subsequently taken. Figures 11 and 12 show photomicrographs of the slurries obtained using granule types 1 at 575°C and 590°C respectively. Figures 13 and 14 show similar photomicrographs for granule types 2. The slurries were obtained by heating the granules up from room temperature to a temperature between the solidus and liquidus of the alloy. The photomicrographs clearly show solid particles surrounded by regions of solidified liquid. A fair amount of porosity is also present, which is due to the stirring arrangement used in the crucible experiments. The porosity is not expected to be present when a thixomolding apparatus is used.

Claims

1. A method for producing a thixotropic alloy comprising: providing a particulate material comprising particles of a metal alloy or composite, wherein said particulate material has a tap density of at least 50% of the theoretical density and wherein a portion of said particles is shaped such that each of said particles in said portion has a ratio of the length of its largest dimension to its effective diameter in the range of 1.2 to 4.0 and has a largest dimension in the range of 0.5 to 5mm, and wherein said portion of said particles comprise at least 40% by weight of said particulate material; heating the particulate material and shearing the particulate material, thereby producing a substantially homogenous mixture of solid particles and liquid.
2. A method as claimed in claim 1 wherein each of said particles in said portion has a ratio of the length of its largest dimension to its effective diameter in the range of 1.2 to 3.0.
3. A method as claimed in claim 1 wherein each of said particles in said portion has a ratio of the length of its largest dimension to its effective diameter in the range of 1.2 to 2.0.

4. A method as claimed in any one of the preceding claims wherein each of said particles in said portion has a largest dimension in the range of 1 to 3mm.
- 5 5. A method as claimed in any one of the preceding claims wherein some of said particles in said portion have an approximately ovoid shape.
6. A method as claimed in any one of the preceding claims, wherein said particles in said portion have a generally tear drop shaped profile or a generally flattened tear drop shaped profile.
- 10 7. A method as claimed in any one of the preceding claims wherein said particles have a substantially smooth surface texture.
8. A method as claimed in any one of the preceding claims wherein said particles comprise an aluminum alloy or an aluminum composite.
- 15 9. A method as claimed in any preceding claim wherein the thixotropic alloy is produced using a screw extruder apparatus.
- 20 10. A method as claimed in claim 9, wherein said heating step comprises heating the particulate material in a first zone to a temperature above the melting point of the particulate material, thereby forming a molten material, and cooling the molten material in a second zone to a temperature below the liquidus temperature but above the solidus temperature of the particulate material, and said shearing step comprises rotating the screw extruder apparatus in the second zone, thereby preventing formation of large crystal structures in the molten material.
- 25 11. A method as claimed in claim 9, wherein said heating step comprises heating the particulate material to a temperature above the solidus temperature but below the liquidus temperature of the particulate material to form a mixture, and said shearing step comprises rotating the screw extruder apparatus, thereby preventing formation of large crystal structures in the molten material.
- 30 12. A method as claimed in any of claims 1 to 8, wherein the thixotropic alloy is produced using any of a rotating plate, a tortuous path extruder and an electromagnetic stirrer.
- 35 13. A method for producing an article, comprising heating and shearing a particulate material by a method according to any one of claims 1 to 12 to produce a substantially homogeneous mixture of solid particles and liquid, injecting the mixture into a mould, allowing the mixture to at least partially solidify in the mould and removing the article from the mould.

Patentansprüche

- 40 1. Verfahren zur Herstellung einer thixotropen Legierung, das folgendes umfasst: Bereitstellen eines teilchenförmigen Materials, das Teilchen einer Metalllegierung oder eines Verbundmetalls umfasst, wobei das teilchenförmige Material eine Klopfdichte von mindestens 50% der theoretischen Dichte aufweist und wobei ein Anteil der Teilchen so geformt ist, dass jedes der Teilchen in dem Anteil ein Verhältnis der Länge seiner größten Dimension zu seinem effektiven Durchmesser im Bereich von 1,2 bis 4,0 aufweist und eine größte Dimension im Bereich von 0,5 bis 5 mm aufweist; und wobei der Anteil der Teilchen mindestens 40 Gew.-% des teilchenförmigen Materials umfasst; Erhitzen des teilchenförmigen Materials und Scheren des teilchenförmigen Materials und **dadurch** Herstellen eines im Wesentlichen homogenen Gemisches von festen Teilchen und Flüssigkeit.
- 45 2. Verfahren nach Anspruch 1, wobei jedes der Teilchen in dem Anteil ein Verhältnis der Länge seiner größten Dimension zu seinem effektiven Durchmesser im Bereich von 1,2 bis 3,0 aufweist.
3. Verfahren nach Anspruch 1, wobei jedes der Teilchen in dem Anteil ein Verhältnis der Länge seiner größten Dimension zu seinem effektiven Durchmesser im Bereich von 1,2 bis 2,0 aufweist.
- 50 4. Verfahren nach einem der vorhergehenden Ansprüche, wobei jedes der Teilchen in dem Anteil eine größte Dimension im Bereich von 1 bis 3 mm aufweist.
- 55

5. Verfahren nach einem der vorhergehenden Ansprüche, wobei einige der Teilchen in dem Anteil eine annähernd eiförmige Form aufweisen.
- 5 6. Verfahren nach einem der vorhergehenden Ansprüche, wobei die Teilchen in dem Anteil ein im Allgemeinen tropfenförmiges Profil oder ein im Allgemeinen abgeftachtes tropfenförmiges Profil aufweisen.
7. Verfahren nach einem der vorhergehenden Ansprüche, wobei die Teilchen eine im Wesentlichen glatte Oberflächentextur aufweisen.
- 10 8. Verfahren nach einem der vorhergehenden Ansprüche, wobei die Teilchen eine Aluminiumlegierung oder ein Verbundaluminium umfassen.
9. Verfahren nach einem der vorhergehenden Ansprüche, wobei die thixotrope Legierung unter Verwendung eines Schneckenextrudergeräts hergestellt wird.
- 15 10. Verfahren nach Anspruch 9, wobei der Aufheizschritt das Erhitzen des teilchenförmigen Materials in einer ersten Zone auf eine Temperatur oberhalb des Schmelzpunkts des teilchenförmigen Materials, und **dadurch** Bilden eines geschmolzenen Materials, und das Abkühlen des geschmolzenen Materials in einer zweiten Zone auf eine Temperatur unterhalb der Liquidustemperatur, jedoch oberhalb der Solidustemperatur des teilchenförmigen Materials umfasst, und wobei der Scherschnitt die Rotation des Schneckenextrudergeräts in der zweiten Zone und **dadurch** Verhindern der Bildung von großen Kristallstrukturen in dem geschmolzenen Material umfasst.
- 20 11. Verfahren nach Anspruch 9, wobei der Aufheizschritt das Erhitzen des teilchenförmigen Materials auf eine Temperatur oberhalb der Solidustemperatur, jedoch unterhalb der Liquidustemperatur des teilchenförmigen Materials unter Bildung eines Gemisches umfasst, und wobei der Scherschnitt die Rotation des Schneckenextrudergeräts und **dadurch** Verhindern der Bildung von großen Kristallstrukturen in dem geschmolzenen Material umfasst.
- 25 12. Verfahren nach einem der Ansprüche 1 bis 8, wobei die thixotrope Legierung unter Verwendung einer rotierenden Platte, eines Extruders mit gekrümmten Wegen und eines elektromagnetischen Rührers hergestellt wird.
- 30 13. Verfahren zur Herstellung eines Gegenstands, das das Aufheizen und Scheren eines teilchenförmigen Materials durch ein Verfahren nach einem der Ansprüche 1 bis 12 unter Herstellung eines im Wesentlichen homogenen Gemisches von festen Teilchen und Flüssigkeit, das Injizieren des Gemisches in eine Form, das zumindest teilweise sich Verfestigenlassen des Gemisches in der Form und das Entnehmen des Gegenstands auf der Form umfasst.
- 35

Revendications

- 40 1. Procédé pour produire un alliage thixotrope, comprenant : la fourniture d'un matériau particulaire comprenant des particules d'un alliage ou composite métallique, dans lequel ledit matériau particulaire a une masse volumique après tassement d'au moins 50 % de la masse volumique théorique, et dans lequel une partie desdites particules est conformée de sorte que chacune desdites particules dans ladite partie a un rapport de la longueur de sa plus grande dimension à son diamètre effectif dans la gamme de 1,2 à 4,0 et a la plus grande dimension dans la gamme de 0,5 à 5 mm, et dans lequel ladite partie desdites particules représente au moins 40 % en poids dudit matériau particulaire; le chauffage du matériau particulaire et le cisaillement du matériau particulaire, produisant ainsi un mélange de particules solides et de liquide essentiellement homogène.
- 45 2. Procédé selon la revendication 1, dans lequel chacune desdites particules dans ladite partie a un rapport de la longueur de sa plus grande dimension à son diamètre effectif dans la gamme de 1,2 à 3,0.
- 50 3. Procédé selon la revendication 1, dans lequel chacune desdites particules dans ladite partie a un rapport de sa plus grande dimension à son diamètre effectif dans la gamme de 1,2 à 2,0.
- 55 4. Procédé selon l'une quelconque des revendications précédentes, dans lequel chacune desdites particules dans ladite partie a la dimension la plus grande dans la gamme de 1 à 3 mm.
5. Procédé selon l'une quelconque des revendications précédentes, dans lequel certaines desdites particules dans ladite partie ont une forme approximativement ovoïde.

6. Procédé selon l'une quelconque des revendications précédentes, dans lequel lesdites particules dans ladite partie ont un profil général en forme de larme ou un profil général en forme de larme aplatie.
- 5 7. Procédé selon l'une quelconque des revendications précédentes, dans lequel lesdites particules ont une texture de surface substantiellement lisse.
8. Procédé selon l'une quelconque des revendications précédentes, dans lequel lesdites particules comprennent un alliage d'aluminium ou un composite d'aluminium.
- 10 9. Procédé selon l'une quelconque des revendications précédentes, dans lequel l'alliage thixotrope est produit en utilisant une extrudeuse à vis.
- 15 10. Procédé selon la revendication 9, dans lequel ladite étape de chauffage comprend le chauffage du matériau particulaire dans une première zone à une température supérieure au point de fusion du matériau particulaire, formant ainsi un matériau fondu, et le refroidissement du matériau fondu dans une seconde zone à une température inférieure à la température liquidus mais supérieure à la température solidus du matériau particulaire, et ladite étape de cisaillement comprend la rotation de l'extrudeuse à vis dans la seconde zone, ce qui empêche la formation de grandes structures cristallines dans le matériau fondu.
- 20 11. Procédé selon la revendication 9, dans lequel ladite étape de chauffage comprend le chauffage du matériau particulaire à une température supérieure à la température solidus mais inférieure à la température liquidus du matériau particulaire pour former un mélange, et ladite étape de cisaillement comprend la rotation de l'extrudeuse à vis, ce qui empêche la formation de grandes structures cristallines dans le matériau fondu.
- 25 12. Procédé selon l'une quelconque des revendications 1 à 8, dans lequel l'alliage thixotrope est produit en utilisant l'un quelconque parmi une plaque rotative, une extrudeuse à trajet sinueux et un agitateur électromagnétique.
- 30 13. Procédé pour produire un article, comprenant les étapes consistant à chauffer et cisailier un matériau particulaire par un procédé selon l'une quelconque des revendications 1 à 12, pour produire un mélange de particules solides et de liquide essentiellement homogène, à injecter le mélange dans un moule, à laisser le mélange solidifier au moins partiellement dans le moule et à enlever l'article du moule.

35

40

45

50

55

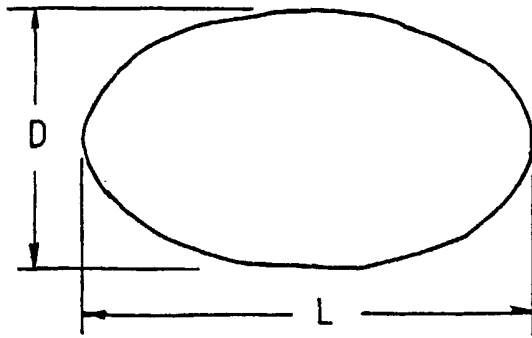


FIG. 1.

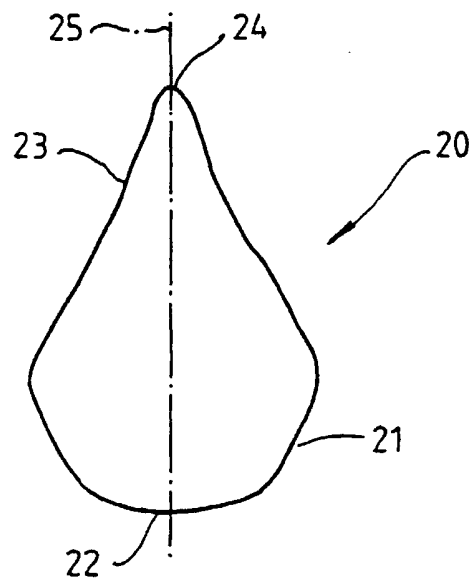


FIG. 3.

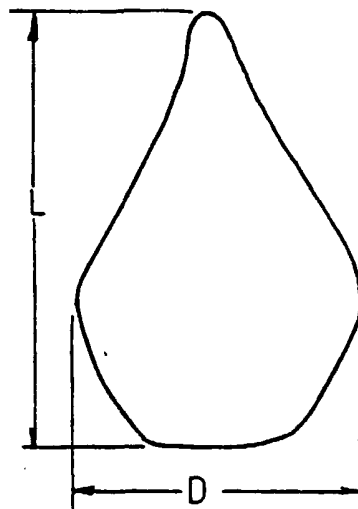
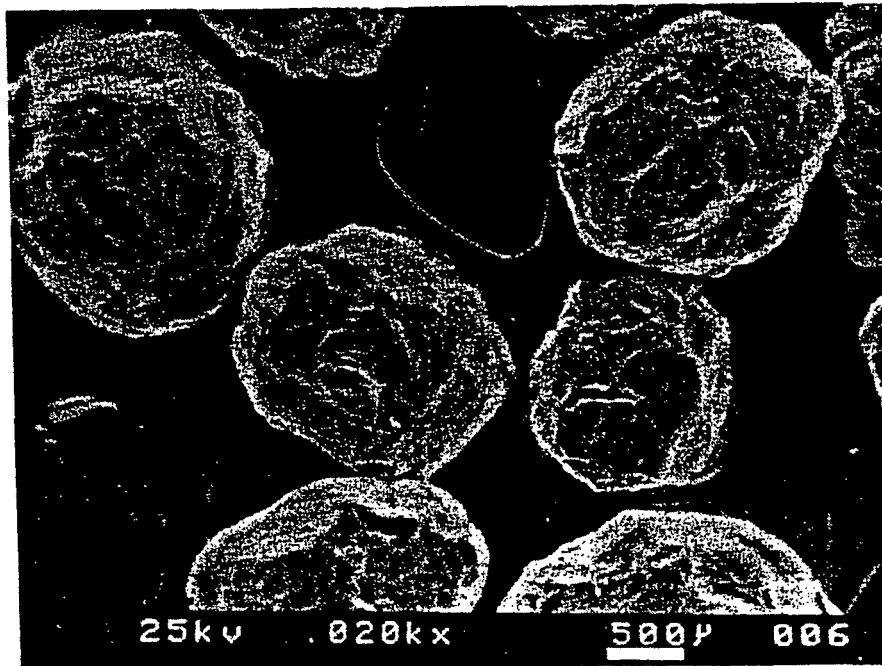
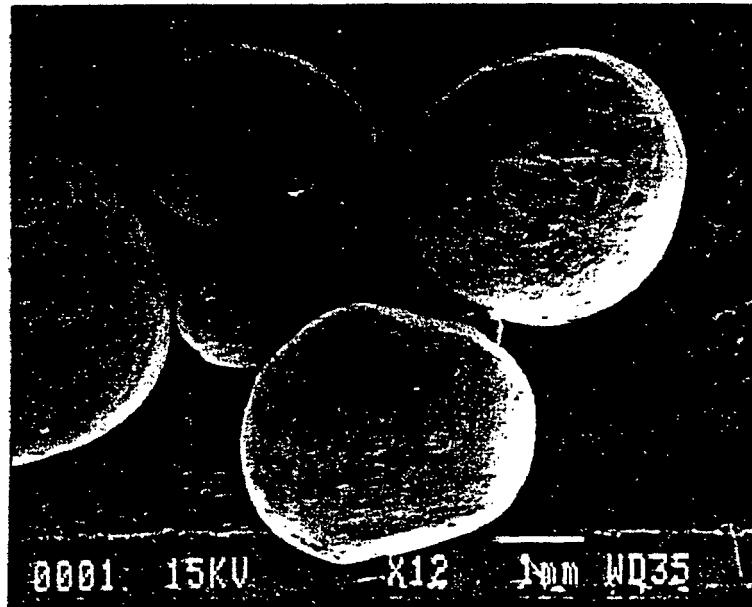


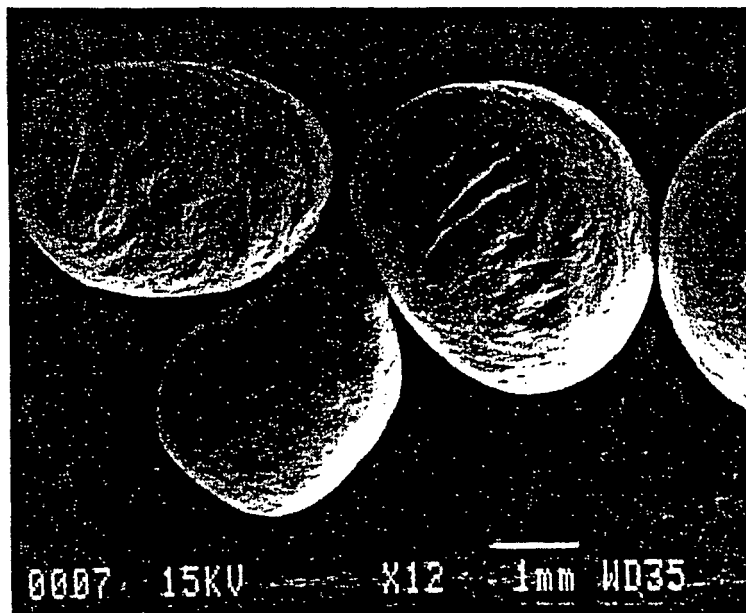
FIG. 4.



III. 2.

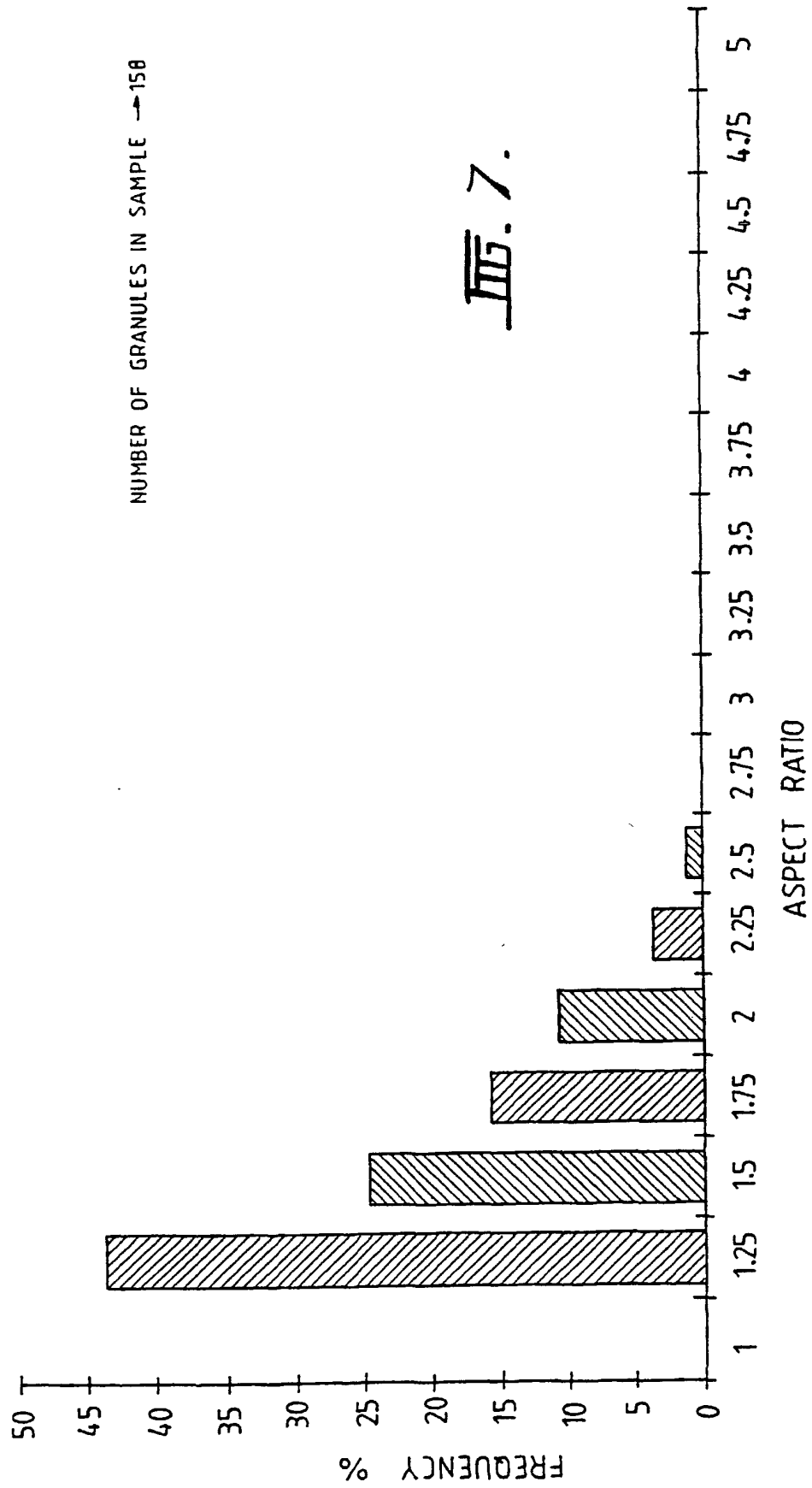


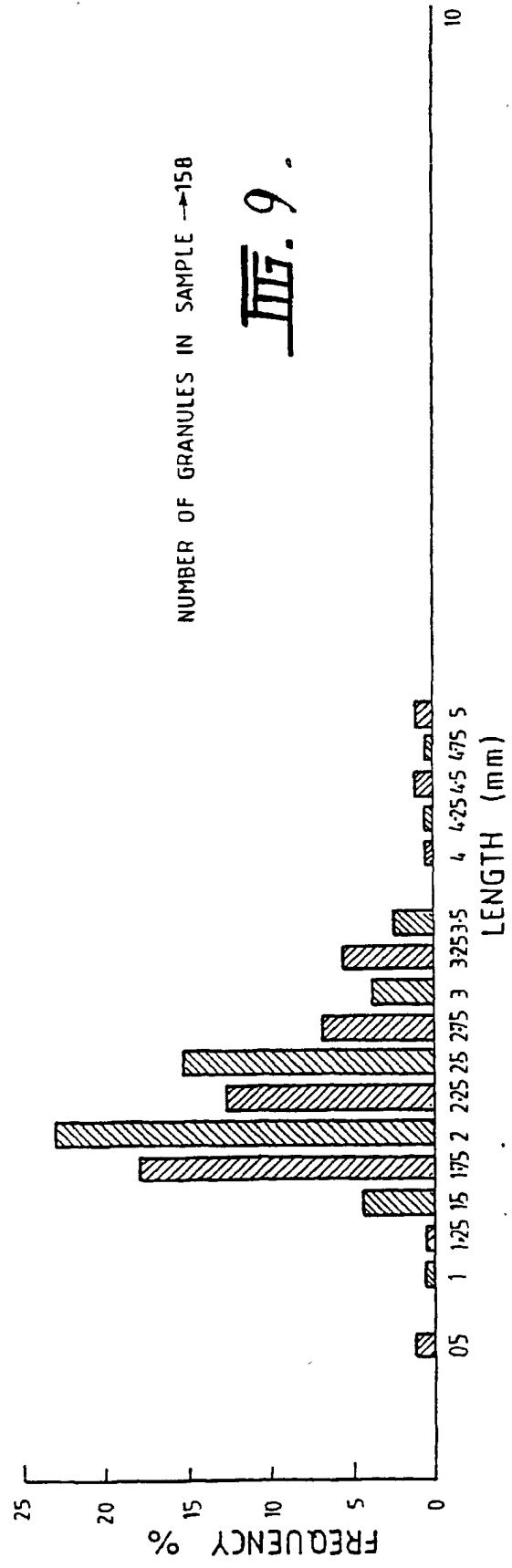
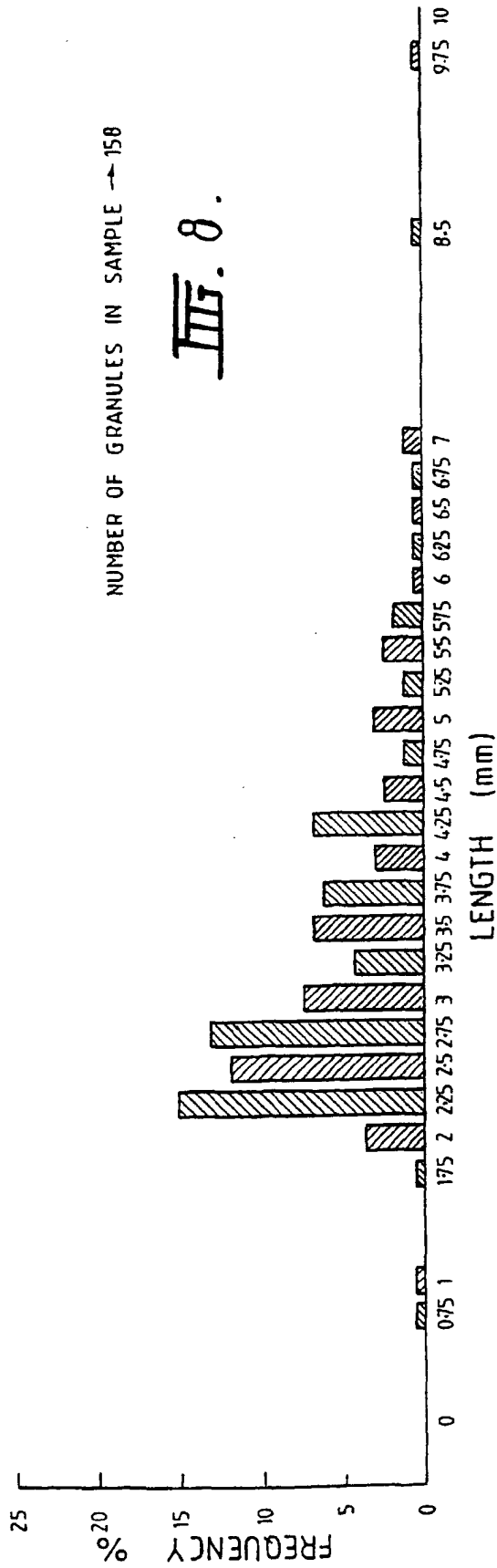
III. 5.



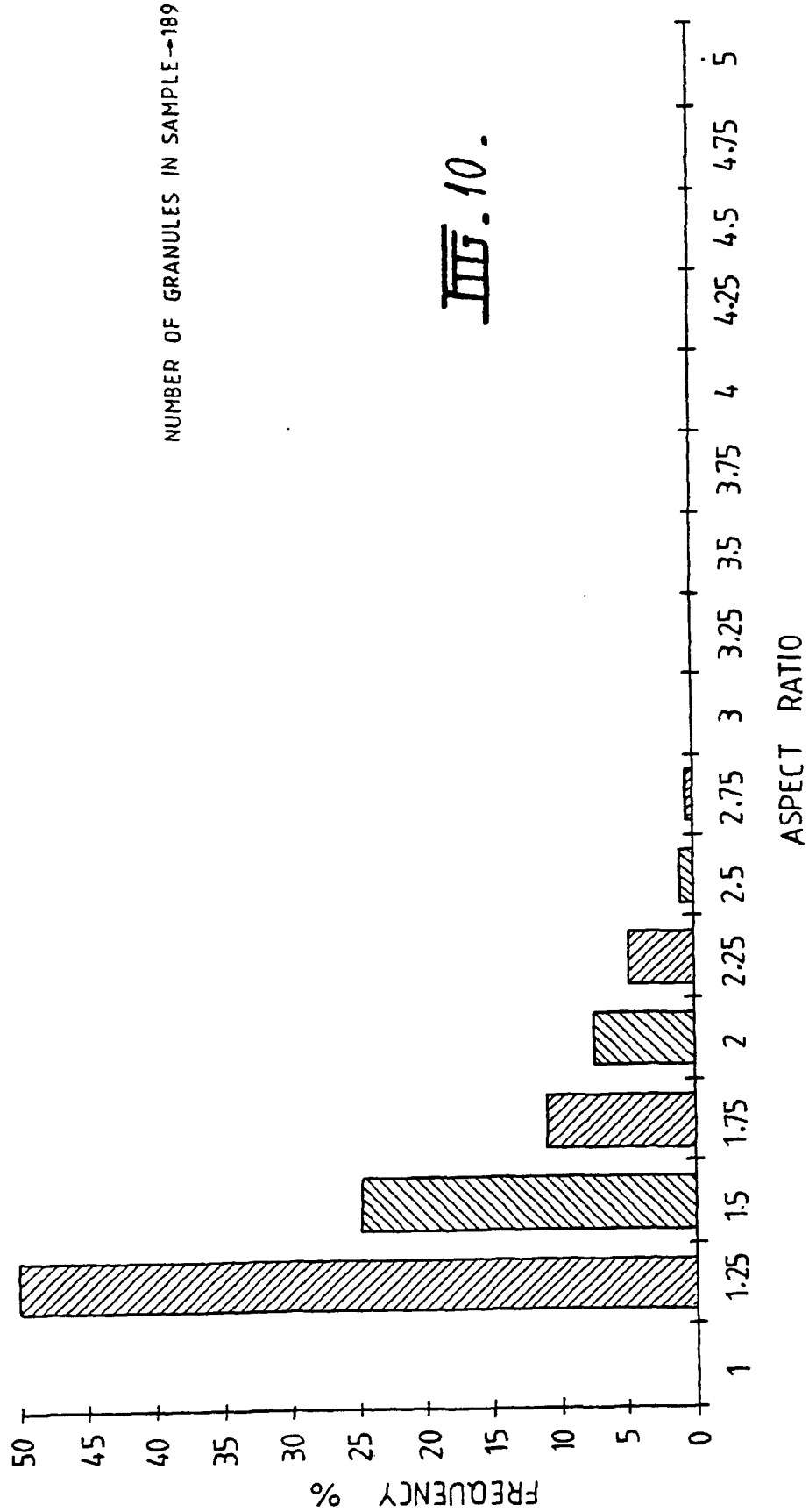
III. 6.

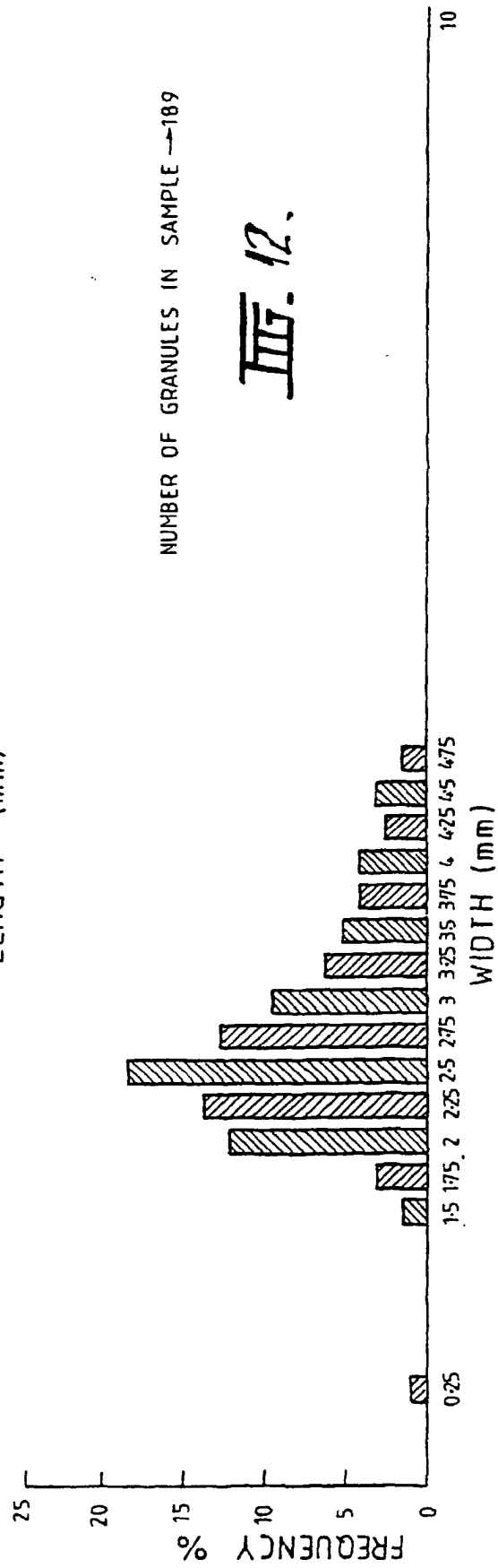
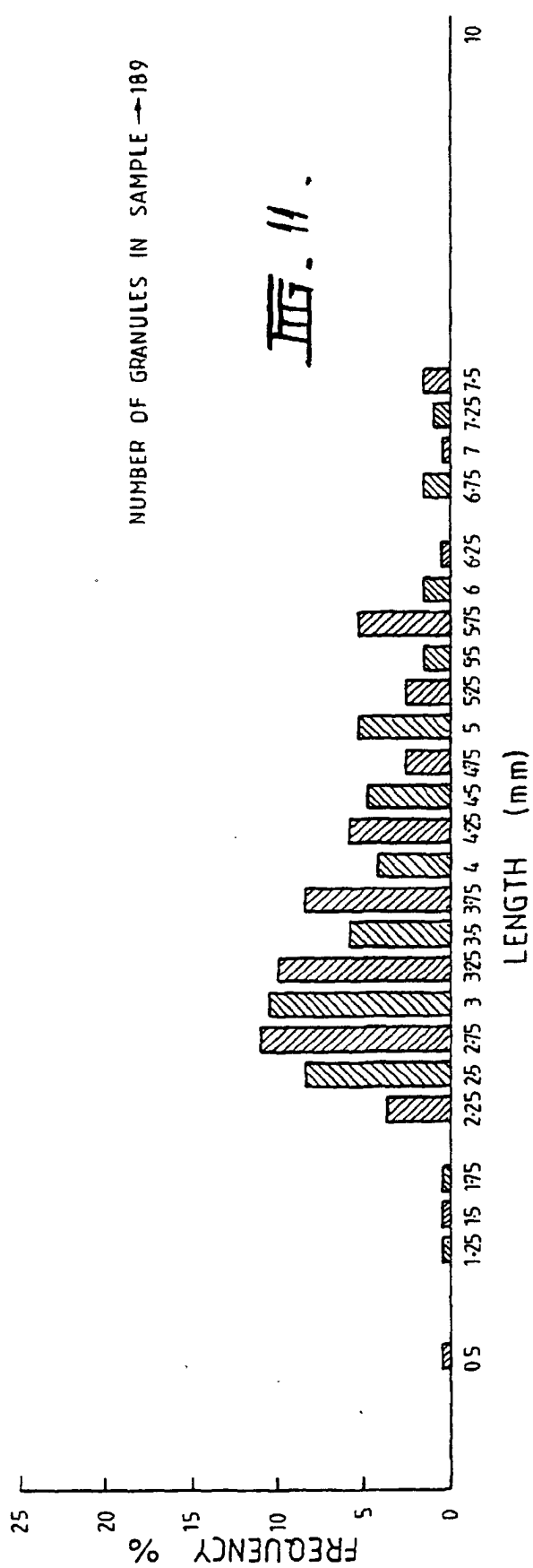
PERCENTAGE FREQUENCY DISTRIBUTION OF ASPECT RATIO FOR GRANULE TYPE #1





PERCENTAGE FREQUENCY DISTRIBUTION OF ASPECT RATIO FOR GRANULE TYPE #2







III. 13.



Mag: 50X



Mag: 100X



Mag: 200X



Mag: 500X

FIG. 14.



Mag: 50X



Mag: 100X

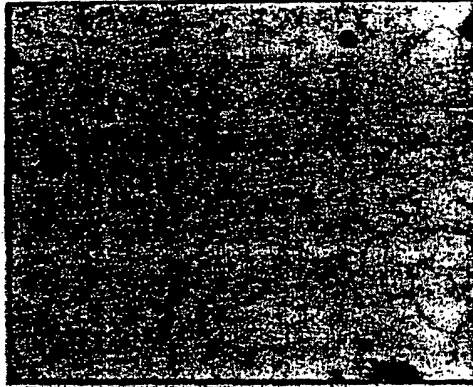


Mag: 200X

III - 15 -



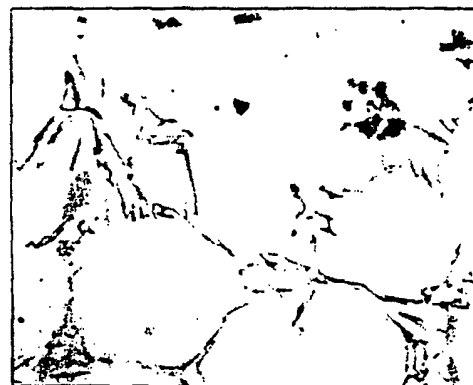
Mag: 500X



Mag: 50X



Mag: 100X

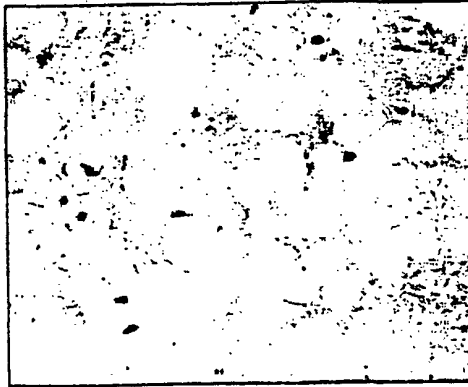


Mag: 200X

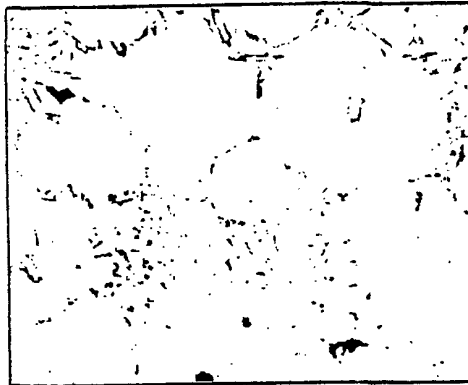
III 16.



Mag: 500X



Mag: 50X

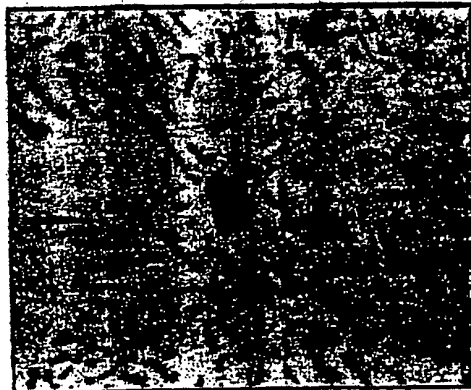


Mag: 100X



Mag: 200X

III 17.



Mag: 500X