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## (11) EP 2 700 460 A1

(12)

## **EUROPEAN PATENT APPLICATION** published in accordance with Art. 153(4) EPC

(43) Date of publication: 26.02.2014 Bulletin 2014/09

(21) Application number: 12774455.5

(22) Date of filing: 20.04.2012

(51) Int Cl.: **B23B 27/14** (2006.01)

(86) International application number: **PCT/JP2012/060651** 

(87) International publication number: WO 2012/144574 (26.10.2012 Gazette 2012/43)

(84) Designated Contracting States:

AL AT BE BG CH CY CZ DE DK EE ES FI FR GB

GR HR HU IE IS IT LI LT LU LV MC MK MT NL NO
PL PT RO RS SE SI SK SM TR

(30) Priority: 20.04.2011 JP 2011093560

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## (54) COATED CUTTING TOOL

(57) An object of the present invention is to provide a coated cutting tool, in which recession of a position of a cutting edge of the tool due to wear or chipping is unlikely to occur, and excellent in wear resistance and chipping resistance.

The coated cutting tool of the present invention comprises a cemented carbide substrate and a film formed on a surface of the substrate by a chemical vapor deposition method, wherein an average film thickness of the film is 3 to 20  $\mu$ m, the film comprises a lower film having

an average film thickness of 2 to 15  $\mu$ m formed on the surface of the substrate, and an upper film having an average film thickness of 1 to 10  $\mu$ m formed on a surface of the lower film on a side opposite to a surface in contact with the substrate, the lower film has an alternately laminated film in which a TiN film having an average film thickness of 10 to 300 nm and a TiCN film having an average film thickness of 0.1 to 0.5  $\mu$ m are alternately laminated, and the upper film has an aluminum oxide film.

## Description

#### **TECHNICAL FIELD**

**[0001]** The present invention relates to a coated cutting tool excellent in chipping resistance and wear resistance, for example, used in continuous cutting of a steel or cast iron, as a matter of course, and excellent in wear resistance even when it is used for milling or interrupted cutting of the above materials.

## **BACKGROUND ART**

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[0002] As a prior art technique of the coated cutting tool, there may be mentioned a cutting tool made of a surface-coated tungsten carbide-based cemented carbide which comprises a hard coating layer constituted by a titanium nitride layer and a titanium carbonitride layer each having a granular crystal structure being alternately laminated two or more layers having an average layer thickness of 3 to 10  $\mu$ m on a surface of the tungsten carbide-based cemented carbide substrate, and at least one layer of the titanium carbonitride layers has a lengthwise grown crystal structure (for example, see Patent Literature 1.).

**[0003]** Also, a cutting tool made of a surface-coated cemented carbide which comprises a WC-based cemented carbide substrate on the surface of which being formed a multi-layered coating layer comprising an alternately laminated layers of TiN and TiCN using a chemical vapor deposition method with a layer thickness of 1 to 5  $\mu$ m may be mentioned as the prior art technique of the coated cutting tool (for example, see Patent Literature 2.).

#### PRIOR ART REFERENCES

**Patent Literatures** 

#### [0004]

[Patent Literature 1] JP H07-136808A [Patent Literature 2] JP H04-033865B

#### SUMMARY OF THE INVENTION

## PROBLEMS TO BE SOLVED BY THE INVENTION

[0005] In recent years, in the field of machining, demand for the quality of a work piece material after processing, in particular, improvement in precision of machining dimension has been increasing. Further, higher hardening of the work piece material has been advanced every year, and in the conventional cutting tools disclosed in the above-mentioned Patent Literature 1 or Patent Literature 2, in the cutting device to which the tool has been attached, recession of the position of the cutting edge of the tool at a relief surface is likely caused by wear or chipping. Therefore, machining dimension of the work piece material tends to go beyond the bounds of the standard scope within a short period of the processing time.

**[0006]** In the processing field, to correct deviation of the machining dimension of a work piece material, correction of the position of the cutting edge of the tool has been frequently done. However, correction of the position of the cutting edge of the tool worsens processing efficiency, so that it has been desired to develop a cutting tool in which recession of the position of the cutting edge of the tool is unlikely caused than before from the processing field.

**[0007]** Thus, an object of the present invention is to provide a coated cutting tool excellent in wear resistance and chipping resistance, in which recession of a position of a cutting edge of the tool due to wear or chipping is unlikely to occur.

## MEANS TO SOLVE THE PROBLEMS

**[0008]** The present inventor has carried out research and development to reduce recession of a position of a cutting edge of the tool by wear or chipping, and as a result, he has found out that a coated cutting tool obtained by forming, on the surface of a cemented carbide substrate, a lower film comprising alternately laminated film which comprises a TiN film having an average film thickness of 10 to 300 nm and a TiCN film having an average film thickness of 0.1 to  $0.5~\mu m$  by the chemical vapor deposition method, and further forming an upper film comprising an aluminum oxide film on the surface of the lower film has excellent wear resistance and chipping resistance.

**[0009]** When the coated cutting tool is used, recession of a position of the cutting edge of the tool by wear or chipping can be suppressed, whereby change in machining dimension of the work piece material can be restrained. Therefore,

a number of an operation for correction of the position of the cutting edge of the tool which has been carried out to restrain change in the above-mentioned machining dimension can be reduced.

Summary of the present invention is as follows.

## [0010]

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- (1) A coated cutting tool comprising a cemented carbide substrate and a film formed on a surface of the substrate by a chemical vapor deposition method, wherein an average film thickness of the film is 3 to 20  $\mu$ m, the film comprises a lower film having an average film thickness of 2 to 15  $\mu$ m formed on the surface of the substrate, and an upper film having an average film thickness of 1 to 10  $\mu$ m formed on a surface of the lower film on a side opposite to a surface in contact with the substrate, the lower film has an alternately laminated film in which a TiN film having an average film thickness of 10 to 300 nm and a TiCN film having an average film thickness of 0.1 to 0.5  $\mu$ m are alternately laminated, and the upper film has an aluminum oxide film.
- (2) The coated cutting tool described in (1), wherein the upper film has an intermediate film comprising at least one metal compound selected from the group consisting of a carbide of Ti, a carboxide of Ti, a carboxide of Ti, a carboxide containing Ti and Al and a carbonitroxide containing Ti and Al, and the intermediate film is contacted with the lower film.
- (3) The coated cutting tool described in (1) or (2), wherein a hardness of the alternately laminated film is 28 GPa to 40 GPa.
- (4) The coated cutting tool described in any one of (1) to (3), wherein a tensile residual stress of the TiCN film in the alternately laminated film measured by a  $\sin^2 \Psi$  method of an X-ray diffraction measurement is 400 MPa or less.
- (5) The coated cutting tool described in any one of (1) to (4), wherein the aluminum oxide film is an  $\alpha$  type aluminum oxide film.
- (6) The coated cutting tool described in any one of (1) to (5), wherein an arithmetic mean roughness Ra of the surface of the upper film on a side opposite to a surface in contact with the lower film is 0.03 to 0.3  $\mu$ m in the coated cutting tool.
- (7) The coated cutting tool described in any one of (1) to (6), wherein an average film thickness of the film is 3.5 to  $18 \mu m$ .
- (8) The coated cutting tool described in any one of (1) to (7), wherein an average film thickness of the lower film is 2 to 13  $\mu$ m.
- (9) The coated cutting tool described in any one of (1) to (8), wherein an average film thickness of the TiN film in the alternately laminated film is 10 to 250 nm.
- (10) The coated cutting tool described in any one of (1) to (9), wherein an average film thickness of the TiCN film in the alternately laminated film is 0.15 to 0.4  $\mu$ m.
- (11) The coated cutting tool described in any one of (1) to (10), wherein an average film thickness of the alternately laminated film is 2 to 15  $\mu$ m.
- (12) The coated cutting tool described in any one of (1) to (11), wherein a number of the laminated layers of the TiN film and the TiCN film of the alternately laminated film is 4 to 120 layers.
- (13) The coated cutting tool described in any one of (2) to (12), wherein an average film thickness of the intermediate film is 0.1 to 2  $\mu$ m.
- (14) The coated cutting tool described in any one of (2) to (13), wherein the intermediate film is a film comprising TiC, TiCO, TiCNO, TiAlCO or TiAlCNO.

#### 45 EFFECTS OF THE INVENTION

**[0011]** The coated cutting tool of the present invention shows excellent wear resistance and chipping resistance. When the coated cutting tool of the present invention is used, in a cutting device to which the tool has been attached, recession of a position of the cutting edge of the tool by wear or chipping can be restrained, change in machining dimension of a work piece material is unlikely caused, and a number of correction times of the position of the cutting edge of the tool can be reduced, whereby it leads to improvement in productivity of various kinds of processing products.

#### EMBODIMENTS TO CARRY OUT THE INVENTION

## <sup>55</sup> [Coated cutting tool]

**[0012]** The coated cutting tool of the present invention comprises a cemented carbide substrate, and a coating formed on the surface of the cemented carbide substrate by the chemical vapor deposition method.

## [Cemented carbide substrate]

**[0013]** The above-mentioned cemented carbide substrate is, for example, a cemented carbide obtained by sintering a mixed powder of a hard phase-forming powder comprising WC, and at least one selected from the group consisting of a carbide, a nitride and a carbonitride of Ti, Zr, Hf, V, Nb, Ta, Cr, Mo and W and mutual solid solutions thereof, and a binder phase-forming powder comprising Co.

**[0014]** The cemented carbide thus obtained is constituted by a hard phase of WC and a binder phase comprising Co as a main component, or, a hard phase of WC, a hard phase comprising at least one selected from the group consisting of a carbide, a nitride and a carbonitride of Ti, Zr, Hf, V, Nb, Ta, Cr, Mo and W and mutual solid solutions thereof, and a binder phase comprising Co as a main component.

[0015] Also, if a  $\beta$ -free layer comprising a hard phase of WC and a binder phase containing Co as a main component is formed at the neighbor of the surface of the cemented carbide substrate comprising a hard phase of WC, a hard phase ( $\beta$  phase) comprising at least one selected from the group consisting of a carbide, a nitride, a carbonitride each containing two or more kinds of metal elements selected from the group consisting of Ti, Zr, Hf, V, Nb, Ta, Cr, Mo and W, and mutual solid solutions thereof, and a binder phase containing Co as a main component, the coated cutting tool of the present invention shows excellent toughness, and shows excellent wear resistance for the long time of the use without generating uneven wear, fracture, etc.

[0016] If the average thickness of the  $\beta$ -free layer is less than 1  $\mu$ m, fracture resistance of the coated cutting tool of the present invention tends to be lowered, while if it exceeds 30 um, plastic deformation resistance of the coated cutting tool tends to be lowered. Therefore, in the present invention, the average thickness of the  $\beta$ -free layer is preferably set to 1 to 30  $\mu$ m, and among these, the average thickness of the  $\beta$ -free layer is more preferably 5 to 25  $\mu$ m in the viewpoint of improving fracture resistance.

**[0017]** In addition, a thickness of the cemented carbide substrate explained above is not particularly limited, and generally 1 to 120 mm.

[Film]

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**[0018]** The film constituting the coated cutting tool of the present invention is formed by the chemical vapor deposition method. If it is formed by the chemical vapor deposition method, adhesion strength between the substrate and the film can be heightened. This is because, it can be considered that the temperature of the chemical vapor deposition method is high, so that diffusion of the cemented carbide substrate component is generated at the interface between the substrate and the film. Therefore, if the film is formed by the chemical vapor deposition method, a coated cutting tool excellent in peeling resistance and chipping resistance can be obtained as compared with the film formed by the physical vapor deposition method.

[0019] If an average film thickness of the film constituting the coated cutting tool of the present invention is less than 3  $\mu$ m, wear resistance of the coated cutting tool is lowered, while if it exceeds 20  $\mu$ m, cutting edge is likely chipped. Therefore, in the present invention, an average film thickness of the film is set to be 3 to 20  $\mu$ m.

**[0020]** Incidentally, in the present specification, the average film thickness can be obtained by measuring thicknesses at the optional five portions from the cross section of the film using a transmission type electron microscope, and obtaining an average value thereof. Hereinafter the same is applied.

**[0021]** Also, in the coated cutting tool of the present invention, an average film thickness of the film is preferably 3.5 to 18  $\mu$ m in the view points of accomplishing excellent wear resistance and restraining chipping.

<Lower film>

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[0022] The film is constituted by the lower film and the upper film, and the lower film is formed on the surface of the cemented carbide substrate by the chemical vapor deposition method. If an average film thickness of the lower film is less than 2  $\mu$ m, wear resistance of the coated cutting tool is lowered, while if it exceeds 15  $\mu$ m, chipping is likely caused from the peeled portion of the film as a starting point. Therefore, in the present invention, the average film thickness of the lower film is to be 2 to 15  $\mu$ m.

[0023] In the coated cutting tool of the present invention, the average film thickness of the lower film is preferably 2 to 13  $\mu$ m in the view points of accomplishing excellent wear resistance and restraining chipping.

(Alternately laminated film)

**[0024]** Such a lower film comprises an alternately laminated film in which a TiN film having an average film thickness of 10 to 300 nm and a TiCN film having an average film thickness of 0.1 to 0.5  $\mu$ m are alternately laminated. The TiN film is excellent in toughness, and the TiCN film is excellent in wear resistance, so that by using the alternately laminated

[0025] In the alternately laminated film, it is extremely difficult to form the TiN film having an average film thickness of less than 10 nm by the chemical vapor deposition method, while if an average film thickness of the TiN film exceeds 300 nm, wear resistance of the coated cutting tool is lowered. From this reason, the average film thickness of the TiN film of the alternately laminated film is to be 10 to 300 nm. Incidentally, the average film thickness of the TiN film is preferably 10 to 250nm in the view point of wear resistance of the coated cutting tool.

[0026] Next, in the alternately laminated film, if an average film thickness of the TiCN film is less than 0.1  $\mu$ m, wear resistance of the coated cutting tool of the present invention is lowered, while if the average film thickness thereof exceeds 0.5  $\mu$ m, chipping is likely caused from the peeled portion of the alternately laminated film as a starting point. From the above, an average film thickness of the TiCN film of the alternately laminated film is set to 0.1 to 0.5  $\mu$ m. Incidentally, in the coated cutting tool, the average film thickness of the TiCN film is preferably 0.15 to 0.4  $\mu$ m in the view points of accomplishing excellent wear resistance and restraining chipping.

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[0027] If the average film thickness of the whole alternately laminated film in which the TiN film and the TiCN film are alternately laminated as explained above is less than 2  $\mu$ m, wear resistance of the coated cutting tool of the present invention is lowered, while if the average film thickness exceeds 15  $\mu$ m, chipping is likely caused from the peeled portion of the film as a starting point. Therefore, the average film thickness of the whole alternately laminated film is preferably 2 to 15  $\mu$ m.

**[0028]** Further, a number of the laminated layers of the TiN film and the TiCN film in the alternately laminated film is generally 4 to 120 layers, preferably 10 to 100 layers, in the viewpoint of wear resistance of the coated cutting tool of the present invention. Here, the number of the laminated layers represents a number of layers in total when the TiN film is counted as one layer and the TiCN film is counted as one layer.

**[0029]** When the alternately laminated film is contacted with the cemented carbide substrate, either of the TiN film or the TiCN film may be contacted with the substrate. Incidentally, in the viewpoint of adhesiveness, the TiN film is preferably contacted with the substrate.

**[0030]** Also, as mentioned hereinbelow, an upper film is formed on the lower film, and when the alternately laminated film is contacted with the upper film, either of the TiN film or the TiCN film may be contacted with the upper film. Incidentally, in the viewpoint of adhesiveness, the TiCN film is preferably contacted with the upper film.

**[0031]** If the hardness of the alternately laminated film exceeds 40 GPa, toughness of the alternately laminated film is lowered, and a tendency can be observed that chipping is likely caused from the peeled portion of the alternately laminated film as a starting point. To the contrary, if the hardness of the alternately laminated film is low and is less than 28 GPa, wear resistance of the coated cutting tool of the present invention tends to be lowered. Therefore, the hardness of the alternately laminated film is preferably 28 GPa to 40 GPa.

**[0032]** The hardness of the alternately laminated film can be adjusted by average film thicknesses of the TiN film and the TiCN film of the alternately laminated film. More specifically, by making average film thicknesses of the respective films of the TiN film and the TiCN film thin, the hardness of the alternately laminated film can be made high. Incidentally, the hardness of the alternately laminated film can be measured, in the coated cutting tool of the present invention, by applying a diamond indenter to the surface of the alternately laminated film which is obtained by removing the films (the upper film, etc.) coated on the surface side of the alternately laminated film by polishing.

**[0033]** A coefficient of thermal expansion of the alternately laminated film is larger than a coefficient of thermal expansion of the cemented carbide substrate, so that a tensile residual stress is generated to the alternately laminated film when the coated cutting tool of the present invention is manufactured. Further, TiN and TiCN have a different coefficient of thermal expansion, so that the tensile residual stress of the TiN film and the tensile residual stress of the TiCN film are different from each other.

[0034] Thus, when residual stress of the TiCN film of the alternately laminated film having sufficiently large average film thickness to measure the residual stress is measured, if the tensile residual stress of the TiCN film exceeds 400 MPa, a tendency of lowering in chipping resistance can be observed. From this reason, tensile residual stress of the TiCN film of the alternately laminated film according to the present invention is preferably 400 MPa or less. Also, tensile residual stress of the TiCN film is generally 100 MPa to 600 MPa.

[0035] The tensile residual stress of the TiCN film of the alternately laminated film can be measured by the conventionally known  $\sin^2\Psi$  method of the X-ray diffraction measurement. Also, the Young's modulus of the TiCN film necessary for calculating the tensile residual stress by the  $\sin^2\Psi$  method is measured by the dynamic hardness tester, etc. In addition, the Poisson's ratio of the TiCN film is to be made 0.2. Incidentally, if tensile residual stress of the TiCN film in the alternately laminated film cannot be measured since the TiCN film which is not included in the alternately laminated film is formed at the side nearer to the surface which is opposed to the substrate of the coated cutting tool of the present invention than the alternately laminated film, the TiCN film which is not included in the alternately laminated film is removed by polishing, etc., to measure the stress.

[0036] Incidentally, for making the tensile residual stress of the TiCN film 400 MPa or less, the alternately laminated film is so constituted that the TiN film having an average film thickness of 10 to 300 nm and the TiCN film having an

average film thickness of 0.1 to 0.5  $\mu m$  are alternately laminated.

<upper film>

**[0037]** An upper film is formed on the surface of the lower film which constitutes the coated cutting tool of the present invention, which surface is opposed to the side of the surface contacting with the cemented carbide substrate. The upper film of the present invention has an aluminum oxide film, so that crater wear resistance of the coated cutting tool is improved.

[0038] If the average film thickness of the upper film is less than 1  $\mu$ m, crater wear resistance of the rake face of the coated cutting tool of the present invention is not improved, while if the average film thickness thereof exceeds 10  $\mu$ m, cutting edge is likely fractured. From this reason, an average film thickness of the upper film is to be 1 to 10  $\mu$ m.

[0039] Incidentally, a crystal form of the aluminum oxide film in the upper film is not particularly limited to  $\alpha$  type,  $\kappa$  type or  $\gamma$  type, and  $\alpha$  type which is stable at high temperatures is preferred. In particular, when the cutting edge becomes high temperatures such as high speed cutting of carbon steel or alloy steel, if the aluminum oxide film is an  $\alpha$  type aluminum oxide film, fracture or chipping unlikely occurs.

[0040] If the average film thickness of such an aluminum oxide film is less than 0.5  $\mu$ m, crater wear resistance at the rake face of the coated cutting tool is not improved, while if the average film thickness thereof exceeds 10  $\mu$ m, cutting edge is likely fractured. From this reason, an average film thickness of the aluminum oxide film is preferably 0.5 to 10  $\mu$ m, more preferably 0.5 to 8  $\mu$ m.

(Intermediate film)

**[0041]** The coated cutting tool of the present invention has an intermediate film comprising at least one metal compound selected from the group consisting of a carbide of Ti, a carboxide of Ti, a carbonitroxide of Ti, a carboxide containing Ti and Al and a carbonitroxide containing Ti and Al, as a layer constituting the upper film, and the intermediate film is preferably contacted with the above-mentioned lower film.

[0042] When an intermediate film with such a constitution is present, adhesiveness between the lower film and the upper film is improved. Incidentally, an average film thickness of the intermediate film is generally 0.1 to 2  $\mu$ m, more preferably 0.3 to 1.0  $\mu$ m. The intermediate film may be a laminated film of a plural number of the films comprising the various metal compounds mentioned above.

[0043] As the above-mentioned intermediate film, there may be specifically mentioned a film comprising TiC, TiCO, TiCNO, TiAlCO or TiAlCNO. Among these, a film comprising a carbonitroxide containing Ti and Al is more preferred. As the intermediate film comprising a carbonitroxide containing Ti and Al, there may be specifically mentioned TiAlCNO. [0044] Moreover, when the intermediate film is formed at the surface opposite to the surface contacting with the cemented carbide substrate of the lower film, and the aluminum oxide film is formed at the surface opposite to the surface contacting with the lower film of the intermediate film, adhesiveness of the aluminum oxide film and the lower film is improved so that it is more preferred.

(Surface film)

**[0045]** Also, to make discrimination of the used corner of the coated cutting tool easy, a surface film comprising at least one selected from the group consisting of a carbonitride and a nitride of Ti is more preferably formed on the surface of the aluminum oxide film at the side opposite to the surface contacting with the intermediate film.

[0046] Incidentally, the surface film may be a laminated film comprising a plural number of the film comprising a carbonitride of Ti and the film comprising a nitride of Ti. Also, the forming method of the surface film is conventionally known, and various known conditions of the chemical deposition may be employed. An average film thickness of such a surface film is generally 0.1 to 5  $\mu$ m.

[Preparation method of film]

**[0047]** As a preparation method of the respective films constituting the film of the coated cutting tool of the present invention, there may be mentioned, for example, the following methods.

**[0048]** The TiN film which constitutes the alternately laminated film of the lower film can be formed by the chemical vapor deposition method using the starting gas composition comprising  $TiCl_4$ : 5.0 to 10.0 mol%,  $N_2$ : 20 to 60 mol% and  $H_2$ : the remainder, at a temperature of 850 to 920°C and a pressure of 100 to 350 hPa.

**[0049]** Also, the TiCN film which constitutes the alternately laminated film can be formed by the chemical vapor deposition method using the starting gas composition comprising  $TiCl_4$ : 10 to 15 mol%,  $CH_3CN$ : 1 to 3 mol%,  $N_2$ : 0 to 20 mol% and  $H_2$ : the remainder, at a temperature of 850 to 920°C and a pressure of 60 to 100 hPa.

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[0050] The  $\alpha$  type aluminum oxide film in the upper film can be formed by the chemical vapor deposition method using the starting gas composition comprising AlCl<sub>3</sub>: 2.1 to 5.0 mol%, CO<sub>2</sub>: 2.5 to 4.0 mol%, HCl: 2.0 to 3.0 mol%, H<sub>2</sub>S: 0.28 to 0.45 mol% and H<sub>2</sub>: the remainder, at a temperature of 900 to 1000°C and a pressure of 60 to 80 hPa.

[0051] Incidentally, the  $\kappa$  type aluminum oxide film can be formed by the chemical vapor deposition method using the starting gas composition comprising AlCl $_3$ : 2.0 to 5.0 mol%, CO $_2$ : 4.2 to 6.0 mol%, CO: 3.0 to 6.0 mol%, HCI: 3.5 to 5.0 mol%, H $_2$ S: 0.3 to 0.5 mol% and H $_2$ : the remainder, at a temperature of 900 to 1020°C and a pressure of 60 to 80 hPa. [0052] The TiAlCNO film, TiC film, TiCO film, TiCNO film and TiAlCO film which can be mentioned as the intermediate film may be formed as follows.

**[0053]** The TiAlCNO film can be formed by the chemical vapor deposition method using the starting gas composition comprising TiCl<sub>4</sub>: 3.0 to 5.0 mol%, AlCl<sub>3</sub>: 1.0 to 2.0 mol%, CO: 0.4 to 1.0 mol%, N<sub>2</sub>: 30 to 40 mol% and H<sub>2</sub>: the remainder, at a temperature of 975 to  $1025^{\circ}$ C and a pressure of 90 to 110 hPa.

**[0054]** The above-mentioned TiC film can be formed by the chemical vapor deposition method using the starting gas composition comprising  $TiCl_4$ : 1.0 to 3.0 mol%,  $CH_4$ : 4.0 to 6.0 mol% and  $H_2$ : the remainder, at a temperature of 990 to 1030°C and a pressure of 50 to 100 hPa.

**[0055]** The above-mentioned TiCO film can be formed by the chemical vapor deposition method using the starting gas composition comprising TiCl<sub>4</sub>: 1.2 to 3.2mol%, CO: 3.0 to 5.0mol% and H<sub>2</sub>: the remainder, at a temperature of 960 to 1025°C and a pressure of 170 to 210 hPa.

**[0056]** The above-mentioned TiCNO film can be formed by the chemical vapor deposition method using the starting gas composition comprising  $TiCl_4$ : 1.0 to 3.0mol%, CO: 2.0 to 4.0 mol%,  $N_2$ : 4 to 6 mol% and  $H_2$ : the remainder, at a temperature of 960 to 1025°C and a pressure of 60 to 100 hPa.

**[0057]** The above-mentioned TiAlCO film can be formed by the chemical vapor deposition method using the starting gas composition comprising TiCl<sub>4</sub>: 0.5 to 1.5 mol%%, AlCl<sub>3</sub>: 3.0 to 5.0 mol%, CO: 2.0 to 4.0 mol% and H<sub>2</sub>; the remainder, at a temperature of 975 to 1025°C and a pressure of 60 to 100 hPa.

[Surface roughness at the surface of the coated cutting tool]

**[0058]** By adjusting the surface roughness of the cemented carbide substrate constituting the coated cutting tool of the present invention and forming conditions of the films, an arithmetic mean roughness Ra of the surface of the coated cutting tool, more specifically, the surface opposed to the side of the face contacting with the lower film of the upper film of the tool can be adjusted. According to this procedure, welding of the work piece material to the coated cutting tool can be restrained.

[0059] The arithmetic mean roughness Ra at the surface of the coated cutting tool is usually 0.03 to 0.5  $\mu$ m, and preferably 0.03 to 0.3  $\mu$ m.

[0060] Also, it is preferred that the surface of the above-mentioned coated cutting tool is subjected to surface treatment such as polishing by a grindstone and polishing by wet blast, to adjust the arithmetic mean roughness Ra at the surface of the coated cutting tool to 0.03 to 0.3  $\mu$ m. Incidentally, in the present specification, the arithmetic mean roughness Ra is a value measured according to JIS B0601 (2001).

## **EXAMPLES**

[0061] In the following, the present invention is explained in more detail by referring to Examples.

[Production of coated cutting tool]

<sup>5</sup> **[0062]** A mixed powder comprising 89% by weight of WC powder having an average particle size of 4.5 μm, 2% by weight of TiCN powder having an average particle size of 1.5 μm, 2% by weight of (Ta,Nb)C powder having an average particle size of 1.5 μm and 7% by weight of Co powder having an average particle size of 1.5 μm was sintered to obtain a WC-(Ti,W,Ta,Nb)(C,N)-Co series cemented carbide.

[0063] The cemented carbide was processed to an ISO standard CNMG 120412 shaped insert to obtain a cemented carbide substrate. Incidentally, at the neighbor of the surface of the cemented carbide substrate, a  $\beta$ -free layer comprising WC and Co had been formed. An average thickness of the  $\beta$ -free layer at the flank surface of the cemented carbide substrate was 15  $\mu$ m. On the cemented carbide substrate was formed a film having a film constitution shown in the following Table 1 by the chemical vapor deposition method (excluding Comparative product 10).

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10	-			Total film thickness of	whole film (µm)	3.0	20.0	17.0	18.0	10.0	14.1	14.2	15.0	15.0	15.0	15.0			
15				Whole layer	mmoniess (μm) of upper film	1.0	8.0	2.0	10.0	1.0	7.0	7.0	7.0	7.0	7.0	7.0			
20						Surface film	TiN (μm)	0.2	0.5	0.5	0.5	0.2	0.5	0.5	0.5	0.5	0.5	0.5	
			Upper film	Surface film	TiCN (μm)	0.0	0.5	0.0	0.5	0.0	0.5	0.5	0.5	0.5	0.5	0.5			
25	1]	mii	E.	Ē	Film		Aluminu m oxide film	Al <sub>2</sub> O <sub>3</sub> (μm)	0.5~(lpha)	6.0~(lpha)	1.0 $(\alpha)$	8.0~(lpha)	0.5~(lpha)	5.0~(lpha)	5.0~(lpha)	5.0 (K)	5.0~(lpha)	5.0~(lpha)	5.0~(lpha)
30 35	[Table 1]	Fi		Intermediate film	TiAlCNO (μm)	0.3	1.0	0.5	1.0	0.3	1.0	1.0	1.0	1.6	1.0	1.0			
40				Whole layer	(µm) of lower film	2.0	12.0	15.0	8.0	9.0	7.14	7.2	8.0	8.0	8.0	8.0			
45			Lower film	nated film	Number of Iaminated Iayers	10	09	100	40	30	28	48	40	20	40	40			
			1	Altemately laminated film	TiCN (µm)	0.20	0.20	0.20	0.20	0:30	0:50	0.10	0.20	0:50	0.20	0.20			
50	ļ			Alte	TiN (mn)	200	200	100	200	300	10	200	200	300	200	200			
55				Sample No.		Present product 1	Present product 2	Present product 3	Present product 4	Present product 5	Present product 6	Present product 7	Present product 8	Present product 9	Present product 10	Present product 11			

5		Surface treatment			Polishing	Polishing	Polishing	Polishing	Polishing	Polishing	Polishing	Polishing	Polishing	
10				Total film thickness of	whole film (μm)	14.2	2.2	20.2	15.3	13.0	15.3	18.0	12.8	17.0
15				Whole layer	uncariess (µm) of nber film	6.0	1.0	7.0	7.0	7.0	7.0	2.0	0.8	11.0
20				Surface film	TiN (μm)	0.5	0.2	9.0	9.0	9.0	9.0	6.0	0.2	0.5
			Upper film	Surface film	TiCN (mm)	0.5	0.0	0.5	0.5	0.5	0.5	0.0	0.0	0.5
25	(pər	ed) n	ח	Aluminum oxide film	Al <sub>2</sub> O <sub>3</sub> (μm)	4.0 (α)	0.5 (α)	5.0~(lpha)	5.0~(lpha)	5.0~(lpha)	$5.0 (\alpha)$	1.0 (α)	0.3 (α)	9.0 (α)
3 <i>0</i> 35	(continued)	Film		Intermediate film	TiAlCNO (μm)	1.0	0.3	1.0	1.0	1.0	1.0	0.5	0.3	1.0
40				Whole layer	(μm) of lower film	8.2	1.2	13.2	8.25	0.9	8.25	16.0	12.0	0.9
45			Lower film	inated film	Number of laminated layers	2	4	44	90	48	77	08	09	0ε
			_	Alternately laminated film	TiCN (m <sup>rd</sup> )	8.00	0:30	08.0	0.20	90.0	99.0	0.20	0.20	0.20
50				Alten	NiT (mu)	200	300	300	350	200	200	200	200	200
55				Sample No.		Comparative product 1	Comparative product 2	Comparative product 3	Comparative product 4	Comparative product 5	Comparative product 6	Comparative product 7	Comparative product 8	Comparative product 9

					Π	<del> </del>
5			Surface treatment			ninated film is ace film is/are
10			Total film thickness of	whole film (µm)		alternately lam is/are the surfa
15			Whole layer	unckriess (μm) of upper film	thod	CN film in the TiN film which oxide film.
20			Surface film	TiN (μm)	deposition me	bstrate, the Ti SN film and/or the aluminum
		Upper film	Surface film	TiCN (μm)	N film (average film thickness: 10.0 $\mu$ m) by physical deposition method	d carbide su n, and the TiC
25	Film		Aluminum oxide film	Al <sub>2</sub> O <sub>3</sub> (μm)		the cemente num oxide film th the interme
30	L		Intermediate film	TiAlCNO (μm)	rage film thickr	*The film thicknesses are all an average film thickness. In the obtained films, the TiN film in the alternately laminated film is contacted with the cemented carbide substrate, the TiCN film in the alternately laminated film is contacted with the aluminum oxide film, and the TiCN film and/or TiN film which is/are the surface film is/are film is/are film is/are film is/are film is/are the surface on the surface opposed to the side of the surface contacting with the intermediate film of the aluminum oxide film.
35			Whole layer thickness (μm) of lower film		TiAIN film (ave	ninated film is n is contacted le of the surfac
40		ılm		Number of laminated layers		thickness. emately lar rmediate filr ed to the sid
45		Lower film	Altemately laminated film			verage film m in the alt Im, the inte face oppose
50			ıltemately	TiCN (hm)		re all an ave the TiN film mediate film on the surface
				TiN (mm)		knesses a ed films, τ the inter his order o
55			Sample No.		Comparative product 10	*The film thicknesses are all an average film thickness. *In the obtained films, the TiN film in the alternately lamina contacted with the intermediate film, the intermediate film is laminated in this order on the surface opposed to the side of

[0064] In Table 1, a number of the laminated layers of the alternately laminated film is shown to be a total number of the layers be formed, when the TiN film is counted as one layer and the TiCN film is counted as one layer.

[0065] The TiN film of the alternately laminated film was formed by using the starting gas composition comprising  $TiCl_4$ : 9.0mol%,  $N_2$ : 40mol% and  $H_2$ : the remainder, under the conditions of a temperature of 900°C and a pressure of 160 hPa.

**[0066]** The TiCN film of the alternately laminated film was formed by using the starting gas composition comprising TiCl<sub>4</sub>: 10.8mol%, CH<sub>3</sub>CN: 1.3mol%, N<sub>2</sub>: 13.5mol% and H<sub>2</sub>: the remainder, under the conditions of a temperature of 900°C and a pressure of 90 hPa.

**[0067]** Under these coating conditions, the TiN film and the TiCN film were alternately laminated until these became a predetermined number of the laminated layers. Incidentally, the lower films comprise an alternately laminated film alone in both of Present products and Comparative products.

**[0068]** The TiAlCNO film of the intermediate film was formed by using the starting gas composition comprising  $TiCl_4$ : 4.0 mol%,  $AlCl_3$ : 1.2 mol%, CO: 0.6 mol%,  $N_2$ : 34 mol% and  $H_2$ : the remainder, under the conditions of a temperature of 1000°C and a pressure of 100 hPa.

[0069] The a type aluminum oxide film of the upper film was formed by using the starting gas composition comprising AlCl<sub>3</sub>: 2.3 mol%, CO<sub>2</sub>: 3.6 mol%, HCl: 2.0 mol%, H<sub>2</sub>S: 0.3 mol% and H<sub>2</sub>: the remainder, under the conditions of a temperature of 1000°C and a pressure of 70 hPa.

[0070] The  $\kappa$  type aluminum oxide film (Present product 8) of the upper film was formed by using the starting gas composition comprising AlCl<sub>3</sub>: 2.5 mol%, CO<sub>2</sub>: 4.5 mol%, CO: 4.4 mol%, HCl: 4.0 mol%, H<sub>2</sub>S: 0.4 mol% and H<sub>2</sub>: the remainder, under the conditions of a temperature of 1000°C and a pressure of 70 hPa.

**[0071]** The TiCN film of the upper film was formed by using the starting gas composition comprising TiCl<sub>4</sub>: 7.3 mol%,  $N_2$ : 11.6 mol%,  $CH_3CN$ : 1.2 mol% and  $H_2$ : the remainder, under the conditions of a temperature of 1000°C and a pressure of 90 hPa.

[0072] The TiN film of the upper film was formed by using the starting gas composition comprising  $TiCl_4$ : 9.0 mol%,  $N_2$ : 40 mol% and  $H_2$ : the remainder, under the conditions of a temperature of 1000°C and a pressure of 160 hPa.

[0073] In Comparative product 10, the TiAIN film was formed by the physical vapor deposition method, the TiAIN film was peeled off from the cemented carbide substrate at the cutting edge of the coated cutting tool before subjecting to the machining test, so that the cutting properties could not be evaluated. In addition, measurement of an arithmetic mean roughness Ra, etc., of the surface of the sample shown below was not also carried out. The reason why the TiAIN film was naturally peeled off before the machining test can be considered that adhesive strength between the TiAIN film and the substrate was not sufficient, and compressive residual stress of the TiAIN film was extremely high.

## [Measurement of characteristics]

**[0074]** An arithmetic mean roughness Ra at the surface of the coated cutting tool sample (the surface at the side opposite to the surface contacting with the lower film of the upper film) obtained by the chemical vapor deposition method as mentioned above was measured by using a surface property measuring instrument (SURFPAK-SV) manufactured by Mitutoyo Corporation.

**[0075]** Further, the upper film at the sample surface was removed by diamond lap grinding, and a hardness of the alternately laminated film revealed to the surface was measured by using a dynamic hardness meter (NANOINDENTER manufactured by MTS Corporation).

**[0076]** Further, a tensile residual stress of the TiCN film in the alternately laminated film was measured by using an X-ray diffraction device (RINT-TTRIII) manufactured by RIGAKU according to the  $\sin^2 \Psi$  method.

[0077] These measurement results are shown in the following Table 2.

[Table 2]

Tensile residual stress (MPa) of TiCN film Hardness (GPa) of Arithmetic mean roughness Sample No. constituting alternately Ra (μm) of sample surface alternately laminated film laminated film Present product 1 280 0.05 32.6 Present product 2 183 0.30 35.0 Present product 3 149 0.35 39.5 Present product 4 181 0.48 31.2 Present product 5 351 0.13

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(continued)

	Sample No.	Tensile residual stress (MPa) of TiCN film constituting alternately laminated film	Arithmetic mean roughness Ra (μm) of sample surface	Hardness (GPa) of alternately laminated film
•	Present product 6	378	0.15	29.0
•	Present product 7	104	0.15	39.8
	Present product 8	180	0.10	34.7
	Present product 9	396	0.20	28.1
	Present product 10	175	0.35	35.2
	Present product 11	15	0.28	35.4
	Comparative product 1	500	0.25	26.8
	Comparative product 2	396	0.10	28.2
	Comparative product 3	232	0.34	28.0
	Comparative product 4	383	0.15	26.5
	Comparative product 5	118	0.13	38.0
	Comparative product 6	415	0.20	27.3
	Comparative product 7	154	0.50	34.7
•	Comparative product 8	177	0.15	34.8
	Comparative product 9	215	0.35	35.1

**[0078]** Compositions of the films of the coated cutting tools of Present products 1 to 11 and Comparative products 1 to 9 were measured by an energy dispersion X-ray analyzer attached to a transmission electron microscope. Also, when an average film thickness (an average value of 5 portions) of the respective films was measured from the cross-section of the film by using the transmission electron microscope, then, each showed the same value as the target film thickness (the value shown in Table 1). Moreover, when a crystal form of the aluminum oxide film was examined by using an X-ray diffractometer, then, the objective crystal form (that shown in Table 1) could be obtained in each case.

## [Cutting test]

**[0079]** As a machining test, two samples were prepared for each sample, cylindrical S53C (hardness:  $H_B270$ ) having a diameter of 120 mm and a length of 400 mm was used as a work piece material, and an edge cutting was carried out by the following mentioned cutting conditions.

<Cutting conditions>

## <sub>45</sub> [0080]

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Cutting speed: 200 m/min, Depth of cut: 2 mm, Feed: 0.3 mm/rev,

Cutting atmosphere: wet (using water-soluble emulsion),

Each cutting time: 15 min, Number of test: twice

After completion of the machining test, flank wear width was measured, and the presence or absence of chipping was confirmed. The machining test was carried out twice. In the following Table 3, an average value of flank wear width of the samples and a number of the samples at which chipping occurred are shown.

## [0081]

[Table 3]

	Cutting test results						
Sample No.	Average value (mm) of flank wear width	Number of samples at which chipping occurred (number)					
Present product 1	0.18	0					
Present product 2	0.15	0					
Present product 3	0.13	0					
Present product 4	0.17	0					
Present product 5	0.20	0					
Present product 6	0.20	0					
Present product 7	0.15	0					
Present product 8	0.18	0					
Present product 9	0.20	0					
Present product 10	0.20	0					
Present product 11	0.15	0					
Comparative product 1	0.25	0					
Comparative product 2	0.35	0					
Comparative product 3	0.80	2					
Comparative product 4	0.25	0					
Comparative product 5	0.28	0					
Comparative product 6	0.75	1					
Comparative product 7	0.92	2					
Comparative product 8	0.54	1					
Comparative product 9	1.22	2					

[0082] From the results shown in Table 3, it can be understood that Present products (the coated cutting tool of the present invention) are superior to Comparative products in wear resistance and chipping resistance. Comparative products show an average value of the flank wear width of 0.25 mm or more, so that these are lower wear resistance than those of Present products. Among these, in Comparative products 3 and 6 to 9 in which chipping has generated, it can be understood that the average value of the flank wear width is 0.54 mm or more, in particular, wear resistance is low. [0083] On the other hand, no chipping was observed in Present products, and an average value of flank wear width was 0.20 mm or less, so that it can be understood that the coated cutting tools of the present invention are excellent in chipping resistance and wear resistance.

## UTILIZABILITY IN INDUSTRY

**[0084]** When the coated cutting tool of the present invention is used for cutting, in the cutting device to which the tool is attached, recession of the position of the cutting edge of the tool by chipping or wear can be restrained. Thus, according to the present invention, change in machining dimension of a work piece material, and a number of operations for correction of the position of the cutting edge of the tool which has conventionally been frequently carried out to suppress the same can be reduced, whereby it leads to improvement in processing productivity.

## Claims

1. A coated cutting tool comprising a cemented carbide substrate and a film formed on a surface of the substrate by

a chemical vapor deposition method,

wherein an average film thickness of the film is 3 to 20  $\mu$ m, the film comprises a lower film having an average film thickness of 2 to 15  $\mu$ m formed on the surface of the substrate, and an upper film having an average film thickness of 1 to 10  $\mu$ m formed on a surface of the lower film on a side opposite to a surface in contact with the substrate, the lower film has an alternately laminated film in which a TiN film having an average film thickness of 10 to 300 nm and a TiCN film having an average film thickness of 0.1 to 0.5  $\mu$ m are alternately laminated, and the upper film has an aluminum oxide film.

- 2. The coated cutting tool according to Claim 1, wherein
- the upper film has an intermediate film comprising at least one metal compound selected from the group consisting of a carbide of Ti, a carboxide of Ti, a carboxide of Ti, a carboxide containing Ti and Al and a carbonitroxide containing Ti and Al, and

the intermediate film is contacted with the lower film.

- 15 **3.** The coated cutting tool according to Claim 1 or 2, wherein a hardness of the alternately laminated film is 28 GPa to 40 GPa.
  - **4.** The coated cutting tool according to any one of Claims 1 to 3, wherein a tensile residual stress of the TiCN film in the alternately laminated film measured by a sin<sup>2</sup> Ψ method of an X-ray diffraction measurement is 400 MPa or less.
  - 5. The coated cutting tool according to any one of Claims 1 to 4, wherein the aluminum oxide film is an  $\alpha$  type aluminum oxide film.
  - 6. The coated cutting tool according to any one of Claims 1 to 5, wherein an arithmetic mean roughness Ra of the surface of the upper film on a side opposite to a surface in contact with the lower film is 0.03 to 0.3 μm in the coated cutting tool.
  - 7. The coated cutting tool according to any one of Claims 1 to 6, wherein an average film thickness of the film is 3.5 to 18  $\mu$ m.
  - 8. The coated cutting tool according to any one of Claims 1 to 7, wherein an average film thickness of the lower film is 2 to 13  $\mu$ m.
- **9.** The coated cutting tool according to any one of Claims 1 to 8, wherein an average film thickness of the TiN film in the alternately laminated film is 10 to 250 nm.
  - **10.** The coated cutting tool according to any one of Claims 1 to 9, wherein an average film thickness of the TiCN film in the alternately laminated film is 0.15 to 0.4 μm.
- 40 **11.** The coated cutting tool according to any one of Claims 1 to 10, wherein an average film thickness of the alternately laminated film is 2 to 15 μm.
  - **12.** The coated cutting tool according to any one of Claims 1 to 11, wherein a number of laminated layers of the TiN film and the TiCN film in the alternately laminated film is 4 to 120 layers.
  - 13. The coated cutting tool according to any one of Claims 2 to 12, wherein an average film thickness of the intermediate film is 0.1 to 2  $\mu$ m.
- **14.** The coated cutting tool according to any one of Claims 2 to 13, wherein the intermediate film is a film comprising TiC, TiCO, TiCNO, TiAlCO or TiAlCNO.

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## INTERNATIONAL SEARCH REPORT

International application No.

PCT/JP2012/060651

		101/012	2012/000031					
A. CLASSIFICATION OF SUBJECT MATTER B23B27/14								
According to International Patent Classification (IPC) or to both national classification and IPC								
B. FIELDS SE	ARCHED							
	Minimum documentation searched (classification system followed by classification symbols)							
	323B27/14, B23B51/00, B23C3/16, B23P15/28, C23C16/30, C23C16/40							
Jitsuvo	Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched Jitsuyo Shinan Koho 1922-1996 Jitsuyo Shinan Toroku Koho 1996-2012 Kokai Jitsuyo Shinan Koho 1971-2012 Toroku Jitsuyo Shinan Koho 1994-2012							
Electronic data b	Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)							
C. DOCUMEN	VTS CONSIDERED TO BE RELEVANT		1					
Category*	Citation of document, with indication, where ap	propriate, of the relevant passages	Relevant to claim No.					
A	JP 10-180506 A (Kyocera Corp 07 July 1998 (07.07.1998), paragraphs [0014] to [0017] (Family: none)	.),	1-14					
A	A JP 2007-245269 A (Mitsubishi Materials Corp.), 27 September 2007 (27.09.2007), paragraphs [0006], [0009], [0027] (Family: none)							
A	2-14							
× Further do	ocuments are listed in the continuation of Box C.	See patent family annex.						
* Special cate "A" document d	ternational filing date or priority cation but cited to understand							
	to be of particular relevance  the principle or theory underlying the invention  the principle or theory underlying the invention  "X" document of particular relevance; the claimed invention cannot be							
filing date "L" document w	filing date considered novel or cannot be considered to involve an inventive							
cited to esta	cited to establish the publication date of another citation or other  "Y" document of particular relevance; the claimed invention cannot be							
special reason (as specified)  document referring to an oral disclosure, use, exhibition or other means document published prior to the international filing date but later than the priority date claimed  considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art document member of the same patent family								
Date of the actua	al completion of the international search	Date of mailing of the international sea	rch report					
03 July	y, 2012 (03.07.12)	17 July, 2012 (17.	07.12)					
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		Talanhana Na						

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## INTERNATIONAL SEARCH REPORT

International application No.
PCT/JP2012/060651

		101/012	.012/060631				
C (Continuation). DOCUMENTS CONSIDERED TO BE RELEVANT							
Category*	Citation of document, with indication, where appropriate, of the relev	ant passages	Relevant to claim No.				
А	JP 9-117806 A (Nachi-Fujikoshi Corp.), 06 May 1997 (06.05.1997), paragraph [0011] (Family: none)		1-14				
A	(Family: none)  JP 7-136808 A (Mitsubishi Materials Corp 30 May 1995 (30.05.1995), paragraph [0015] (Family: none)	o.),	1-14				

Form PCT/ISA/210 (continuation of second sheet) (July 2009)

## REFERENCES CITED IN THE DESCRIPTION

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